

**INCORPORATION OF ALKYLAMINE INTO METAL-ORGANIC
FRAMEWORKS FOR CARBON DIOXIDE CAPTURE**

A Dissertation

by

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ABSTRACT

The escalating atmospheric CO₂ concentration is one of the most urgent environmental concerns of our age. To effectively capture CO₂, various materials have been studied, such as aqueous alkanolamine solutions, zeolites, activated carbons. However, none of these materials fulfill all the crucial performance parameters for CO₂ capture materials, including but not limited to, high CO₂ selectivity, reasonable regeneration cost, and stability under the operational conditions. Metal-organic frameworks (MOFs) are an emerging class of porous materials constructed from metal-containing nodes and organic linkers. Owing to their high porosity, high surface area, and tunable functionality, MOFs have been investigated extensively to date as promising candidates to be applied in real-world CO₂ capture scenarios. Among them, alkylamine-modified MOFs exhibit significantly improved CO₂ selectivity because of the chemical interaction between the surface functionalities of MOFs and CO₂ molecules.

In most cases, alkylamine molecules are integrated into MOFs via the coordination bonds formed between open metal sites (OMSs) and amine groups. Thus, the alkylamine density as well as the corresponding CO₂ uptake in MOFs are severely restricted by the density of OMSs. To overcome this limit, other approaches to incorporating alkylamine into MOFs are highly desired. Therefore, a new method based on the Brønsted acid-base reaction between MOFs and alkylamine was explored for alkylamine grafting. The pristine MOF utilized is Cr-MIL-101-SO₃H (MIL = Materials of Institute Lavoisier), which has a high density of inherent sulfonic acid groups to interact with amine groups. A systematic optimization of the

experimental conditions involved in alkylamine tethering was conducted to maximize the CO₂ uptakes of the obtained modified MOFs. The study reveals that experimental factors, such as alkylamine structure and solvent species, play an important role in determining the alkylamine loading amount in MOFs and their accompanying CO₂ capacities. This conclusion was further applied in the case of Cr-MIL-101. With judiciously selected alkylamine and solvent species, the CO₂ uptake of the alkylamine-functionalized Cr-MIL-101 was dramatically enhanced compared with the previously reported results. The improvement was rationalized from a thermodynamic perspective.

DEDICATION

To my families and friends

For their love and support

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Contributors

This work was supervised by a dissertation committee consisting of Professor Hong-Cai Zhou [advisor], Professor Marcetta Darensbourg, Professor James Batteas of the Department of Chemistry [Home Department] and Professor Hong Liang of the Department of Mechanical Engineering [Outside Department].

The Langmuir fitting of the CO₂ adsorption isotherms and the calculation of isosteric heat of CO₂ adsorption in Chapter II were conducted by Mr. Wolfgang Verdegaaal. The breakthrough experiments and relevant calculations in Chapter III were performed by Dr. Zhigang Hu and Prof. Dr. Dan Zhao of the Department of Chemical and Biomolecular Engineering in National University of Singapore. All the other work in this dissertation was completed by the student independently.

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NOMENCLATURE

MOF	metal-organic framework
OMS	open metal site
MIL	Materials of Institute Lavoisier
DMF	<i>N, N</i> -dimethylformamide
EA	elemental analysis
EDX	energy dispersive X-ray spectroscopy
FTIR	Fourier transform infrared spectroscopy
MeOH	methanol
DCM	dichloromethane
THF	tetrahydrofuran
CH	cyclohexane
MS	mass spectroscopy
NMR	nuclear magnetic resonance
PXRD	powder X-ray diffraction
en	ethylenediamine
DETA	diethylenetriamine
TETA	triethylenetetramine
TEPA	tetraethylenepentamine
PEHA	pentaethylenehexamine
TAEA	tris(2-aminoethyl)amine
$-Q_{st}$	isosteric heat of adsorption

RT	room temperature
DRIFTS	diffuse reflectance infrared Fourier transform spectroscopy
ATR	attenuated total reflectance
SALC	symmetry-adapted linear combination

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CHAPTER I

INTRODUCTION

Human society has developed into a stage where economic prosperity is predominantly dependent on the exploitation of fossil fuels, which are the main energy sources. However, combustion of fossil fuels for power has inevitably brought the emission of large amounts of CO₂ into the atmosphere and led to global warming.¹ Therefore, it is urgent to develop materials that can effectively remove CO₂ from gas mixtures. Current CO₂ scrubbers are mostly aqueous alkanolamine solutions, which suffer from high regeneration energy due to the large portions of water in their composition.¹ This propels the investigation of solid sorbents that have low heat capacities as new materials for CO₂ separation. As a new class of porous solids, MOFs are considered promising materials to perform CO₂ separation. MOFs are primarily utilized as sorbents to achieve CO₂ separation. This is based on differences in interaction between gas species and frameworks and the consequent selective adsorption of target gas. In recent years, a growing number of explorations have also been done to fabricate membranes with MOFs for CO₂ separation, which rely on differences in either the diffusion rates of gas molecules within membranes or their molecular sizes.^{2, 3} There are generally four types of CO₂ capture that can be performed using MOFs, namely post-combustion capture, pre-combustion capture, oxy-fuel combustion, and direct capture from air.^{1, 4}

1.1 Post-Combustion Capture

The flue gas generated in current power plants is mainly comprised of N₂ (73-77%) and CO₂ (15-16%), with a total pressure of approximately 1 bar. After removal of

SO_x, the flue gas is anticipated to interact with the CO₂ scrubber at temperatures between 40 and 60 °C.¹ Thus, post-combustion capture is aimed at separating CO₂ from N₂ at 40 °C or higher, with CO₂ partial pressure equivalent to 0.15 bar.

In the context of sorbent-based separation, Mg-MOF-74 (CPO-27-Mg) is notable for its high CO₂ uptakes at low pressure under dry conditions (5.28 mmol/g at 40 °C and 0.15 bar), which stem primarily from the high density of open metal sites (OMSs).⁵⁻⁷ OMSs have strong interaction with CO₂ molecules, but are occupied more preferably by H₂O. This leads to a dramatically decreased CO₂ uptake of Mg-MOF-74 when it is exposed to moisture. Considering the measurable amount of water vapor (5-7%) in flue gas, it is definitely necessary to develop methods to endow MOFs with good selectivity towards CO₂ even in the presence of water.¹ Attempts were made by introducing alkylamines into MOFs to mimic the chemisorption of CO₂ in alkanolamine solutions.⁸⁻¹⁰ In 2012, Long and coworkers reported functionalizing Mg₂(dobpdc) (H₄dobpdc = 4,4'-dihydroxy-3,3'-biphenyldicarboxylic acid) with *N,N'*-dimethylethylenediamine (mmen) via the coordination bonds formed between the OMSs on Mg₂(dobpdc) and amine groups.⁹ For each mmen molecule, one end is coordinated to an unsaturated Mg center, while the other end is dangling in the channel to capture CO₂ in a chemisorptive manner (**Figure 1a**). The high density of amine groups in mmen-Mg₂(dobpdc) results in its good selectivity for CO₂ even in the presence of moisture.¹¹ Its CO₂ uptake is 3.5 mmol/g at 40 °C and 0.15 bar (**Table 1**). Particularly, a steep step is observed in the low-pressure region of the isotherm. Spectroscopic and diffraction measurements indicate that the unique step in the CO₂ isotherm of mmen-M₂(dobpdc) is attributed to a cooperative CO₂ insertion

mechanism that involves a phase-change in the capture process (**Figure 1b**). The CO₂ pressure at which the phase-change occurs can be tuned by altering temperature and the metal species in mmen-M₂(dobpdc). For a specific mmen-M₂(dobpdc), the phase-change pressure would increase from below 0.15 bar to above 1 bar when the temperature is elevated approximately 50 °C.¹¹ Thus, a large working capacity can be achieved with only a small temperature swing when the adsorption and desorption temperatures are tuned to make the phase-change pressures in the two isotherms right before the adsorption pressure and after the desorption pressure, respectively (**Figure 1c**). Moreover, the identity of the metal species in mmen-M₂(dobpdc) affects the bond strength of mmen-M, and influences the phase-change temperature accordingly. Compared with regular MOF sorbents, the adjustable phase-change step in the CO₂ isotherm of mmen-M₂(dobpdc) help it successfully circumvent the disadvantage of large temperature swings in regeneration (**Figure 1d**), making it extremely favorable for industrial CO₂ capture.

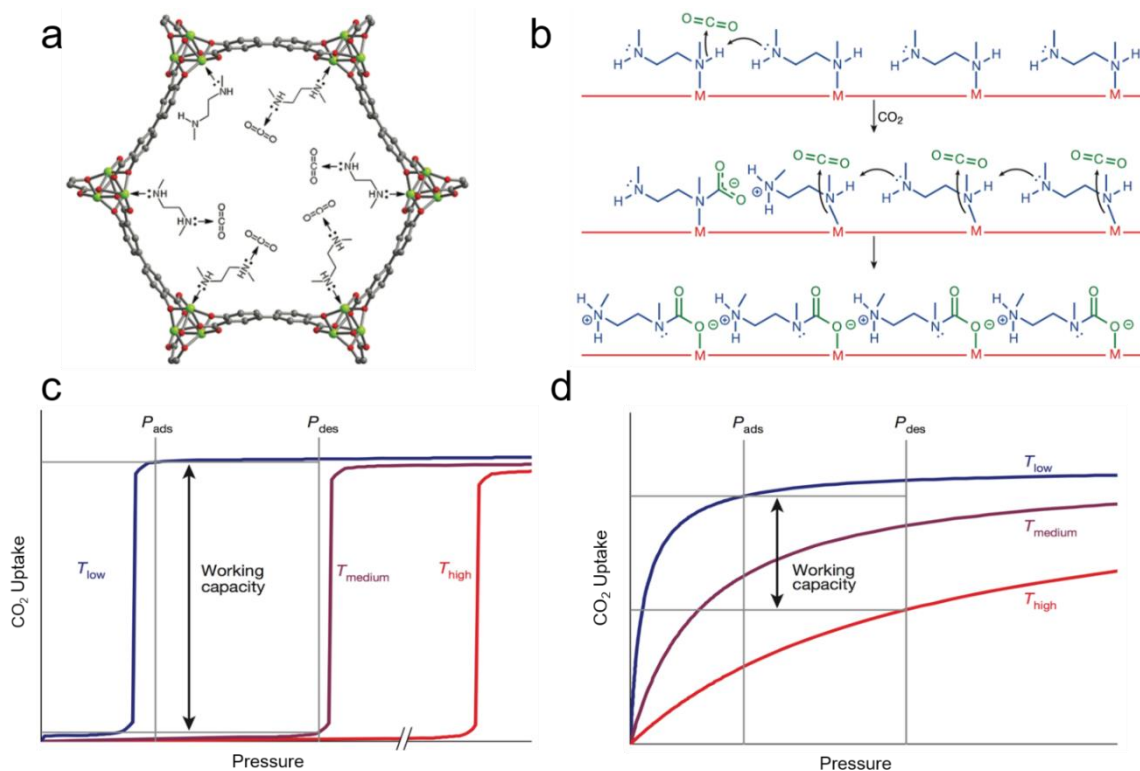


Figure 1. (a) Structure of the amine-appended CO₂ adsorbent mmen-Mg₂(dobpdc). Atom color legend: gray (carbon), red (oxygen), and green (magnesium). Reprinted with permission from ref. ⁹. Copyright 2012 American Chemical Society. (b) Illustration of the cooperative insertion mechanism for CO₂ adsorption in mmen-M₂(dobpdc). Comparison of the idealized CO₂ adsorption isotherms of (c) a phase-change adsorbent and (d) a classical microporous adsorbent at varied temperatures. Reprinted with permission from ref. ¹¹. Copyright 2015 Nature Publishing Group.

Works on integrating other alkylamines into Mg₂(dobpdc) were also reported subsequently, such as dmen-Mg₂(dobpdc) (dmen = *N,N*-dimethylethylenediamine),¹² en-Mg₂(dobpdc) (en = ethylenediamine),^{13, 14} men-Mg₂(dobpdc) (men = 1-methylethylenediamine),¹⁴ and den-Mg₂(dobpdc) (den = 1,1-dimethylethylenediamine).¹⁴ These works mainly focus tuning the alkylamine structure to optimize the performance of

the functionalized MOF in terms of its working capacity, regeneration energy, recyclability, water stability, etc.

Table 1. CO₂ uptakes of selected MOFs. Data are all extracted from CO₂ adsorption isotherms.

Compounds	CO ₂ uptake (mmol/g)	Temperature (°C)	Pressure (mbar)	Ref.
Mg-MOF-74	5.28	40	150	6
m _{men} -Mg ₂ (dobpdc)	3.5 ^a	40	150	11
	3.0 ^a	25	0.4	
d _{men} -Mg ₂ (dobpdc)	3.1 ^a	40	150	12
e _n -Mg ₂ (dobpdc)	3.53	40	150	14
	2.83	25	0.39	13
m _{en} -Mg ₂ (dobpdc)	3.6	40	150	14
d _{en} -Mg ₂ (dobpdc)	2.55 ^a	40	150	14
Cr-MIL-101-SO ₃ H-TAEA ^b	2.28	40	150	15
	1.12	20	0.4	
[Mg ₂ (dobdc)(N ₂ H ₄) _{1.8}]	4.9 ^a	40	150	16
	3.89	25	0.4	
SIFSIX-3-Zn	2.3 ^a	45	150	17
SIFSIX-3-Cu	2.4 ^a	45	150	18
	1.24	25	0.4	
SIFSIX-3-Ni	2.2 ^a	45	150	19

^a Estimations are made from CO₂ adsorption isotherms in cases where specific values were not reported.

^b TAEA = tris(2-aminoethyl)amine.

The foregoing examples all concern alkylamine modification on Mg₂(dobpdc), an expanded variant of Mg-MOF-74, because the sizes of the alkylamines molecules necessitate an isostructure with larger channels to avoid steric hindrance or intermolecular hydrogen bond formation. Nevertheless, amine functionalization was still achieved in Mg-MOF-74 in 2016 by introducing hydrazine into the framework.¹⁶ Consequently, an

ultrahigh density of free amine groups (6.01 mmol/g) was attained in the generated new material $[\text{Mg}_2(\text{dobdc})(\text{N}_2\text{H}_4)_{1.8}]$ (H_4dobdc = 2,5-dihydroxy-1,4-benzenedicarboxylic acid), which exhibited exceptionally high CO_2 uptakes in both single- or multi-component gas adsorption tests.

Apart from utilizing OMSs in MOFs to tether amines to enhance the affinity towards CO_2 , other functional moieties can also be employed to achieve this goal. For example, the Brønsted acidic sites in MOFs, like sulfonic acid groups, can be harnessed to fix amine through a Brønsted acid-base reaction for CO_2 capture.¹⁵ In addition, monodentate hydroxide is also demonstrated to have strong yet reversible interaction with CO_2 in $[\text{Co}^{\text{II}}\text{Co}^{\text{III}}(\text{OH})\text{Cl}_2(\text{bbta})]$ (MAF-X27_{ox}, H_2bbta = 1*H*, 5*H*-benzo(1,2-*d*:4,5-*d'*)bistriazole, MAF = metal azolate framework) via the formation and decomposition of bicarbonate.²⁰ It enables the framework to selectively adsorb CO_2 from gas mixtures even at humid conditions. Some MOFs, such as SIFSIX-3-M (M = Cu, Zn, Ni), are designed with suitable pore sizes and favorable arrays of inorganic anions to afford enhanced CO_2 binding affinity. Although the interaction between CO_2 and these MOFs are physical interaction, SIFSIX-3-M still exhibit good selectivity for CO_2 at humid conditions.¹⁷⁻¹⁹

Explorations were also made in the realm of MOF-derived membranes for post-combustion CO_2 capture.^{2, 21-24} Compared with synthesizing MOF adsorbents, developing MOF membranes to separate CO_2 from N_2 is more effortful. The main difficulty lies in the comparable kinetic diameters of CO_2 (3.30 Å) and N_2 (3.64 Å), making it challenging to prepare a MOF-derived membrane with simultaneous high permeance and high CO_2/N_2 selectivity.

1.2 Pre-Combustion Capture

Pre-combustion CO₂ capture requires the decarbonation of fuels into H₂ and concomitant CO₂. CO₂ is to be separated from high-pressure (5-40 bar) gas mixtures for the purpose of emitting zero CO₂ during the subsequent combustion step.¹ Because of the evident difference between the kinetic diameters of H₂ (2.89 Å) and CO₂ (3.30 Å),¹ separating the two species based on their distinctive sizes using molecular sieve membranes are mechanistically favorable. However, there exists a long-term bottleneck in the fabrication of membranes with evenly distributed and uniformly sized pores. In 2014, a breakthrough was made by Li, Yang and coworkers in this area.³ They successfully prepared molecular sieve nanosheets from a layered MOF, Zn₂(bIm)₄ (bIm⁻ = benzimidazolate), by wet ball milling and exfoliation. The apertures on a single-layered Zn₂(bIm)₄ sheet are estimated to be 2.9 Å, which would function as channels for small-sized H₂ but also as filters to impede the passage of CO₂. In addition to the good CO₂/H₂ selectivity, the straightforward apertures, constructed from four flat bIm molecules, would also facilitate the fast transport of passing H₂ for a high H₂ permeance. Besides these inherent structural advantages, the temperature under which Zn₂(bIm)₄ nanosheets were coated onto α -Al₂O₃ support was fine tuned to minimize ordered restacking between the nanosheets because lamellar restacking would block the pathway for H₂ and significantly affect its permeance. Accordingly, the generated membrane achieves a simultaneous high H₂ permeance (2700 GPU) and high H₂/CO₂ selectivity (291) in the binary CO₂/H₂ separation test.³ It successfully overcomes the Robeson's upper limit on previous MOF

membranes whose separation effects rely upon the different diffusion rates of the gas species.²⁵⁻²⁸

This MOF-derived membrane serves as a good example to illustrate the advantages of membrane-based separation, such as simple separating procedure, energy economy, and high productivity. By contrast, there is scarce experimental research concerning pre-combustion CO₂ capture with MOF sorbents.²⁹⁻³² In fact, high CO₂/H₂ selectivity can be achieved in most aforementioned MOFs in post-combustion CO₂ capture section. The restrictions in this area do not exist in CO₂/H₂ separation itself but in the expensive large-scale gasification of fuels. Nevertheless, this field still holds promise if the drawbacks can be overcome.

1.3 Oxy-Fuel Combustion

Oxy-fuel combustion is the ignition of fossil fuels in nearly pure O₂. The flue gas generated is thus almost completely CO₂ after removal of water, rendering a more facile capture step of CO₂.¹ In this approach, the separation target is transformed into producing pure O₂ from air. Traditional cryogenic distillation requires a large energy input, which fails to be a viable solution if CO₂ capture from oxy-fuel combustion is to be implemented extensively.¹ Therefore, alternative strategies that consume less energy but still separate O₂/N₂ effectively is highly desired.

Membrane-based separation would not be feasible in this scenario, because the kinetic diameters of O₂ (3.46 Å) and N₂ (3.64 Å) are extremely close.¹ In the context of sorbent-based separation, selective adsorption of one gas species with an absolute physisorption mechanism is also not applicable, due to the strong resemblance between

the physical properties of O₂ and N₂ (such as quadruple moment and polarizability). In contrast, separation that harnesses the different chemisorptive behaviors of the two gases would be very effective. Because O₂ has a high propensity to accept electrons from redox-active metal sites, while N₂ is not endowed with such a feature.¹ Much progress has been made in this aspect to date.

Cr²⁺ and Fe²⁺ are metal species with relatively strong reducing ability. With these unsaturated redox-active metal centers, Cr₃(btc)₂ (H₃btc = 1,3,5-benzenetricarboxylic acid), Cr-BTT (H₃BTT = 1,3,5-Tris(2*H*-tetrazol-5-yl)benzene), and Fe₂(dobdc) have the tendency to donate electrons to O₂ to have strong O₂ affinity and marked O₂/N₂ selectivity.³³⁻³⁵ However, Cr^{II}-MOFs display gradual decreases in O₂ capacity over multiple cycles,^{34, 35} while Fe₂(dobdc) would lose its recyclability at temperatures above 222 K for irreversible oxidation.³³ Subsequently, more explorations were made with Co^{II}-MOFs. Unlike previous Co-MOFs built with weak-field carboxylate-terminated ligands, strong-field N-donor ligands are adopted in the cases of Co-BTTri (H₃BTTri = 1,3,5-tri(1*H*-1,2,3-triazol-5-yl)benzene) and Co-BDTriP (H₃BDTriP = 5,5'-(5-(1*H*-pyrazol-4-yl)-1,3-phenylene)bis(1*H*-1,2,3-triazole)) (**Figure 2a** and **2b**).³⁶ The σ-orbitals in BTTri³⁻ and BDTriP³⁻ are higher in energy compared with those in weak-field ligands. Upon their formation of molecular orbitals with d orbitals from the metal, e_g* orbitals would be lifted to a higher energy level, making the electrons in these elevated e_g* orbitals have a stronger inclination to transfer to O₂ (**Figure 3**). This gives rise to stronger O₂ binding and a higher O₂/N₂ selectivity in MOFs. Consequently, Co-BTTri has an O₂ loading of 3.3 mmol/g at 0.21 bar and 195 K, with an IAST (IAST = ideal adsorbed solution theory) selectivity of

41 at the same temperature (**Figure 2c**).³⁶ Its isostructure, Co-BDTriP, in which one triazolate terminal is replaced by a more electron-donating pyrazolate group, exhibits greater O₂/N₂ selectivities at temperatures between 195 and 240 K (**Figure 2d**). This further demonstrates the fact that a coordination environment created by stronger-field ligands around the metal center would lead to a MOF with higher O₂ affinity. In addition to the enhanced selectivities, Co^{II}-MOFs also have good recyclability and water stability, which are also important criteria to consider for the design of future O₂ adsorbents.

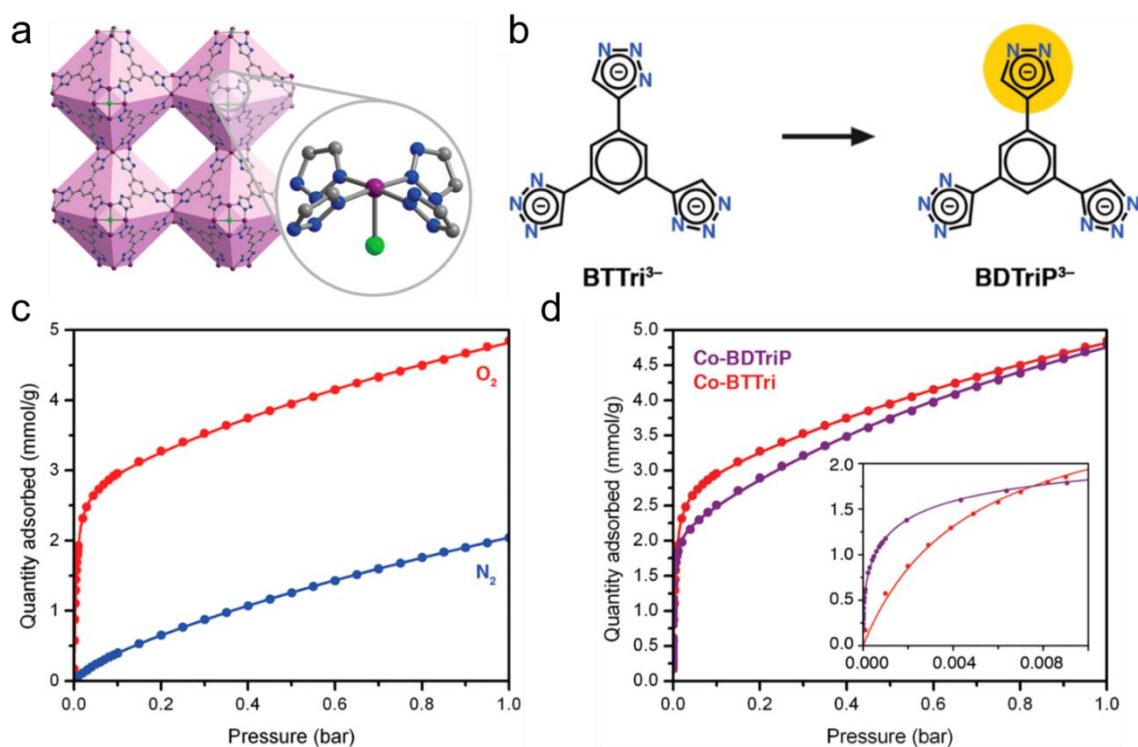


Figure 2. (a) Structure of Co-BTtri. Atom color legend: purple (cobalt), gray (carbon), blue (nitrogen), green (chlorine). (b) Molecular structures of BTtri³⁻ and BDtriP³⁻. (c) O₂ and N₂ adsorption isotherms of Co-BTtri at 195 K. Solid lines represent corresponding dual-site Langmuir fit curves. (d) O₂ adsorption isotherms of Co-BDtriP and Co-BTtri at 195 K. Solid lines represent corresponding Langmuir fits. Reprinted with permission from ref. ³⁶. Copyright 2016 American Chemical Society.

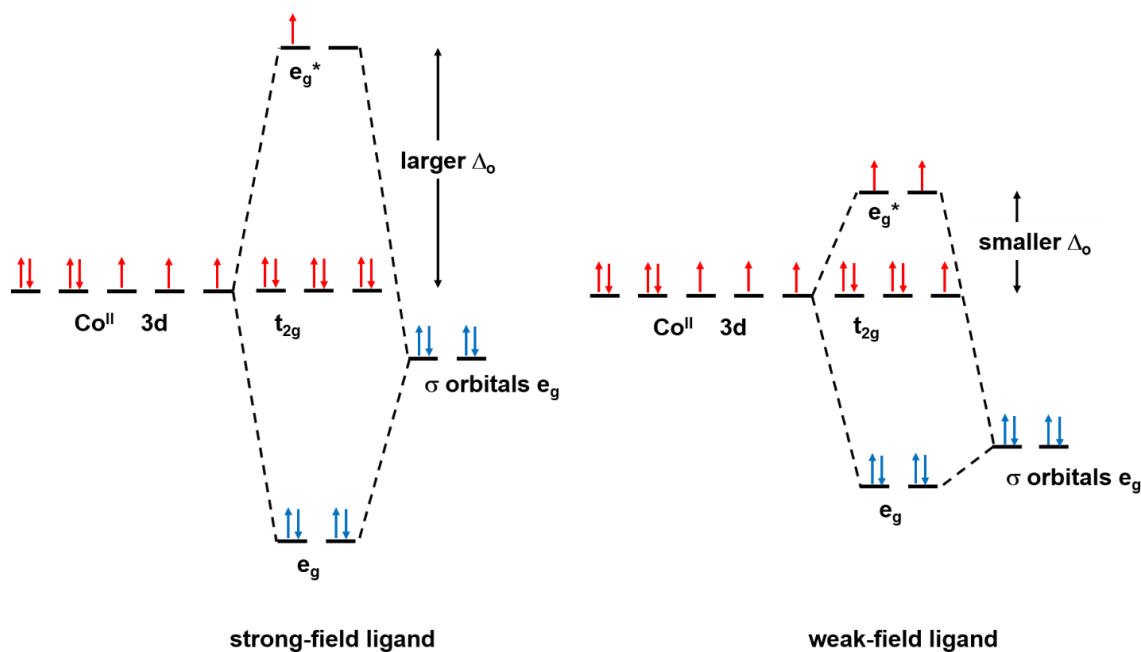


Figure 3. σ -bond molecular orbital diagrams of Co^{2+} coordinated by strong- (left) and weak-field ligands (right) in O_h symmetry. The coordination sphere around Co^{2+} is simplified into O_h symmetry for easy illustration. Only σ -orbitals of SALCs (SALC = symmetry-adapted linear combination) with e_g symmetry are shown for clarity.

1.4 Direct Capture from Air

All the previous scenarios are CO_2 capture from stationary point sources, which are intended for slowing the increasing rate of atmospheric CO_2 concentration. To scavenge the CO_2 that has already been released into air, a concept called “negative carbon technology” was recently put forward, which necessitates the practice of direct CO_2 capture from air (or direct air capture, DAC).⁴ In addition, DAC can also contribute to maintaining a low CO_2 level in submarines or spacecraft, so that a habitable environment can be guaranteed in these confined spaces.⁴ The current atmospheric CO_2 level is 400

ppm, so the operational condition of DAC is 0.4 mbar and 25 °C.^{1,4} The methods to endow MOFs with the ability to scavenge CO₂ at extremely low pressure are similar to those discussed in the section of post-combustion capture, including but not limited to, incorporating functional groups that have strong yet reversible chemical interaction with CO₂, and rational tailoring of the size and shape of internal pores for enhanced physical interaction between MOFs and CO₂. The DAC performance of some notable MOFs is summarized in **Table 1**. As for DAC with MOF-derived membranes, few examples have been reported. The similar molecular sizes of CO₂, N₂ and O₂, extremely low atmospheric CO₂ concentration, and complex air composition all make this research area very challenging.

Considering the current status in the development of MOF materials for CO₂ separation, this work focuses on the study of alkylamine-functionalized MOFs as adsorbents for post-combustion CO₂ capture and direct CO₂ capture from air.

CHAPTER II

**INCORPORATION OF ALKYLAMINE INTO METAL-ORGANIC
FRAMEWORKS THROUGH A BRØNSTED ACID-BASE REACTION FOR CO₂
CAPTURE ***

2.1 Introduction

Carbon capture has become increasingly important for the pressing need to curb the climbing concentration of atmospheric CO₂, which could cause severe global warming.^{37, 38} It is also necessary to capture CO₂ directly from air for some applications, such as the removal of CO₂ from in the controlled atmospheres of submarines or aircraft.^{39, 40} Tremendous effort has been devoted to studying various materials for CO₂ capture.^{1, 39, 41-48} Among them, metal-organic frameworks (MOFs) are considered to be promising candidates for their high porosity, tunable structure, and low heat capacity.^{5, 17, 18, 49-59} In most cases, CO₂ is physically adsorbed in MOFs.⁶⁰⁻⁶² This leads to very low selectivity of CO₂ over the coexisting species in flue gas or air with higher polarity, such as H₂O,^{1, 63, 64} which severely restrict the CO₂ adsorption capacity of MOFs under moist conditions. The problem can be solved by introducing alkylamines into the frameworks,^{9-13, 65} because the amine groups can specifically chemisorb CO₂ from gas mixtures through reversible chemical reactions.¹ So far, alkylamines are mainly incorporated via the coordination bonds between amine groups and open metal sites (OMSs).^{9, 11-13, 65, 66} Although some alkylamine integrated MOFs with high CO₂ uptakes have been obtained through this

*Reprinted with permission from ref. 15. H. Li, K. Wang, D. Feng, Y.-P. Chen, W. Verdegaal and H.-C. Zhou, Incorporation of Alkylamine into Metal–Organic Frameworks through a Brønsted Acid–Base Reaction for CO₂ Capture. *ChemSusChem*, 2016, **9**, 2832-2840. Copyright 2016 John Wiley and Sons.

approach, intrinsic disadvantages of this method still exist. Firstly, high density of OMSs in MOFs is greatly desired in this method to achieve a large alkylamine loading amount, which is a prerequisite for high CO₂ uptake in MOFs. However, the OMS density in a MOF is restricted by the inherent structures of inorganic nodes and organic linkers. They inevitably set an upper limit on alkylamine density and corresponding CO₂ uptake in MOFs. Secondly, the method is only applicable to MOFs with OMSs. Thus it excludes many MOFs,⁶⁷⁻⁶⁹ covalent organic frameworks (COFs),⁷⁰ and porous polymer networks (PPNs),^{42,71} which have no OMSs but retain high chemical stability to probably survive in alkylamine tethering conditions. Therefore, introducing other amine grafting sites can not only overcome the upper limit of alkylamine density in MOFs, but also expand the range of material candidates for amine incorporation, which is of great importance to advance the development of CO₂ adsorbents.

Herein, by employing the Brønsted acid sites in a highly stable MOF, Cr-MIL-101-SO₃H,^{72,73} we intend to integrate alkylamine into this mesoporous adsorbent through a Brønsted acid-base interaction. The experimental conditions in alkylamine tethering process are also to be investigated systematically to optimize the CO₂ capture performance of the resulting material.

2.2 Experimental Section

2.2.1 Chemicals

Chromium (III) nitrate nonahydrate (Cr(NO₃)₃·9H₂O), monosodium 2-sulfoterephthalic acid (H₂BDC-SO₃Na), hydrofluoric acid (HF), *N,N*-dimethylformamide (DMF), hydrochloric acid (HCl), methanol (MeOH), acetone, cyclohexane (CH),

tetrahydrofuran (THF), dichloromethane (DCM), ethylenediamine (en), diethylenetriamine (DETA), triethylenetetramine (TETA), tetraethylenepentamine (TEPA), pentaethylenehexamine (PEHA), tris(2-aminoethyl)amine (TAEA) were purchased from VWR. All commercial chemicals were used without further purification unless otherwise mentioned.

2.2.2 Instruments

Powder X-ray diffraction (PXRD) was carried out with a BRUKER D8-Focus Bragg-Brentano X-ray Powder Diffractometer equipped with a Cu sealed tube ($\lambda = 1.54178 \text{ \AA}$) at 40 kV and 40 mA. Fourier transform infrared (FTIR) data were recorded on a SHIMADZU IR Affinity-1 instrument with diffuse reflectance attachment or attenuated total reflectance (ATR) attachment. Images and analyses of SEM/EDS were taken by FEI Quanta 600 FE-SEM. The Quanta 600 FEG is a field emission scanning electron microscope capable of generating and collecting high-resolution and low-vacuum images. It is equipped with a motorized x-y-z-tilt-rotate stage, providing the following movements: x = y = 150 mm (motorized); z = 65 mm (motorized); Tilt +70 degrees to -5 degrees (motorized); Source: Field emission gun assembly with Schottky emitter source. Voltage: 200 V to 30 kV. Beam Current: >100 nA. Equipment associated with the Quanta 600 includes: conventional Everhart-Thornley detector, back-scattered electron detector, IR-CCD chamber camera, Oxford EDS system equipped with X-ray mapping and digital imaging, HKL/Oxford EBSD system incl. geological phase database for phase ID, Gatan panchromatic cathodoluminescence detector with RGB filters and a Zyvex S100

nanomanipulator. N₂ and CO₂ adsorption-desorption isotherms were measured using a Micromeritics ASAP 2020 system at different temperatures.

2.2.3 Preparation of Cr-MIL-101-SO₃H

A mixture of Cr(NO₃)₃·9H₂O (2.00g, 5 mmol), H₂BDC-SO₃Na (2.70g, 10 mmol), deionized water (30 mL), and HF (48-51 wt%, 0.3 mL) was heated at 463 K for 24 h.⁷² The as-synthesized solid was washed with deionized water, DMF and acetone in sequence and was dried in air. The obtained green crystalline powder of Cr-MIL-101-SO₃Na(H) was post-synthetically acidified in 60 mL dilute HCl solution (0.08 M, the solution was prepared with 2 mL concentrated HCl, 150 mL deionized water and 150 mL MeOH) at 80 °C for 12h to give Cr-MIL-101-SO₃H.⁷³ The resultant green solid was washed with deionized water and acetone in sequence and finally dried overnight at 130 °C under vacuum prior to further use.

2.2.4 Procedure of Tethering Alkylamine onto Cr-MIL-101-SO₃H

A sample of fully activated Cr-MIL-101-SO₃H (100 mg, 0.108 mmol) was added to a 40 mL vial full of anhydrous solution containing 2 equiv. alkylamine (en, DETA, TETA, TEPA, PEHA or TAEA).that has already been dispersed The aforementioned solvent was newly collected from solvent still (CH₂Cl₂, THF, DCM or MeOH). The suspension was stirred at RT for 5 min before alkylamine modified Cr-MIL-101-SO₃H was separated from the solution. The solid was washed with anhydrous cyclohexane for three times, desiccated under Schlenk line and activated at 80 °C under vacuum before gas uptake measurements were taken. Note: 1 equiv. refers to the amount of a certain alkylamine if we assume all the sulfonic acid groups in Cr-MIL-101-SO₃H are to be

consumed in a 1:1 ratio reaction with alkylamine. The volume of 1 equiv. respective alkylamine was calculated and listed in **Table 2**.

Table 2. Summary of the volume of 1 equiv. alkylamine for amine modification on Cr-MIL-101-SO₃H sample.

Alkylamine	Cr-MIL-101-SO ₃ H	1 equiv. alkylamine
en	100 mg, 0.108 mmol	21.7 uL
DETA	100 mg, 0.108 mmol	35.1 uL
TETA	100 mg, 0.108 mmol	48.4 uL
TEPA	100 mg, 0.108 mmol	61.9 uL
PEHA	100 mg, 0.108 mmol	79.5 uL
TAEA	100 mg, 0.108 mmol	48.6 uL

Note: 1 equiv. refers to the amount of a certain alkylamine if we assume all the sulfonic acid groups in Cr-MIL-101-SO₃H are to be consumed in a 1:1 ratio reaction with alkylamine.

2.2.5 Gas Sorption Measurement of Alkylamine-Modified Cr-MIL-101-SO₃H

The gas sorption isotherms were collected using Micromeritics ASAP 2020 system. N₂ uptakes were measured at 77 K, while CO₂ uptakes were measured at 0 °C, 20 °C, and 40 °C.

2.2.6 PXRD Measurement of Alkylamine-Modified Cr-MIL-101-SO₃H

The PXRD patterns of the samples were collected using a BRUKER D8-Focus Bragg-Brentano X-ray Powder Diffractometer equipped with a Cu sealed tube ($\lambda = 1.54178 \text{ \AA}$) at 40 kV and 40 mA. The ranges of 2θ are all set to be 2-35 degree.

2.2.7 EDX Measurement of Alkylamine-Modified Cr-MIL-101-SO₃H

The sample was first desiccated to remove moisture or solvent residues. After that, it is finely ground before being applied on a conductive tape which is adhered to the

EDX sample holder. After the sample was placed into the SEM/EDX instrument, the inner pressure was decreased below 10^{-6} torr before measurements were conducted. The scanning areas were all set to be approximately $40 \times 40 \mu\text{m}^2$ to detect the element species.

2.2.8 FTIR Measurement

The liquid TAEA was desiccated with molecular sieves at RT overnight to make anhydrous TAEA. The FTIR spectrum of the resulting TAEA was collected using a SHIMADZU IR Affinity-1 instrument with ATR attachment. The FTIR spectra of pristine Cr-MIL-101-SO₃H and Cr-MIL-101-SO₃H-TAEA were taken with the same equipment but with a diffuse reflectance attachment.

2.2.9 Calculation of Isotheric Heat of CO₂ Adsorption*

Isotheric heat of CO₂ adsorption of Cr-MIL-101-SO₃H-TAEA was calculated based on the CO₂ uptake data collected at 0 °C, 20 °C and 40 °C with a triple-site Langmuir model. The temperature dependence was introduced in the fit through the Arrhenius approach.

The used Langmuir model ($i = A, B, C$):

$$q = \sum_i q_{\text{sat},i} \frac{b_i p}{1 + b_i p} \quad \text{(Equation 1)}$$

$$b_i = b_{0,i} \exp\left(\frac{E_i}{RT}\right) \quad \text{(Equation 2)}$$

* The Langmuir fitting of the CO₂ adsorption isotherms and the calculation of isotheric heat of CO₂ adsorption were conducted by Mr. Wolfgang Verdegaal.

We used the analytical function of the triple-site Langmuir fit and solved the Clausius-Clapeyron Equation which describes the isosteric heat of adsorption as partial gradient of the logarithmic pressure over the temperature at constant adsorption loading.

$$Q_{st} = RT^2 \left(\frac{\partial \ln p}{\partial T} \right)_q \quad \text{(Equation 3)}$$

2.3 Results and Discussion

2.3.1 Synthesis and Characterization of Cr-MIL-101-SO₃H

Due to the strong electron donating ability of N atoms in amine groups, alkylamine is not only a good ligand to coordinate onto OMSs, but also a strong base, which is able to form ionic bonds with acid. Therefore, an alternative approach to introducing alkylamine into MOFs is to utilize Brønsted acid sites. Since Brønsted acid groups can be integrated into many types of organic linkers conveniently,^{74, 75} this approach can enormously increase the number of MOF candidates for CO₂ capture. In order to test our hypothesis, we chose Cr-MIL-101-SO₃H (**Figure 4-5**) as the pristine MOF to fix alkylamines for the following reasons. Firstly, Cr-MIL-101 is well-known for its ultrastability under harsh environments,⁷⁶ which allows the framework to remain intact in alkylamine solution during the postsynthetic treatment. Secondly, as a strong Brønsted acid, sulfonic acid group can easily form ionic bonds with amine groups, by which alkylamines can also be immobilized in MOFs. In Cr-MIL-101-SO₃H, the density of sulfonic acid groups is 3.25 mmol/g, which is even higher than that of OMSs (2.17 mmol/g).⁷² This provides Cr-MIL-101-SO₃H with the potential to tether more alkylamine molecules than Cr-MIL-101 and therefore have a higher CO₂ uptake. Thirdly, Cr-MIL-

101-SO₃H is mesoporous.⁷⁶ The favorable pore volume permits the incorporation of large alkylamines with more amine groups in a single molecule and thus promises a higher amine group density than that of MOFs only modified with small alkylamines. Last but not the least, the organic ligand to synthesize Cr-MIL-101-SO₃H, namely monosodium 2-sulfoterephthalic acid (BDC-SO₃Na(H)), is commercially available. This is extremely important because large-scale production of the absorbent is definitely necessary for industrial application. The synthesized Cr-MIL-101-SO₃H exhibit good porosity and crystallinity (**Figure 6-7**). Based on the EDX measurement, most Na ions in the framework were replaced by protons (**Figure 8-13** and **Table 3-6**), which ensures sufficient sulfonic acid groups in Cr-MIL-101-SO₃H to interact with alkylamine molecules.



Figure 4. Appearance of Cr-MIL-101-SO₃H.

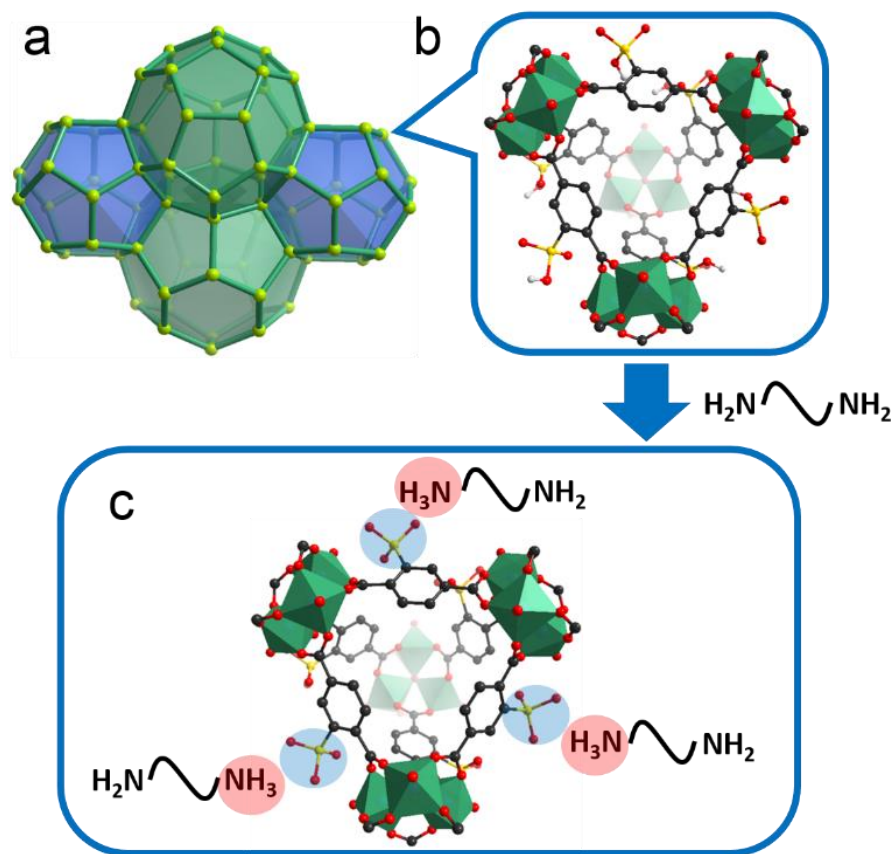


Figure 5. (a) Cr-MIL-101-SO₃H topology. Each line ball represents a supertetrahedron unit, and green and blue polyhedra represent two types of cages. (b) Supertetrahedron unit in Cr-MIL-101-SO₃H. Atom color legend: black (carbon), grey (hydrogen), yellow (sulfur), red (oxygen). (c) Illustration of electrostatic interaction between sulfonate groups and ammonium groups after proton transfer from sulfonic acid groups to amine groups.

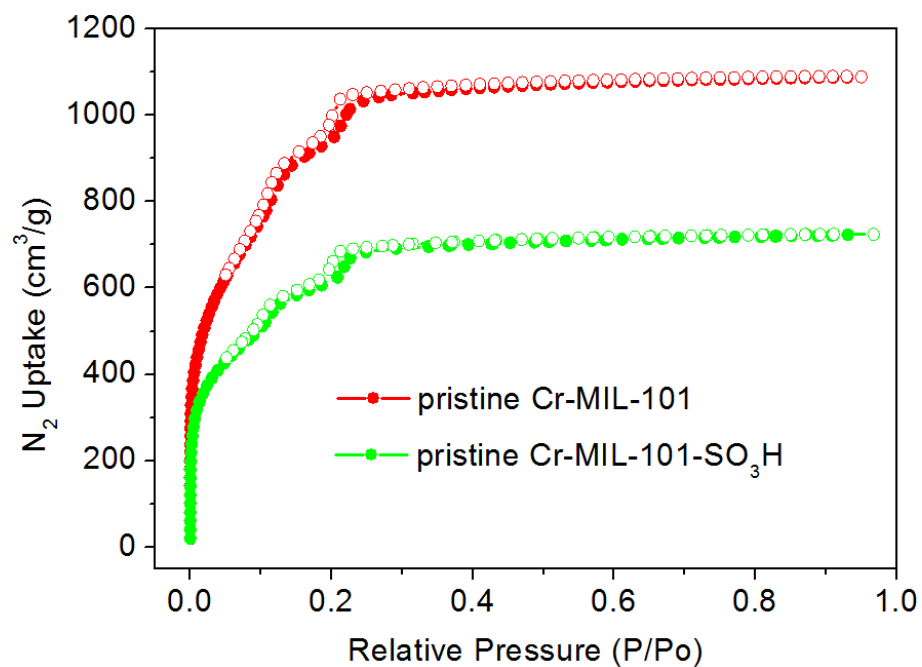


Figure 6. N₂ adsorption/desorption isotherms at 77K of Cr-MIL-101-SO₃H and Cr-MIL-101 samples.

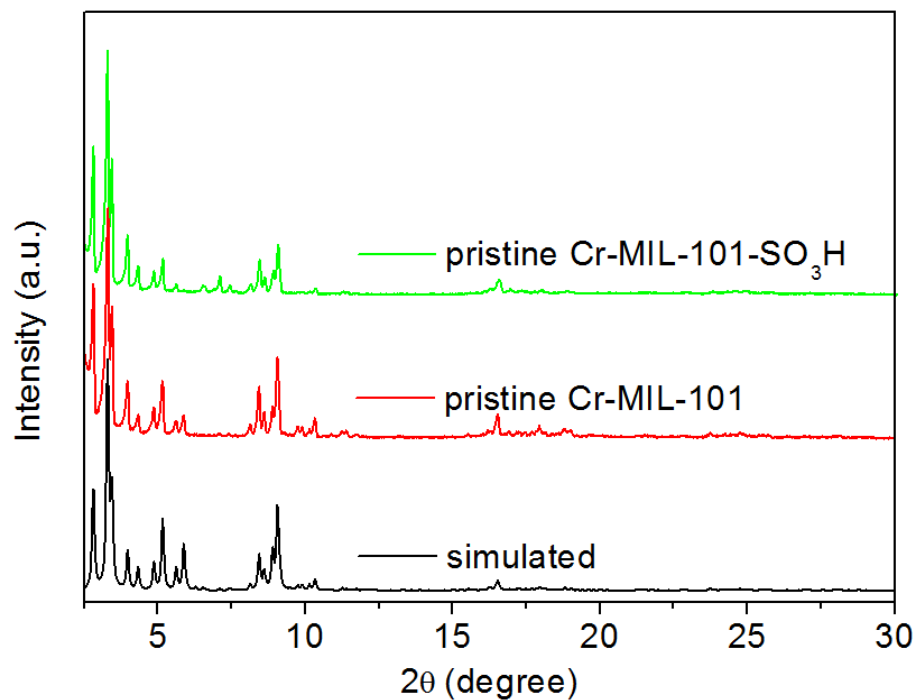


Figure 7. PXRD patterns of Cr-MIL-101-SO₃H and Cr-MIL-101 in comparison with the simulated pattern of Cr-MIL-101.

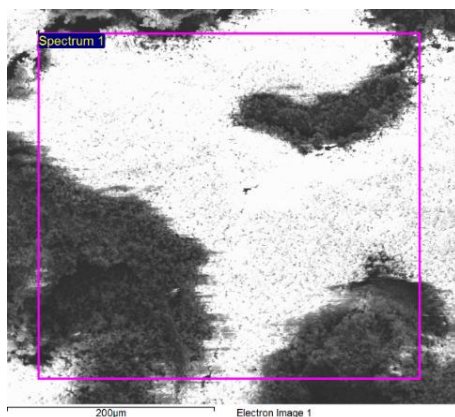


Figure 8. SEM image of Cr-MIL-101-SO₃H sample (1st measurement).

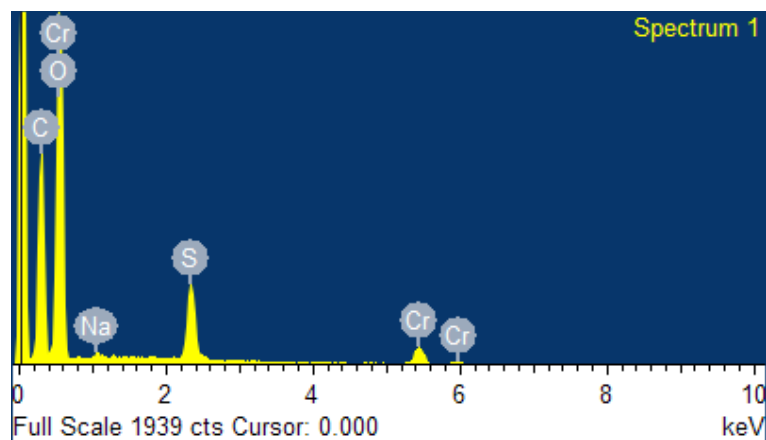


Figure 9. EDX spectrum of Cr-MIL-101-SO₃H sample (1st measurement).

Table 3. EDX data for Cr-MIL-101-SO₃H sample (1st measurement).

Element	Weight%	Atomic%
C K	29.57	44.63
O K	36.94	41.84
Na K	0.30	0.24
S K	7.98	4.51
Cr L	25.22	8.79
Totals	100.00	

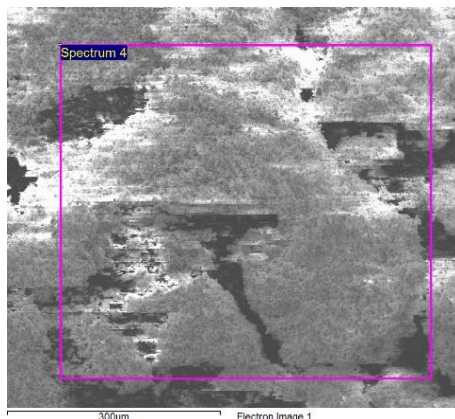


Figure 10. SEM image of Cr-MIL-101-SO₃H sample (2nd measurement).

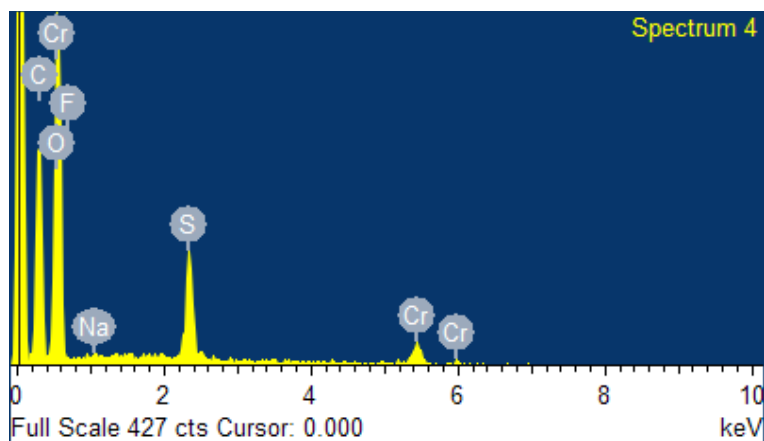


Figure 11. EDX spectrum of Cr-MIL-101-SO₃H sample (2nd measurement).

Table 4. EDX data for Cr-MIL-101-SO₃H sample (2nd measurement).

Element	Weight%	Atomic%
C K	28.18	45.80
O K	29.05	35.45
F K	0.73	0.76
Na K	0.23	0.19
S K	9.02	5.49
Cr L	32.79	12.31
Totals	100.00	

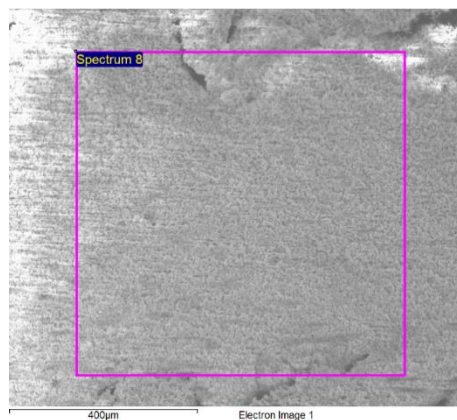


Figure 12. SEM image of Cr-MIL-101-SO₃H sample (3rd measurement).

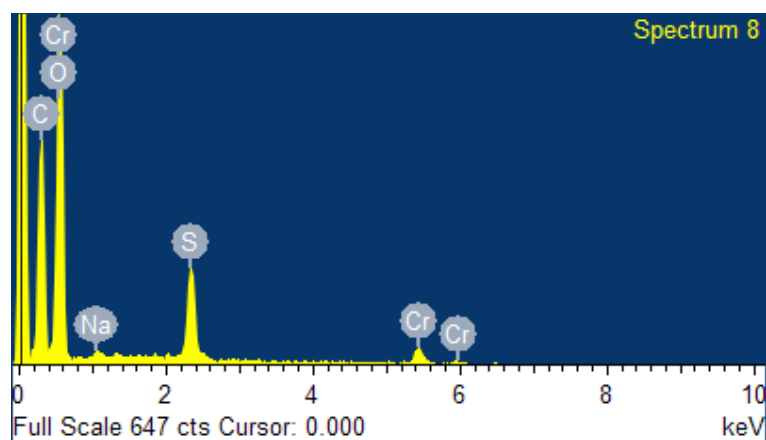


Figure 13. EDX spectrum of Cr-MIL-101-SO₃H sample (3rd measurement).

Table 5. EDX data for Cr-MIL-101-SO₃H sample (3rd measurement).

Element	Weight%	Atomic%
C K	33.15	47.72
O K	37.27	40.28
Na K	0.37	0.28
S K	9.77	5.26
Cr L	19.44	6.46
Totals	100.00	

Table 6. Summary of the EDX data of Cr-MIL-101-SO₃H sample.

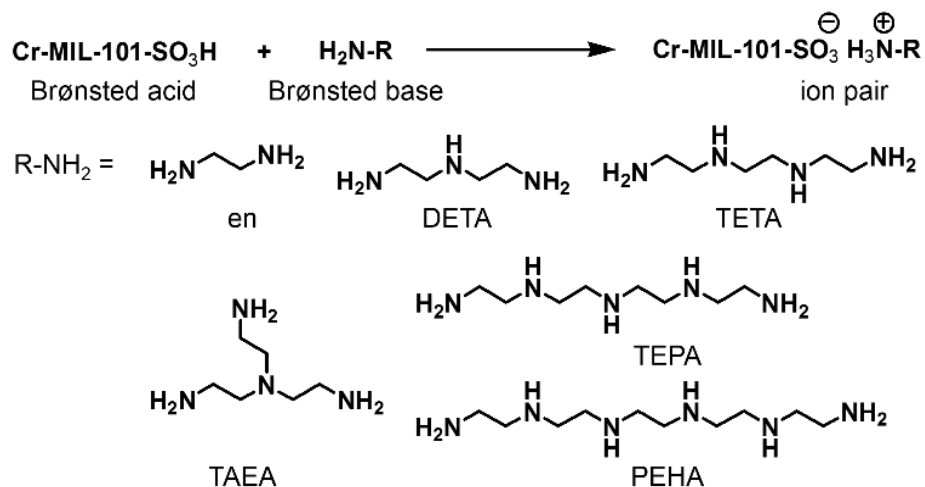
	Na Atomic%	S Atomic%
1st measurement	0.24	4.51
2nd measurement	0.19	5.49
3rd measurement	0.28	5.26
Average	0.24	5.09

$$\text{Na/S molar ratio} = \text{Na Atomic\%} / \text{S Atomic\%} = 0.24\% / 5.09\% = 0.047 < 5\%$$

In order to discover the most suitable alkylamine tethering condition for Cr-MIL-101-SO₃H to generate an adsorbent with the possibly highest CO₂ uptake for this material, we systematically studied experimental factors like alkylamine structure, alkylamine quantity, time and temperature, and solvent, and rationalized the results from a thermodynamic perspective.

2.3.2 Alkylamine Structure

As alkylamine structure significantly affects the CO₂ capture ability of the amine functionalized MOFs according to the reported examples,^{9, 12, 13, 66} the effect of alkylamine structure was first studied by conducting control experiments using a variety of alkylamines (**Scheme 1**). Fully activated Cr-MIL-101-SO₃H samples were added to five different CH solutions containing 2 equiv. ethylene diamine (en), diethylenetriamine (DETA), triethylenetetramine (TETA), tetraethylenepentamine (TEPA), pentaethylenehexamine (PEHA) respectively. These suspensions were stirred at RT for 5 min before the modified MOFs (Cr-MIL-101-SO₃H-X, X = en, DETA, TETA, TEPA or PEHA) were separated from the solutions. The MOFs were washed with anhydrous CH₂ and were activated before the measurements of their gas uptakes (**Figure 14**).



Scheme 1. Chemical structures of the alkylamine molecules and schematic illustration of their electrostatic interaction with Cr-MIL-101-SO₃H.

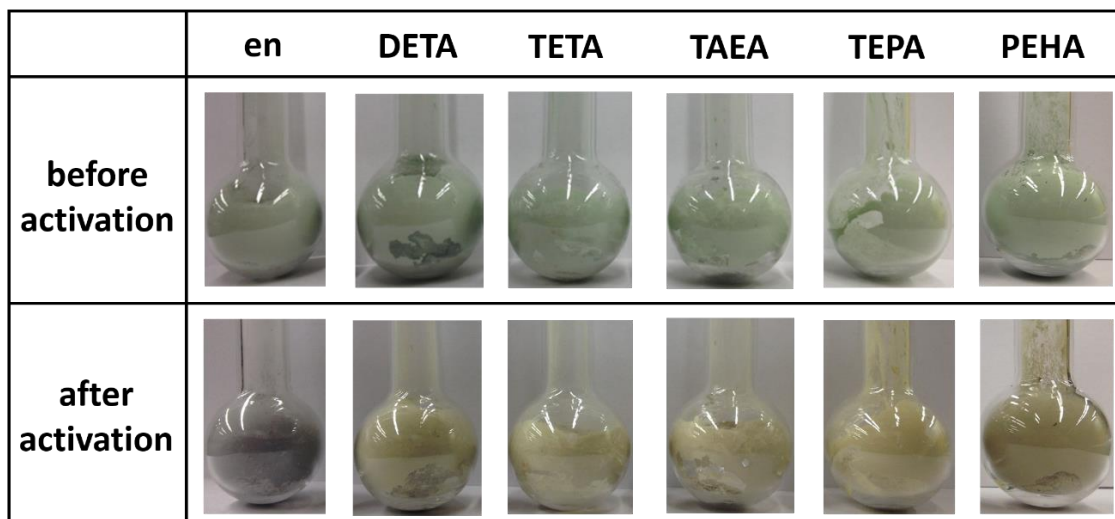


Figure 14. Colors of the different alkylamine modified Cr-MIL-101-SO₃H before and after activation at 80 °C under vacuum.

As the molecular weight of alkylamine increases from en to DETA, the CO₂ uptakes of alkylamine modified MOFs increase from 0.79 mmol/g to 2.23 mmol/g. (The

data are provided at 150 mbar and 20 °C; unless otherwise mentioned, all the other CO₂ adsorption data in the text are given at the same pressure and temperature.) When the amine weight further increases from TETA to PEHA, the CO₂ uptake decreases from 1.94 mmol/g to 1.33 mmol/g. The maximum value occurs when DETA or TETA is applied (**Figure 15**). We propose that the result is related to the amine group density in the modified MOFs, which is determined by the number of amine groups in a single alkylamine molecule and the amount of alkylamine actually incorporated into the MOF. Small molecules like en are able to reach most of the amine grafting sites in Cr-MIL-101-SO₃H, including those located in very small pores. This leads to a large alkylamine loading amount in Cr-MIL-101-SO₃H-en. However, the advantage is dramatically offset by the number of amine groups in each en molecule. Therefore, the overall amine group density in Cr-MIL-101-SO₃H-en is rather limited. For large alkylamines like TEPA, PEHA, though each alkylamine molecule can bring more amine groups into the framework, the bulky structures severely hamper their access to the amine grafting sites. This gives rise to a low overall amine group density in Cr-MIL-101-SO₃H-TEPA or Cr-MIL-101-SO₃H-PEHA as well. DETA and TETA might be two alkylamines that strike a balance between the number of amine group in a single molecule and the amount of alkylamine incorporated into the MOF. Thus, they excel among the other linear alkylamines among the series and produce the modified MOFs with the highest amine group densities and highest CO₂ uptakes. Our explanation is well supported by the N/S atomic ratio in elemental analysis (**Table 7**).

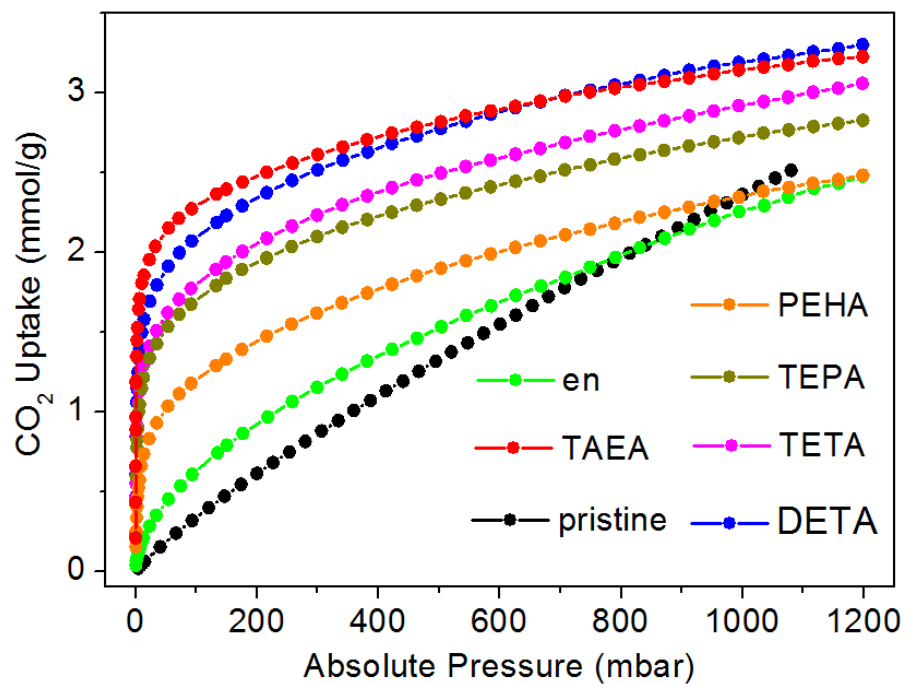


Figure 15. CO₂ uptakes at 20 °C of Cr-MIL-101-SO₃H modified with 2 equiv. different alkylamines. Control experiments were designed with alkylamine structure as the only variable. The other experimental factors are as follows: alkylamine quantity (2 equiv.), reaction time (5 min), reaction temperature (RT), and solvent (CH).

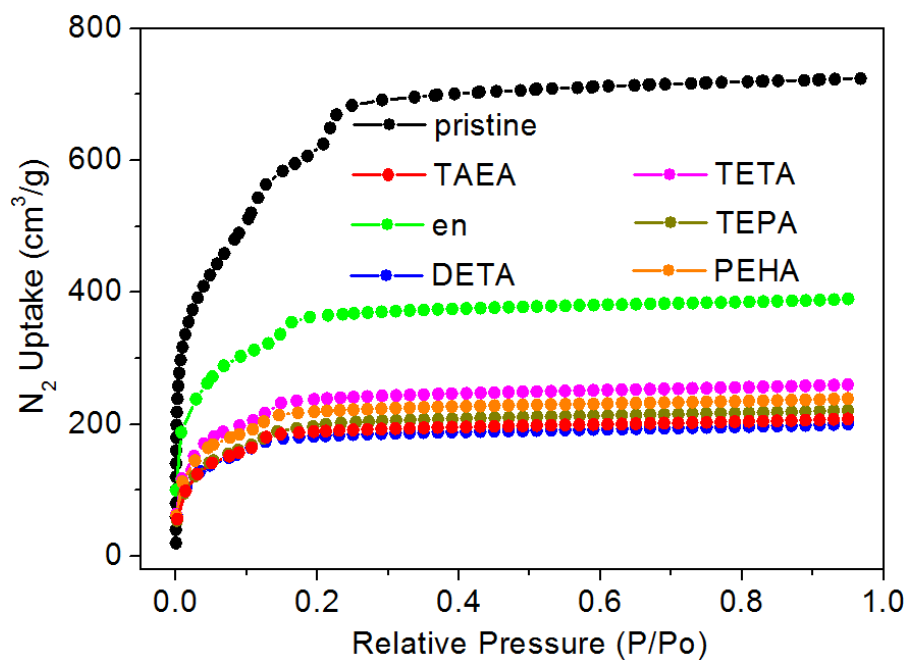


Figure 16. N₂ uptakes at 77 K of Cr-MIL-101-SO₃H modified with 2 equiv. different alkylamines. Control experiments were designed with alkylamine structure as the only variable. The other experimental factors are as follows: alkylamine quantity (2 equiv.), reaction time (5 min), reaction temperature (RT), and solvent (CH).

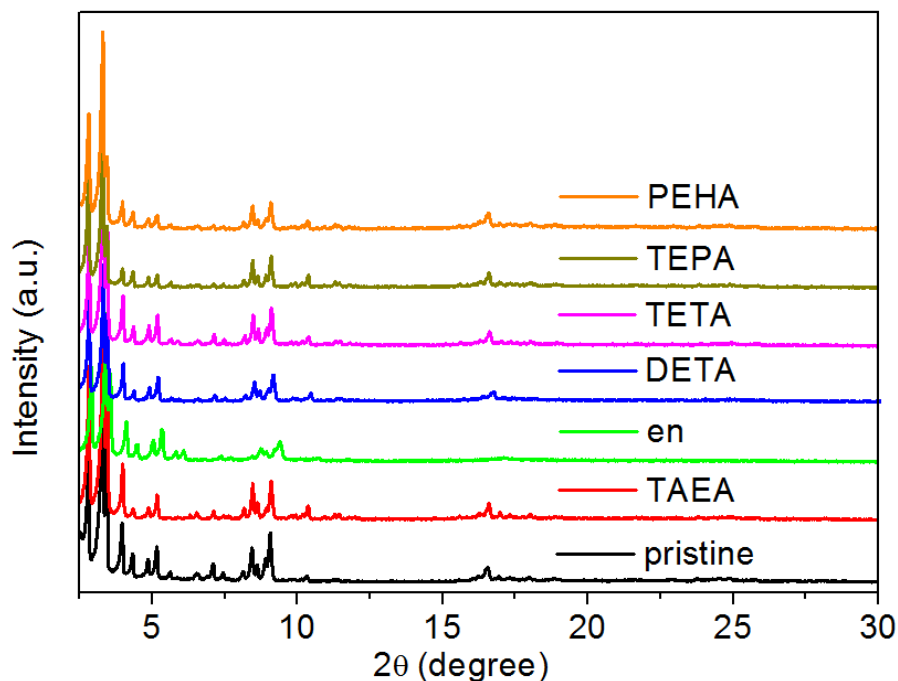


Figure 17. PXRD patterns of Cr-MIL-101-SO₃H modified with 2 equiv. different alkylamines. Control experiments were designed with alkylamine structure as the only variable. The other experimental factors are as follows: alkylamine quantity (2 equiv.), reaction time (5 min), reaction temperature (RT), and solvent (CH).

Table 7. Summary of the N/S atomic ratios of Cr-MIL-101-SO₃H modified with 2 equiv. different alkylamines. Control experiments were designed with alkylamine structure as the only variable. The other experimental factors are as follows: alkylamine quantity (2 equiv.), reaction time (5 min), reaction temperature (RT), and solvent (CH). The data were obtained by EA.

Alkylamine structure	N weight%	S weight%	N/S atomic ratio
en	10.03	6.92	3.31
DETA	13.26	5.85	5.18
TETA	11.34	5.07	5.11
TEPA	10.98	5.04	4.98
PEHA	10.80	5.57	4.43
TAEA	12.19	5.25	5.31

We also tethered TAEA, the isomer of TETA, into Cr-MIL-101-SO₃H, because these two alkylamines have the same number of amine groups in a molecule. Therefore, a high CO₂ uptake might also occur in the case of Cr-MIL-101-SO₃H-TAEA. It turns out that the CO₂ uptake of TAEA modified Cr-MIL-101-SO₃H (2.39 mmol/g) is even higher than those of TETA (1.94 mmol/g) and DETA (2.23 mmol/g) counterparts. We propose that the branched geometry of TAEA is more compact than the linear one of TETA, making TAEA migrate into Cr-MIL-101-SO₃H and have access to the amine grafting sites more easily than TETA. Thus the larger alkylamine loading amount in Cr-MIL-101-SO₃H-TAEA leads to its higher amine group density (**Table 7**), and higher CO₂ uptake. Based on the aforementioned, TAEA was chosen to be the optimal alkylamine to incorporate into Cr-MIL-101-SO₃H.

2.3.3 Alkylamine Quantity

Pore volume distribution indicates that the population of mesopores in Cr-MIL-101-SO₃H-TAEA significantly decreases (**Figure 16** and **Figure 18**), while its PXRD pattern suggests there is no collapse or phase change of the framework (**Figure 17**). These reveal the successful incorporation of TAEA into the cages of the framework. To probe the chemical nature of the captured TAEA in Cr-MIL-101-SO₃H-TAEA, IR spectra of Cr-MIL-101-SO₃H, Cr-MIL-101-SO₃H-TAEA and anhydrous TAEA were collected (**Figure 19-22**). Cr-MIL-101-SO₃H-TAEA exhibits two new peaks at 3234 and 3147 cm⁻¹, which are attributed to the stretching vibrations of -NH₃⁺.⁷⁷ This suggests the proton transfer from -SO₃H to -NH₂ and the formation of ammonium, which is consistent with our hypothesis.

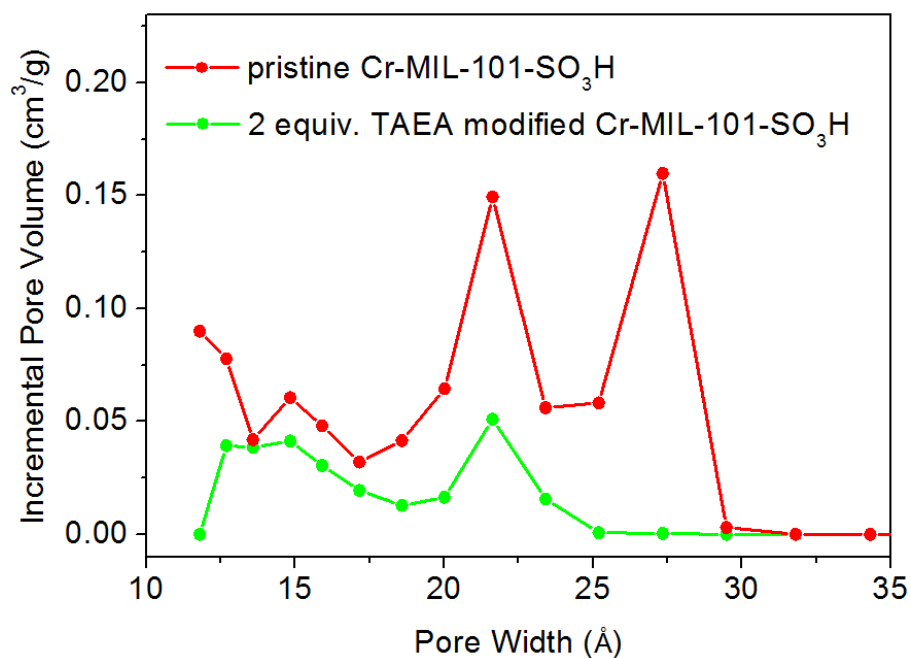


Figure 18. Incremental pore size distribution diagram of Cr-MIL-101-SO₃H-TAEA (modified by 2 equiv. TAEA) in comparison with that of the pristine Cr-MIL-101-SO₃H.

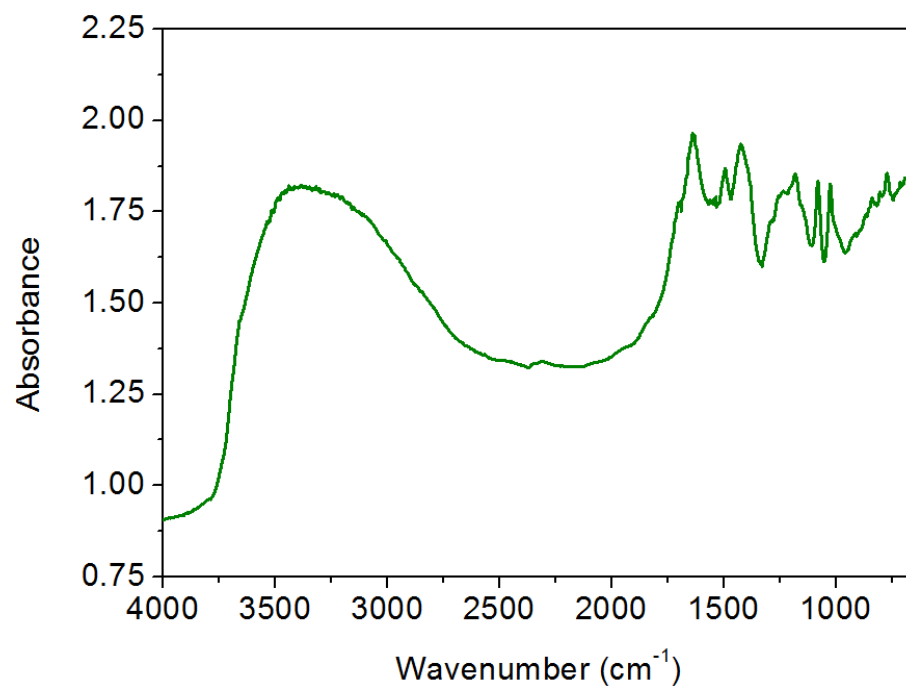


Figure 19. DRIFT spectrum of Cr-MIL-101-SO₃H under N₂ atmosphere.

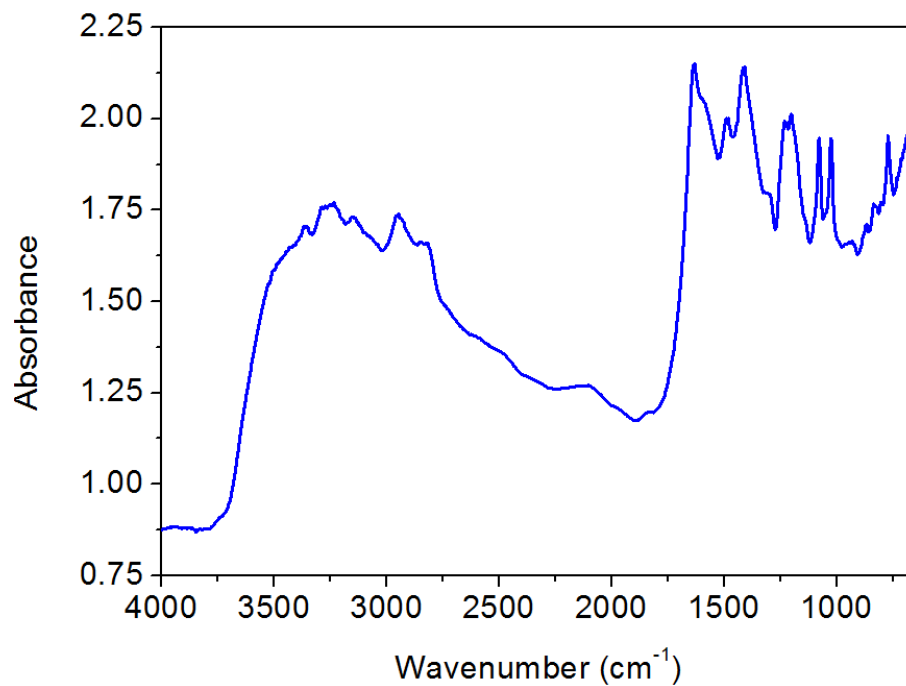


Figure 20. DRIFT spectrum of Cr-MIL-101-SO₃H-TAEA under N₂ atmosphere.

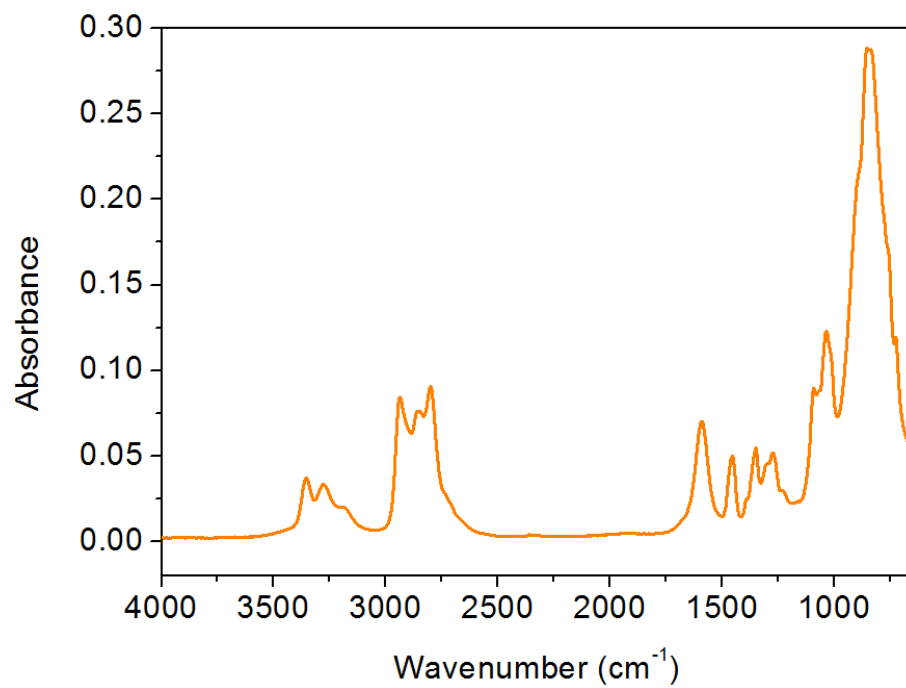


Figure 21. ATR-IR spectrum of anhydrous TAEA under N₂ atmosphere.

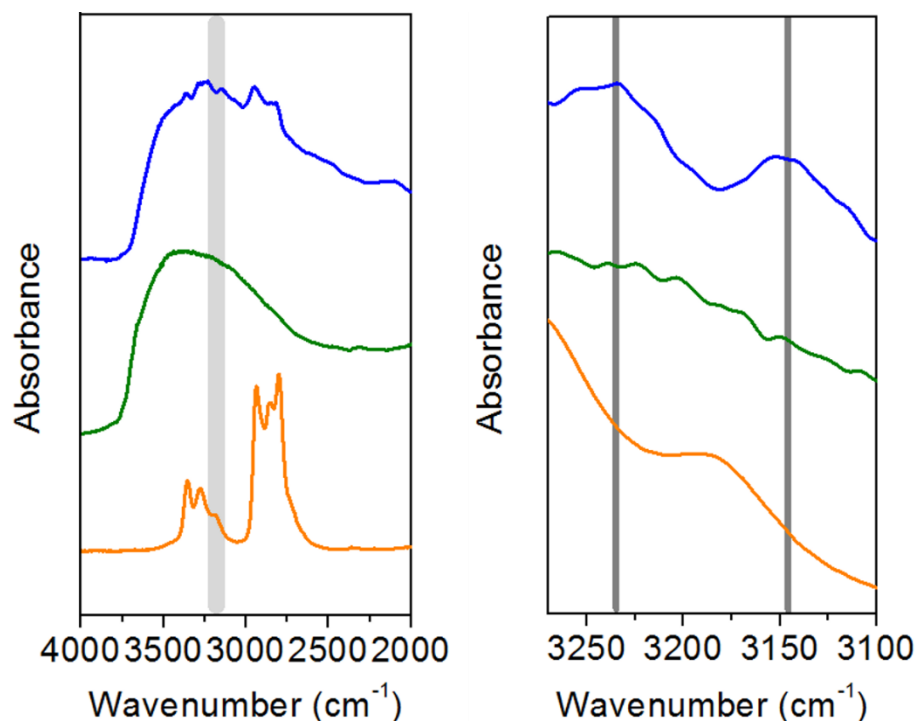


Figure 22. Comparison of the FTIR spectra of Cr-MIL-101-SO₃H-TAEA (blue), pristine Cr-MIL-101-SO₃H (green), and anhydrous TAEA (orange). (Left) The FTIR spectra in the range of stretching frequency with the region of interest highlighted in grey. (Right) The expanded region of interest with the characteristic peaks of Cr-MIL-101-SO₃H-TAEA marked at 3234 and 3147 cm⁻¹.

It is worth noting that the pore volume of the Cr-MIL-101-SO₃H modified by 2 equiv. TAEA is 0.32 cm³/g and the mesopores still exist (**Figure 16**). This indicates that there is still extra room in the framework to accommodate more alkylamine for a higher CO₂ uptake. Therefore, attempt was further made by enhancing the quantity of TAEA in solution to be 4, 8, 12 and 16 equiv., while maintaining the other experimental factors (**Figure 23-25**). The order of CO₂ uptakes in terms of TAEA quantity is 2 equiv. (2.40 mmol/g) < 4 equiv. (2.56 mmol/g) < 8 equiv. (2.75 mmol/g) > 12 equiv. (2.56 mmol/g) > 16 equiv. (2.00 mmol/g) (**Figure 23**). When TAEA quantity is small, the amount of the

TAEA loaded into the MOF is the main limiting factor for CO₂ uptake. More TAEA molecules can be incorporated into Cr-MIL-101-SO₃H as TAEA quantity increases, which leads to a higher CO₂ uptake. As TAEA quantity continues to increase, TAEA molecules occupy more space inside the framework (**Table 8**), eventually making some amine grafting sites inapproachable for CO₂. In this situation, accessible pore volume becomes the determining factor for the CO₂ uptake of the adsorbent. Thus, a modest quantity of TAEA is crucial to guarantee sufficient amine groups in MOF as well as easy access to amine groups for CO₂ molecules. With limited experiments, we chose 8 equiv. TAEA to be the optimal amount for alkylamine tethering in Cr-MIL-101-SO₃H.

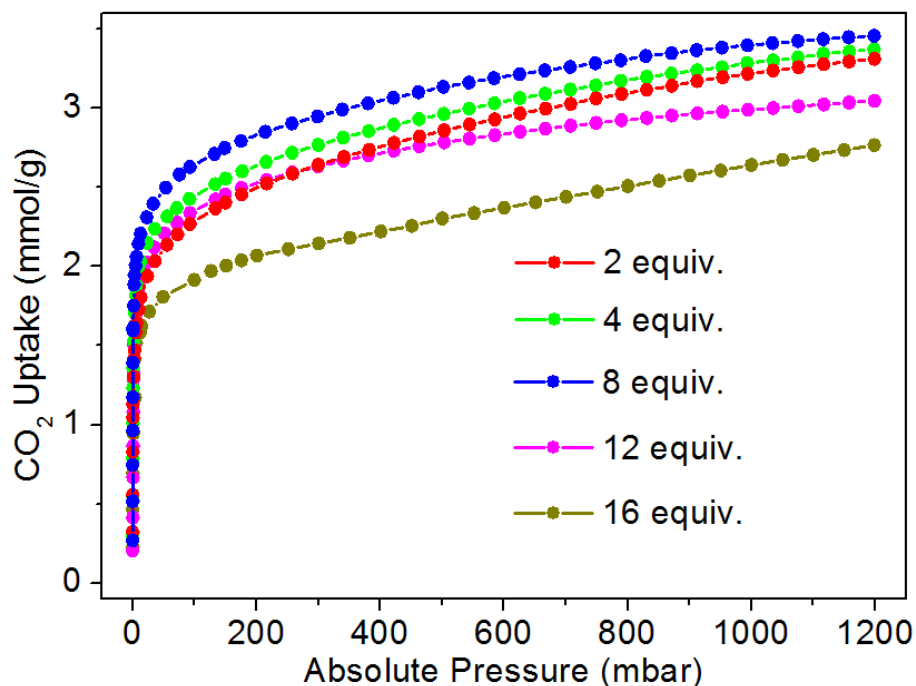


Figure 23. CO₂ uptakes at 20 °C of Cr-MIL-101-SO₃H modified by different amounts of TAEA. Control experiments were designed with alkylamine quantity as the only variable. The other experimental factors are as follows: alkylamine (TAEA), reaction time (5 min), reaction temperature (RT), and solvent (CH).

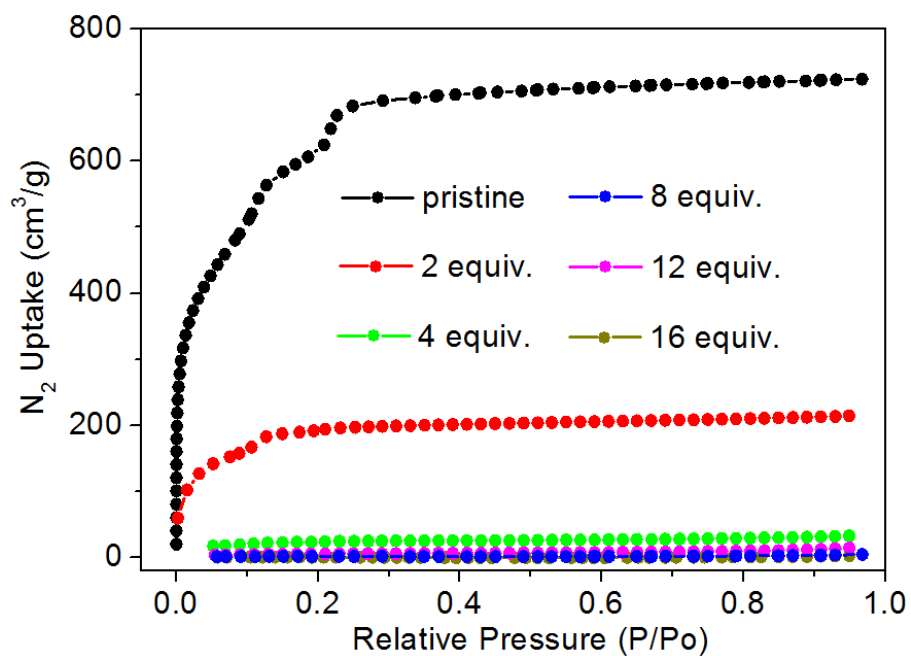


Figure 24. N₂ uptakes at 77 K of Cr-MIL-101-SO₃H modified by different amounts of TAEA. Control experiments were designed with alkylamine quantity as the only variable. The other experimental factors are as follows: alkylamine (TAEA), reaction time (5 min), reaction temperature (RT), and solvent (CH).

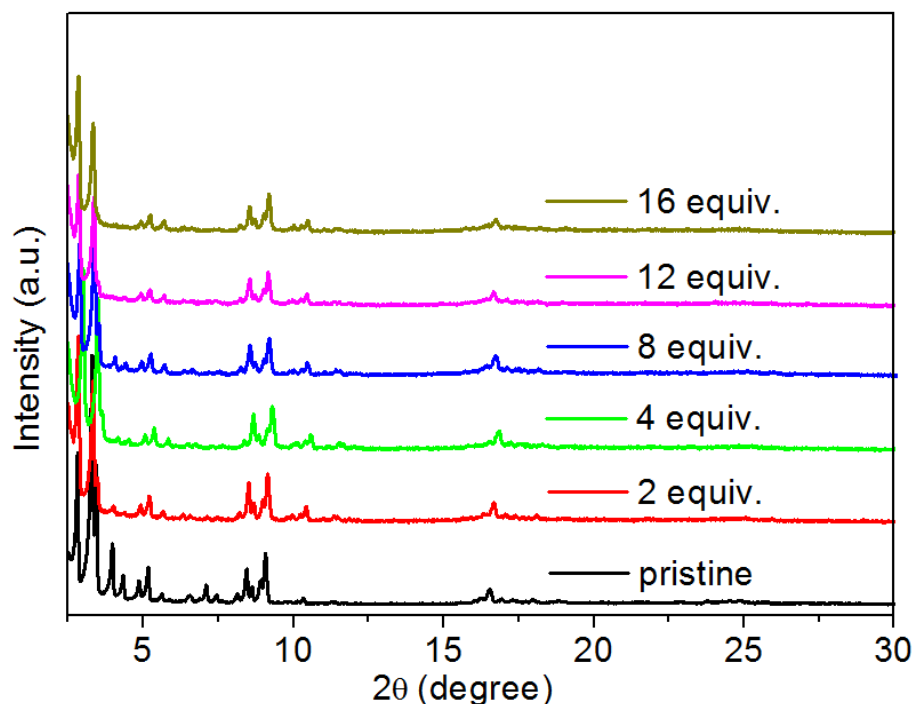


Figure 25. PXRD patterns of Cr-MIL-101-SO₃H modified by different amounts of TAEA. Control experiments were designed with alkylamine quantity as the only variable. The other experimental factors are as follows: alkylamine (TAEA), reaction time (5 min), reaction temperature (RT), and solvent (CH).

Table 8. Summary of the N/S ratios of Cr-MIL-101-SO₃H modified by different amounts of TAEA. Control experiments were designed with alkylamine quantity as the only variable. The other experimental factors are as follows: alkylamine (TAEA), reaction time (5 min), reaction temperature (RT), and solvent (CH). The data were obtained by EA.

Alkylamine quantity	N weight%	S weight%	N/S atomic ratio
2 equiv.	12.19	5.25	5.31
4 equiv.	13.42	4.78	6.42
8 equiv.	16.14	4.76	7.75
12 equiv.	16.20	4.65	7.96
16 equiv.	16.07	4.56	8.06

2.3.4 Time and Temperature

Although many examples of alkylamine functionalization in MOFs have been reported,^{9, 12, 13, 65, 66} a comprehensive optimization of amine tethering process is still missing, which inspires us to think whether the maximum CO₂ uptakes have been achieved for certain MOFs. More importantly, it is unclear why a certain amine tethering condition is adopted for a MOF and whether it is still applicable to other MOFs. To determine the optimal experimental condition of our system for the maximum CO₂ uptake, as well as to reveal the key factors that influence the integration of alkylamine into MOFs, we further studied other experimental factors, namely the reaction time and temperature in amine tethering.

Two series of control experiments were designed with amine tethering time and temperature as the only variable respectively. In the first set of experiments, reaction time was tuned to be 10 min, 30 min, 1h and 12 h, while reaction temperature was maintained at RT (**Figure 26-28**). Compared with Cr-MIL-101-SO₃H-TAEA prepared in 5 min at RT (2.75 mmol/g), the adsorbent prepared in 10 min at RT exhibits almost the same high CO₂ uptake (2.65 mmol/g) (**Figure 26**). However, Cr-MIL-101-SO₃H-TAEA samples prepared for a longer time show a decline in CO₂ uptakes, which might be attributed to the gradual decomposition of the frameworks, as supported by the obvious attenuated peaks in the PXRD pattern of the sample prepared in 12 h. In the second set, the reaction time was kept to be 5 min but reaction temperatures were increased to 40 °C and 60 °C (**Figure 29-31**). The result shows that the samples prepared at RT (2.75 mmol/g), 40 °C

(2.78 mmol/g) and 60 °C (2.69 mmol/g) in 5 min possess almost the same amount of CO₂ uptake (**Figure 29**).

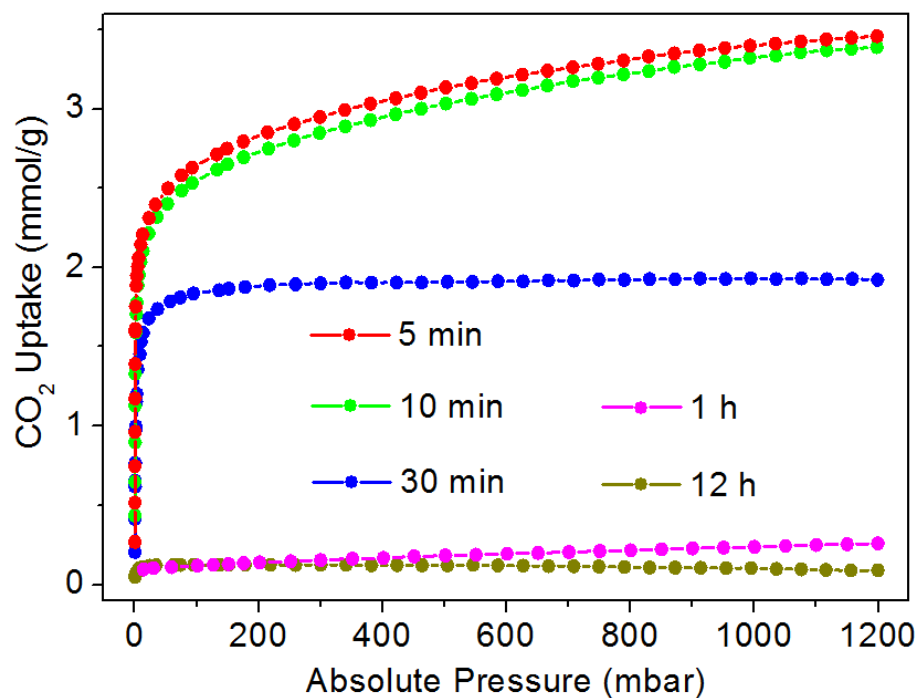


Figure 26. CO₂ uptakes at 20 °C of Cr-MIL-101-SO₃H-TAEA prepared in different periods of time. Control experiments were designed with reaction time as the only variable. The other experimental factors are as follows: alkylamine (TAEA), alkylamine quantity (8 equiv.), reaction temperature (RT), and solvent (CH).

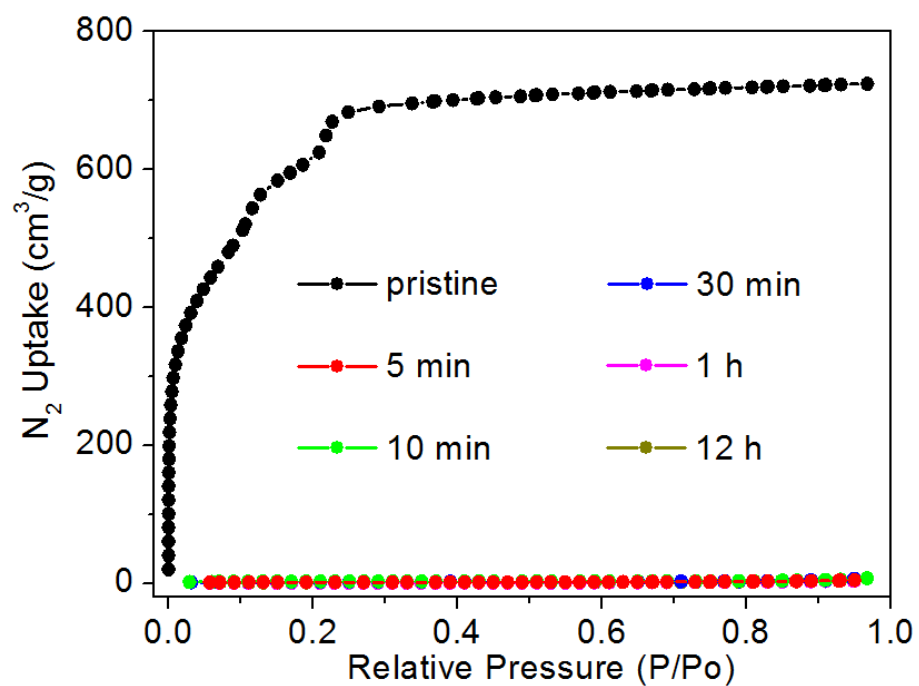


Figure 27. N₂ uptakes at 77 K of Cr-MIL-101-SO₃H-TAEA prepared in different periods of time. Control experiments were designed with reaction time as the only variable. The other experimental factors are as follows: alkylamine (TAEA), alkylamine quantity (8 equiv.), reaction temperature (RT), and solvent (CH).

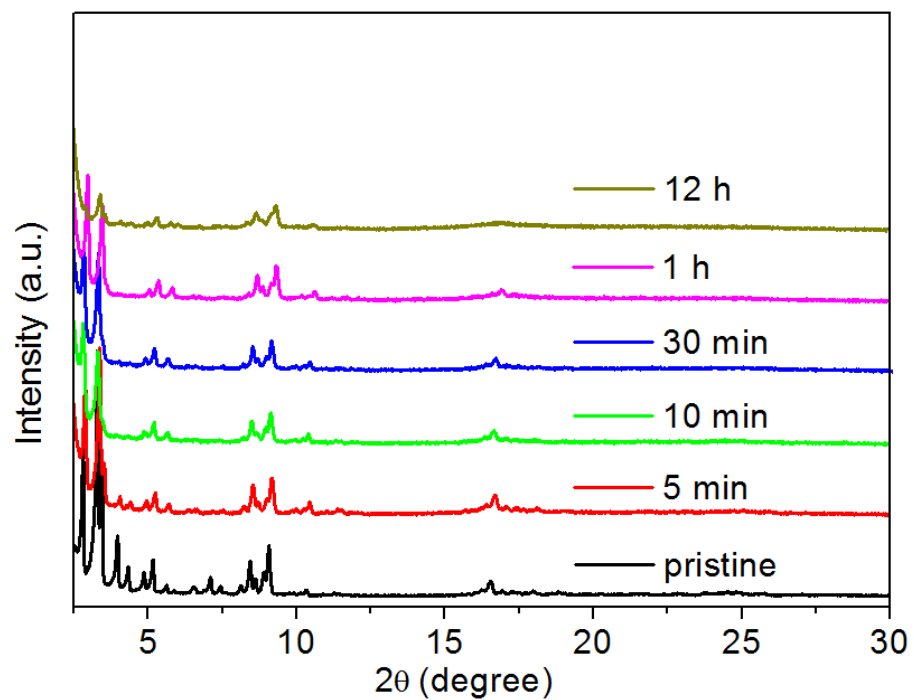


Figure 28. PXRD patterns of Cr-MIL-101-SO₃H-TAEA prepared in different periods of time. Control experiments were designed with reaction time as the only variable. The other experimental factors are as follows: alkylamine (TAEA), alkylamine quantity (8 equiv.), reaction temperature (RT), and solvent (CH).

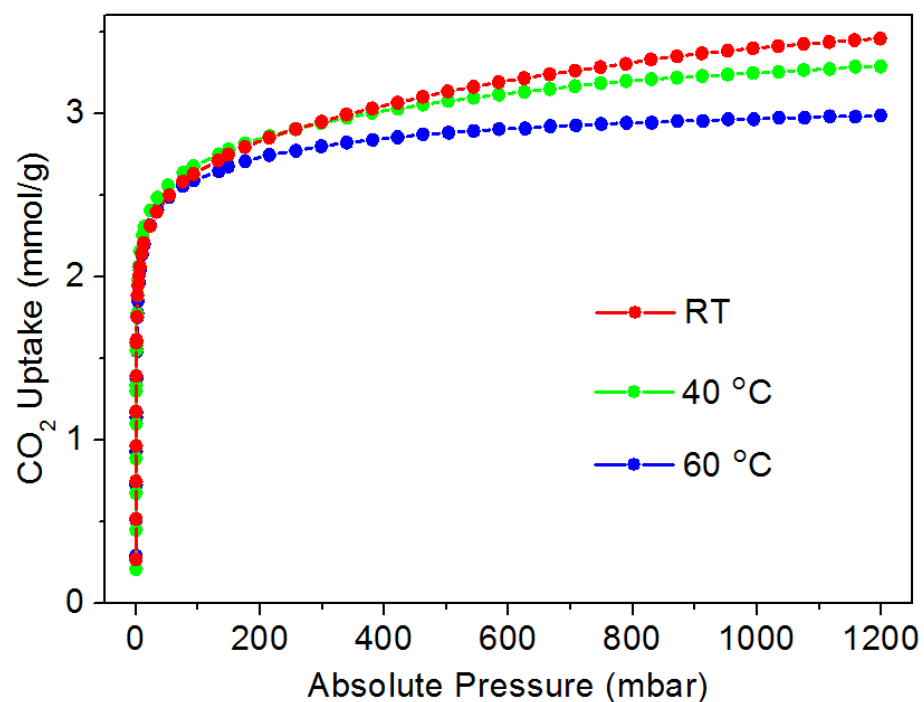


Figure 29. CO₂ uptakes at 20 °C of Cr-MIL-101-SO₃H-TAEA prepared under different temperatures. Control experiments were designed with reaction temperature as the only variable. The other experimental factors are as follows: alkylamine (TAEA), alkylamine quantity (8 equiv.), reaction time (5 min), and solvent (CH).

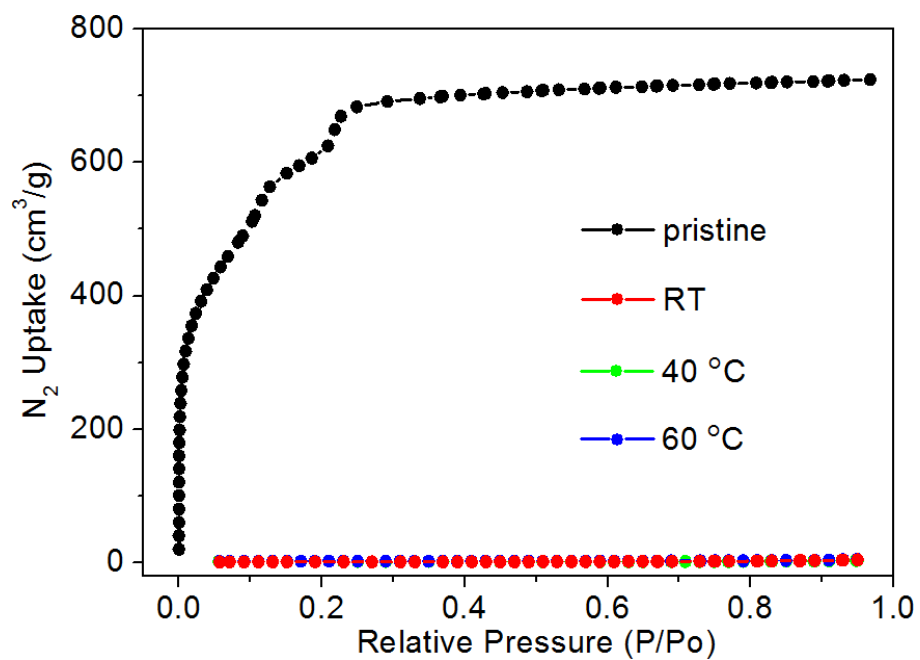


Figure 30. N₂ uptakes at 77 K of Cr-MIL-101-SO₃H-TAEA prepared under different temperatures. Control experiments were designed with reaction temperature as the only variable. The other experimental factors are as follows: alkylamine (TAEA), alkylamine quantity (8 equiv.), reaction time (5 min), and solvent (CH).

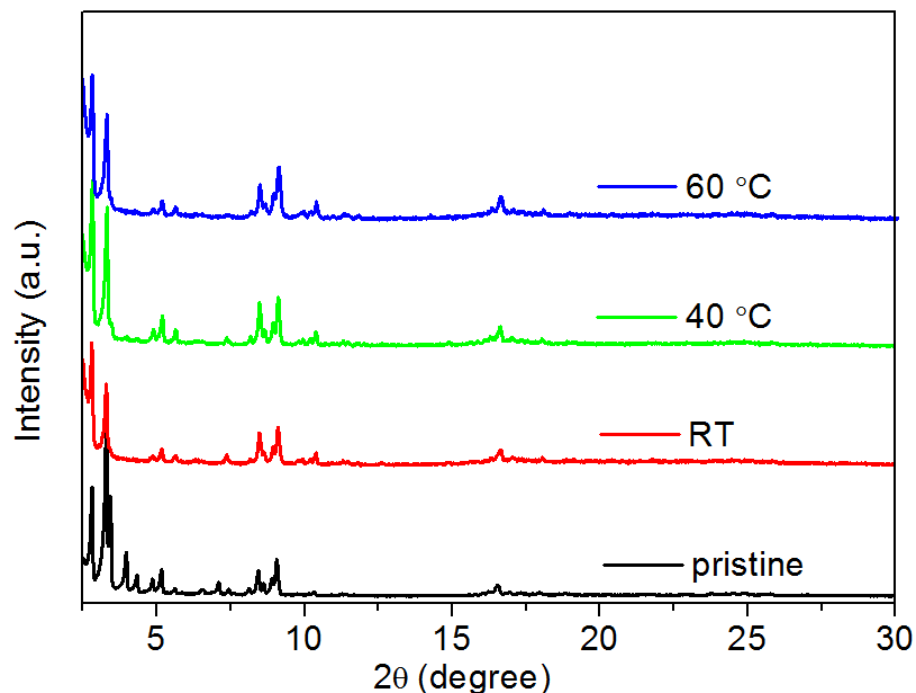


Figure 31. PXRD patterns of Cr-MIL-101-SO₃H-TAEA prepared under different temperatures. Control experiments were designed with reaction temperature as the only variable. The other experimental factors are as follows: alkylamine (TAEA), alkylamine quantity (8 equiv.), reaction time (5 min), and solvent (CH).

Both sets of the experiments reveal that the distribution equilibrium of TAEA between the solution phase and the MOF phase can be reached within 5 min at RT. It suggests that this amine tethering method is highly time and energy efficient compared with the precedent works in which the suspensions are usually refluxed or stirred under RT for more than 12 h.^{9, 12, 13, 65, 66} We propose there are two main reasons accounting for the easy achievement of the distribution equilibrium. Firstly, the ionic bond between –SO₃⁻ and –NH₃⁺ makes the alkylamine not so strongly localized to the amine grafting site as in a coordination bond, which facilitates the migration of alkylamine molecules deep into the framework. Secondly, Cr-MIL-101-SO₃H is a mesoporous structure with large

cage windows,⁷⁶ making it facile for large alkylamine molecules to diffuse and reach the amine tethering sites located in the interior part of the framework. Therefore, based on the control experiments, the optimal amine tethering time and temperature for this system are 5 min and RT.

2.3.5 Solvent

As mentioned, the incorporation of TAEA into Cr-MIL-101-SO₃H can be regarded as the distribution of TAEA between solution phase and the MOF phase. Therefore, the nature of the solvent might also affect the amine loading process, which has been significantly overlooked in the previously reported works. To seek an optimal solvent for our system, we selected three other anhydrous solvents besides CH, namely tetrahydrofuran (THF), dichloromethane (DCM), and methanol (MeOH), to perform control experiments (**Figure 32-34**). The CO₂ uptakes of products prepared in different solvents are CH (2.75 mmol/g) > THF (2.43 mmol/g) > DCM (2.29 mmol/g) > MeOH (1.38 mmol/g) (**Figure 32**), and those of amine loading amount and porosity are exactly the reverse (**Table 9** and **Figure 32**). It again shows that the CO₂ uptake of Cr-MIL-101-SO₃H-TAEA is higher when a larger amount of TAEA is loaded into the MOF when the accessible pore volume is sufficient. In view of the previous discussion, the distribution equilibrium of TAEA between the solution phase and the MOF phase is reached within 5 min at RT in CH. Therefore, to explain the above sequence, efforts should first be devoted to clarifying whether equilibria were also achieved in the cases of THF, DCM and MeOH.

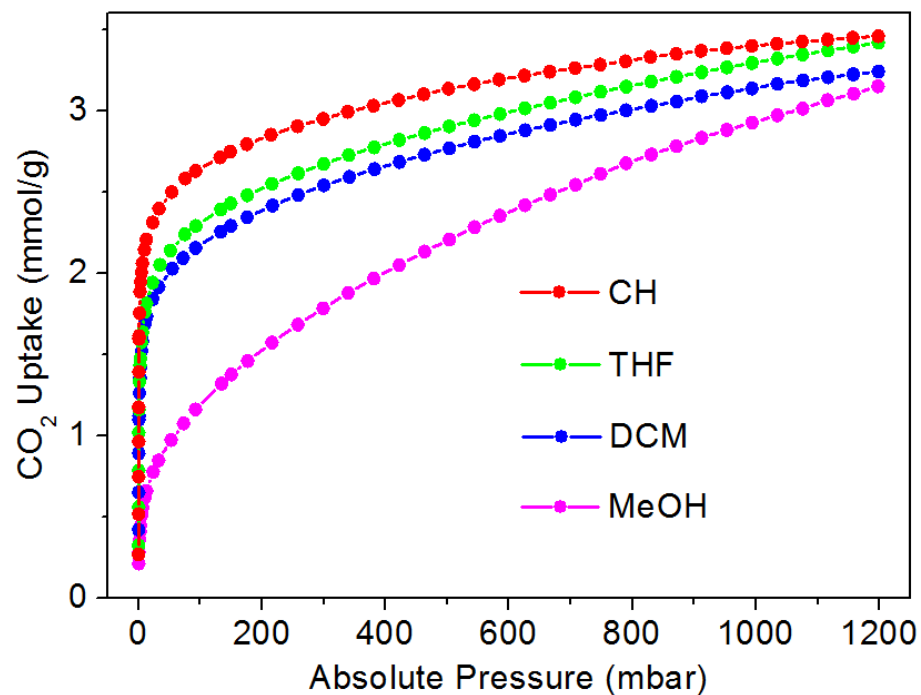


Figure 32. CO₂ uptakes at 20 °C of Cr-MIL-101-SO₃H-TAEA prepared in different solvents. Control experiments were designed with solvent as the only variable. The other experimental factors are as follows: alkylamine (TAEA), alkylamine quantity (8 equiv.), reaction time (5 min), and reaction temperature (RT).

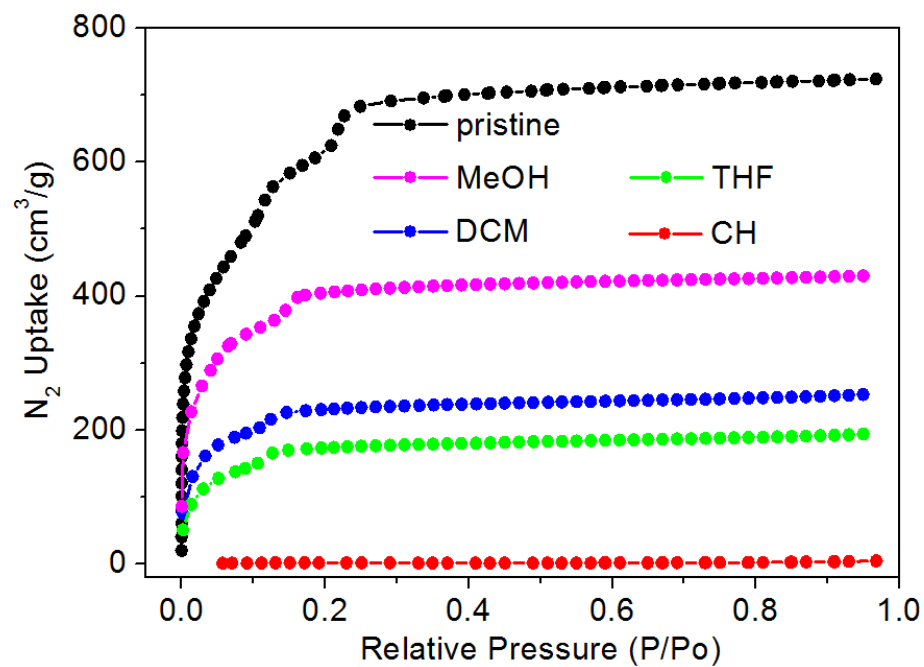


Figure 33. N₂ uptakes at 77 K of Cr-MIL-101-SO₃H-TAEA prepared in different solvents. Control experiments were designed with solvent as the only variable. The other experimental factors are as follows: alkylamine (TAEA), alkylamine quantity (8 equiv.), reaction time (5 min), and reaction temperature (RT).

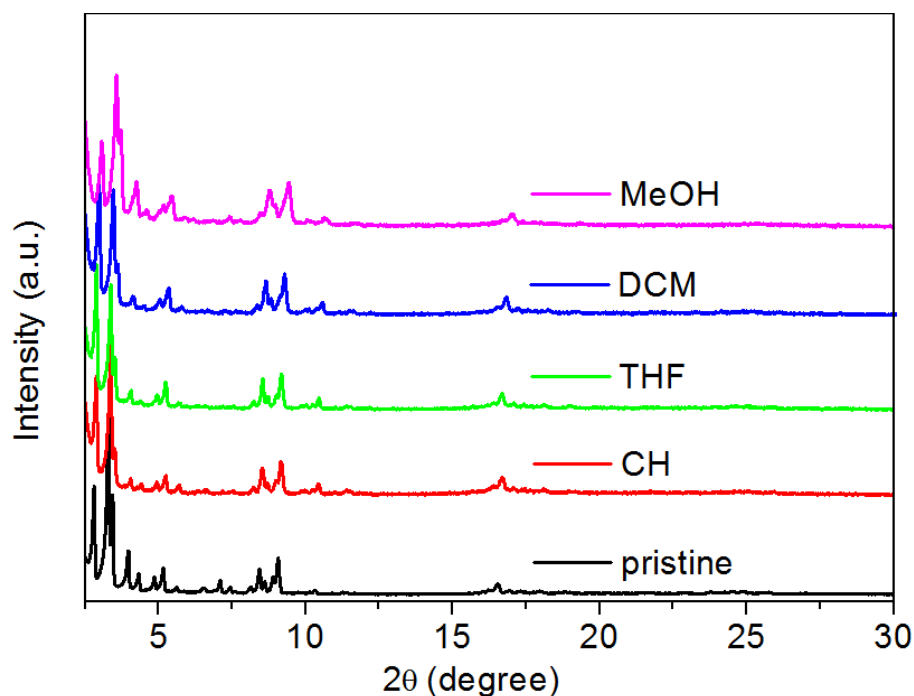


Figure 34. PXRD patterns of Cr-MIL-101-SO₃H-TAEA prepared in different solvents. Control experiments were designed with solvent as the only variable. The other experimental factors are as follows: alkylamine (TAEA), alkylamine quantity (8 equiv.), reaction time (5 min), and reaction temperature (RT).

Table 9. Summary of the N/S ratios of Cr-MIL-101-SO₃H-TAEA prepared in different solvents. Control experiments were designed with solvent as the only variable. The other experimental factors are as follows: alkylamine (TAEA), alkylamine quantity (8 equiv.), reaction time (5 min), and reaction temperature (RT).

Solvent	N weight%	S weight%	N/S atomic ratio
CH	16.14	4.76	7.75
THF	10.97	4.83	5.19
DCM	10.54	5.14	4.69
MeOH	9.79	6.26	3.57

The case of MeOH is selectively examined because it generates a sample with the lowest CO₂ uptake, which is as representative as the sample with the highest CO₂ uptake prepared in CH. Thus we conducted control experiments in MeOH with amine tethering

time extended to 2h and 12h (**Figure 35-37**). Compared with Cr-MIL-101-SO₃H-TAEA prepared in 5 min in MeOH, neither the CO₂ uptakes nor the porosities of these two samples have apparent differences (**Figure 35-36**), which indicates the distribution equilibrium of TAEA between the solution phase and the MOF phase is reached within 5 min in MeOH at RT. Therefore, it is reasonable to assume the equilibria are also achieved in the cases of THF and DCM, and the aforementioned sequence should be explained from a thermodynamic perspective.

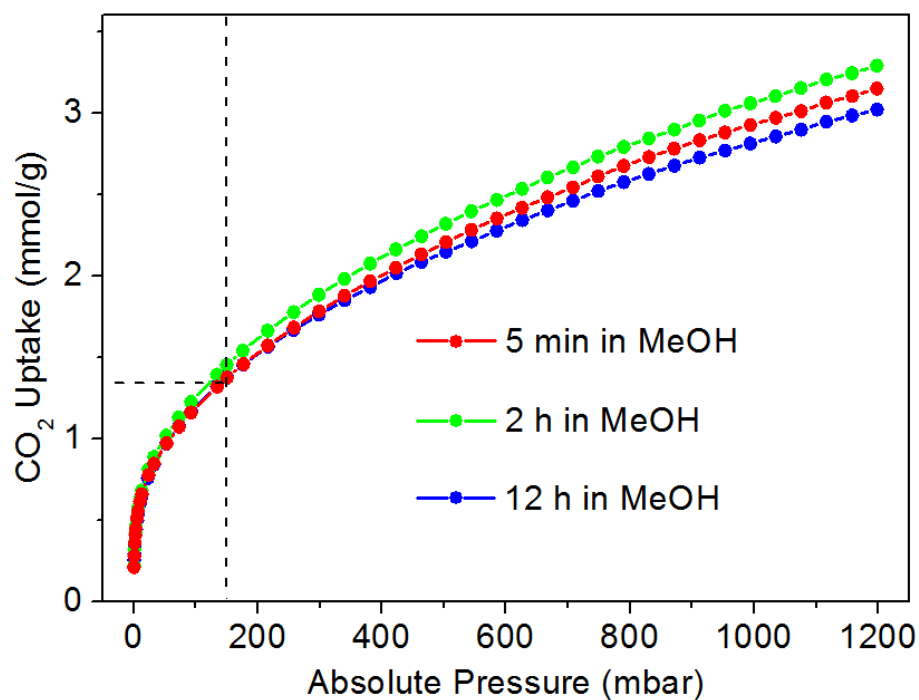


Figure 35. The CO₂ uptakes at 20 °C of Cr-MIL-101-SO₃H-TAEA prepared in different periods of time in MeOH. Control experiments were designed with reaction time as the only variable. The other experimental factors are as follows: alkylamine (TAEA), alkylamine quantity (8 equiv.), reaction temperature (RT), and solvent (MeOH).

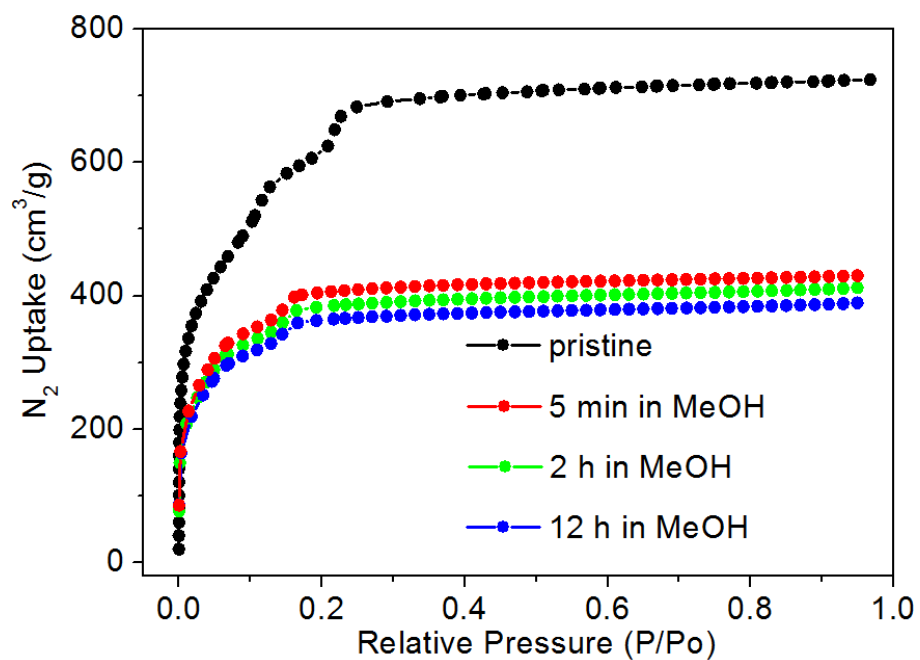


Figure 36. The N_2 uptakes at 77 K of Cr-MIL-101-SO₃H-TAEA prepared in different periods of time in MeOH. Control experiments were designed with reaction time as the only variable. The other experimental factors are as follows: alkylamine (TAEA), alkylamine quantity (8 equiv.), reaction temperature (RT), and solvent (MeOH).

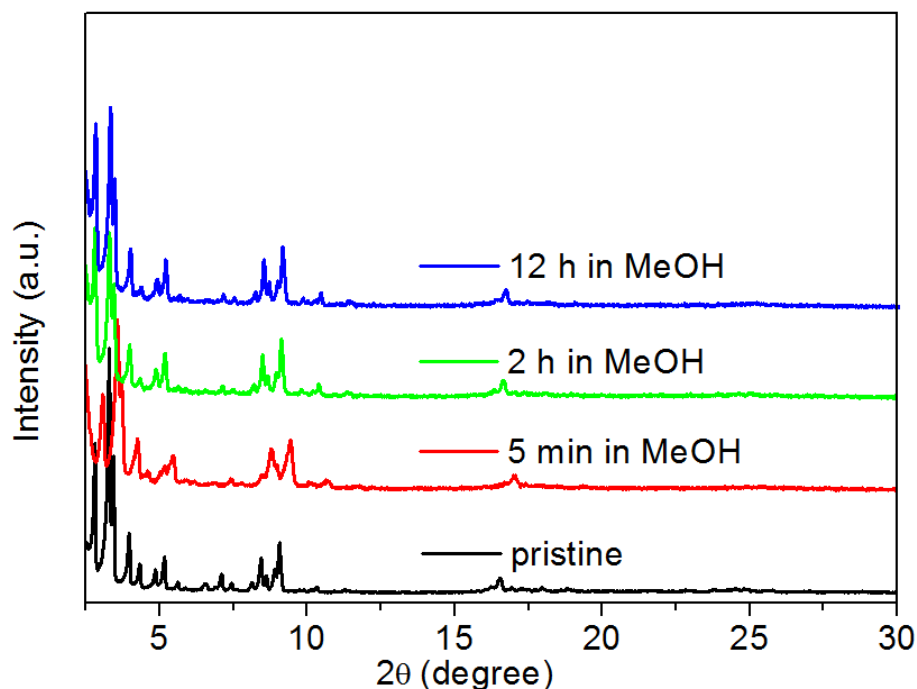


Figure 37. PXRD patterns of Cr-MIL-101-SO₃H-TAEA prepared in different periods of time in MeOH. Control experiments were designed with reaction time as the only variable. The other experimental factors are as follows: alkylamine (TAEA), alkylamine quantity (8 equiv.), reaction temperature (RT), and solvent (MeOH).

Since TAEA contains many polar amine groups, the interaction between TAEA and solvent would be stronger in more polar solvent than that in less polar solvent. As a result, TAEA dispersed in less polar or nonpolar solvents such as CH would have relatively higher chemical potential (**Figure 38**). It serves as a stronger driving force for TAEA molecules to migrate from the solution phase to the MOF phase, and makes the modified MOF have a higher amine group density and a higher CO₂ uptake. Therefore, there should be a reverse relationship between solvent polarity and TAEA loading amount. Our explanation is well supported by the order of solvent polarity (CH (0.006) < THF (0.207) < DCM (0.309) < MeOH (0.762)).⁷⁸ Based on the above thermodynamic

rationalization, a nonpolar solvent, like CH, is the optimal solvent for the incorporation of alkylamine into Cr-MIL-101-SO₃H.

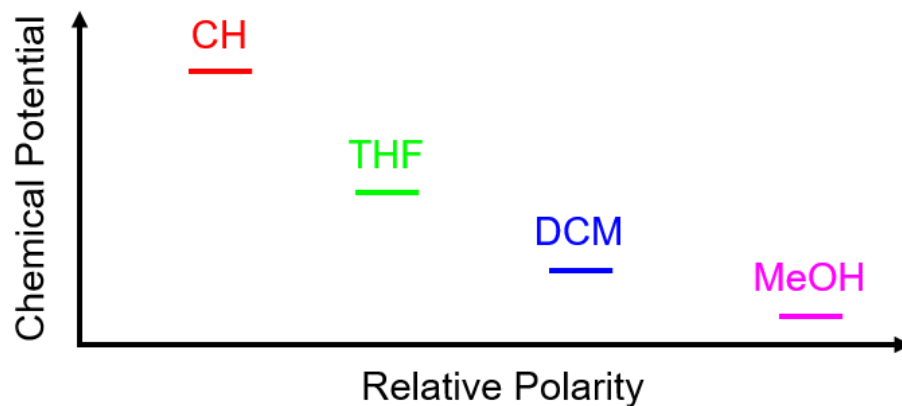


Figure 38. The illustration of the relative chemical potentials of the same alkylamine dispersed in four different solvents.

2.3.6 Optimal Amine Tethering Condition for Cr-MIL-101-SO₃H

According to the results of the above systematic study, the optimal condition to tether TAEA into Cr-MIL-101-SO₃H is to immerse the MOF in CH solution with 8 equiv. TAEA, and stir the suspension for 5 min at RT. The same procedure was also applied to other alkylamines and the gas uptakes of the resulting samples were measured (**Figure 39-41**). Cr-MIL-101-SO₃H-TAEA still excels among other alkylamine modified Cr-MIL-101-SO₃H in CO₂ uptake (**Figure 39**), which is the best CO₂ adsorbent we obtained in this systematic work so far.

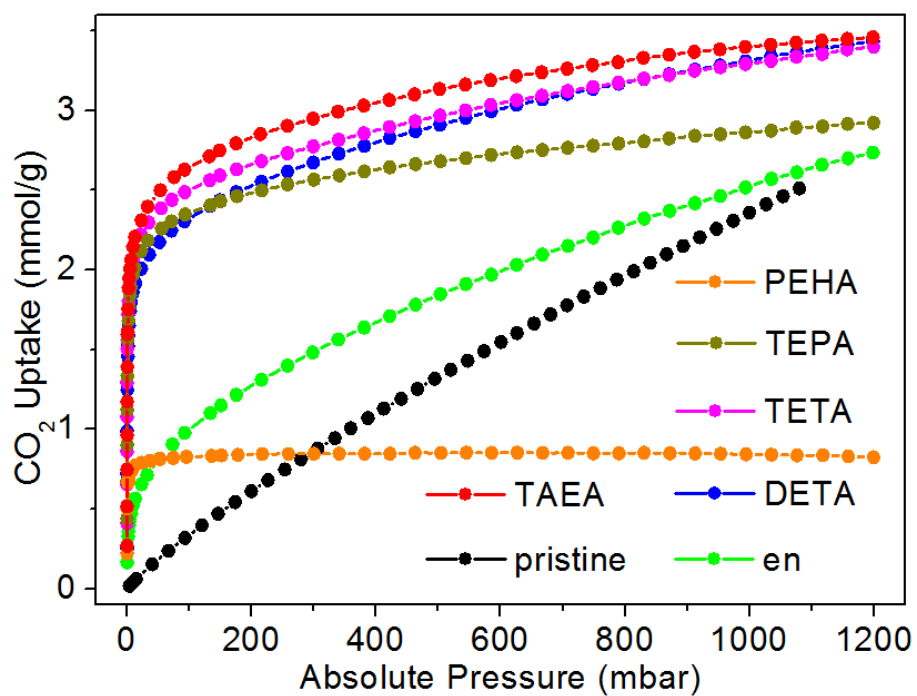


Figure 39. CO₂ uptakes at 20 °C of Cr-MIL-101-SO₃H modified with 8 equiv. different alkylamines. Control experiments were designed with alkylamine structure as the only variable. The other experimental factors are as follows: alkylamine quantity (8 equiv.), reaction time (5 min), reaction temperature (RT), and solvent (CH).

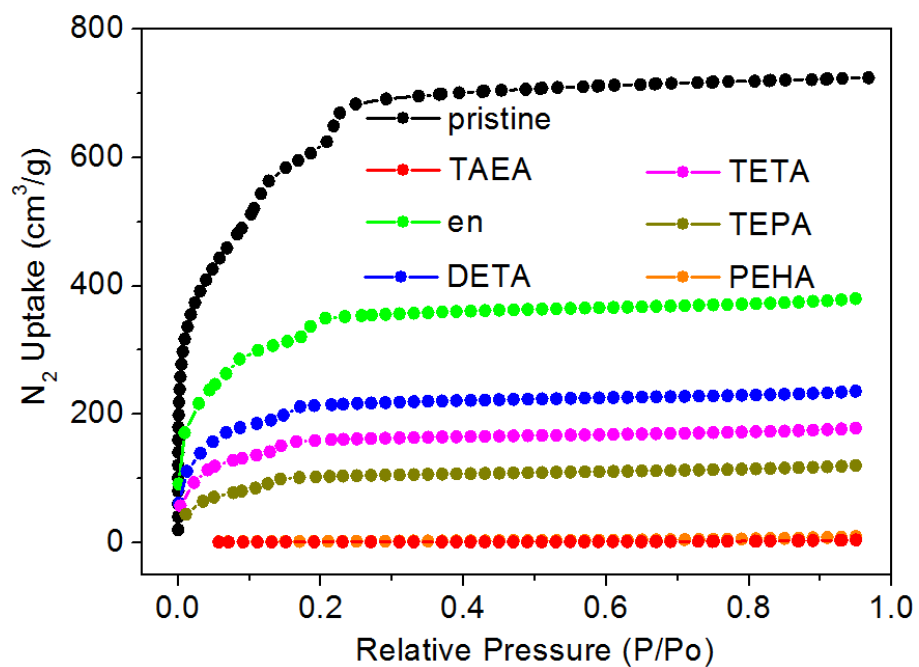


Figure 40. N₂ uptakes at 77 K of Cr-MIL-101-SO₃H modified with 8 equiv. different alkylamines. Control experiments were designed with alkylamine structure as the only variable. The other experimental factors are as follows: alkylamine quantity (8 equiv.), reaction time (5 min), reaction temperature (RT), and solvent (CH).

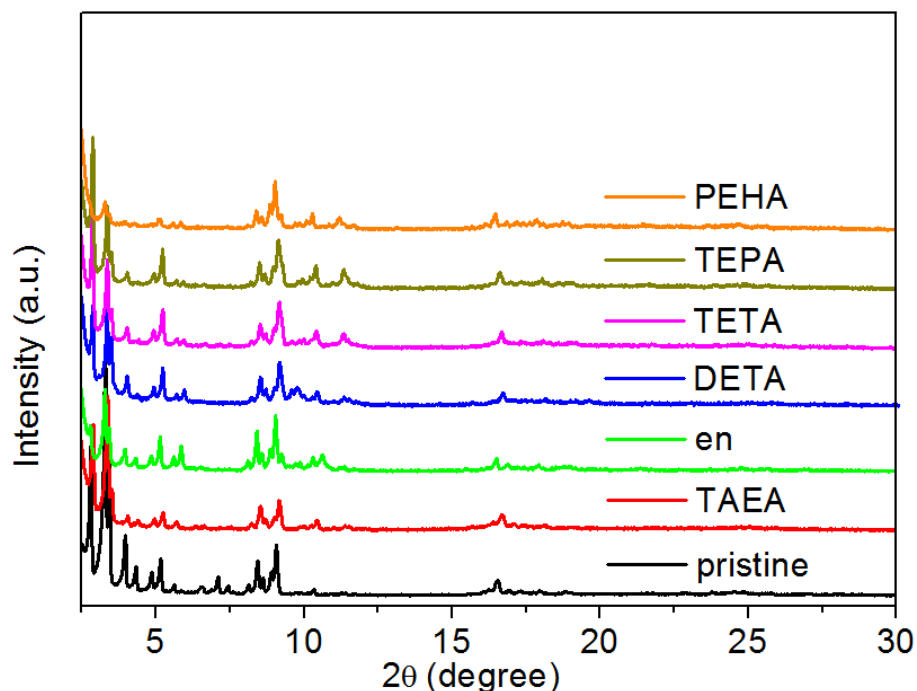


Figure 41. PXRD patterns of Cr-MIL-101-SO₃H modified with 8 equiv. different alkylamines. Control experiments were designed with alkylamine structure as the only variable. The other experimental factors are as follows: alkylamine quantity (8 equiv.), reaction time (5 min), reaction temperature (RT), and solvent (CH).

2.3.7 Isotheric Heat of Adsorption and Recyclability Test

Isotheric heat of CO₂ adsorption of Cr-MIL-101-SO₃H-TAEA was calculated based on the CO₂ uptake data collected at 0 °C, 20 °C and 40 °C (**Figure 42**) with a triple-site Langmuir model (**Equation 1-2**).

Langmuir parameters are listed as follows:

Adsorption site A:

$$q_{\text{sat,A}} = 1.8754 \text{ mmol/g}, \quad b_{0,A} = 1.8888\text{e-}17 \text{ pa}^{-1}, \quad E_A = 86.70 \text{ KJ/mol}$$

Adsorption site B:

$$q_{\text{sat,B}} = 0.7061 \text{ mmol/g}, \quad b_{0,B} = 5,4666\text{e-}13 \text{ pa}^{-1}, \quad E_B = 48.73 \text{ KJ/mol}$$

Adsorption site C:

$$q_{\text{sat,C}} = 2.5056 \text{ mmol/g}, b_{0,\text{C}} = 1.2574 \times 10^{-10} \text{ pa}^{-1}, E_{\text{C}} = 26.04 \text{ KJ/mol}$$

The isosteric heat of CO₂ adsorption is 87 kJ/mol at zero coverage (**Figure 43**). It indicates a strong interaction between CO₂ and Cr-MIL-101-SO₃H-TAEA, which is likely to be the result of the fact that a CO₂ molecule interacts with multiple amine groups in the framework.⁷⁹ The high Q_{st} value is also consistent with its high CO₂ uptake at very low pressure. The N₂ uptakes of Cr-MIL-101-SO₃H-TAEA at 0 °C, 20 °C and 40 °C are all too low to be accurately measured, suggesting the CO₂/N₂ selectivity would be extremely high.

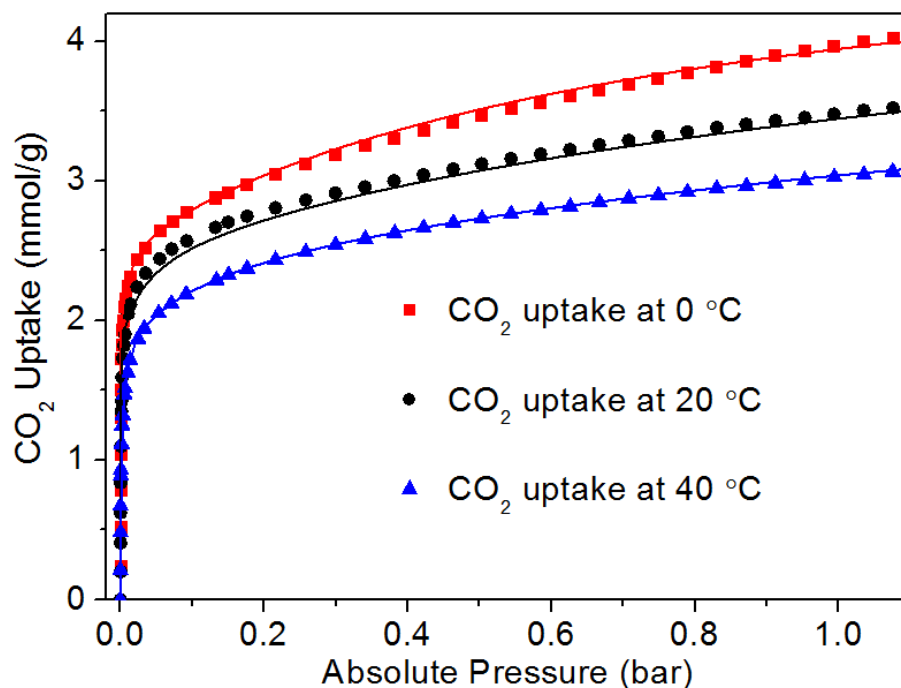


Figure 42. CO₂ adsorption isotherms at 0 °C, 20 °C and 40 °C of Cr-MIL-101-SO₃H-TAEA prepared at the optimal experimental conditions. Each isotherm has a triple-site Langmuir fitting curve.

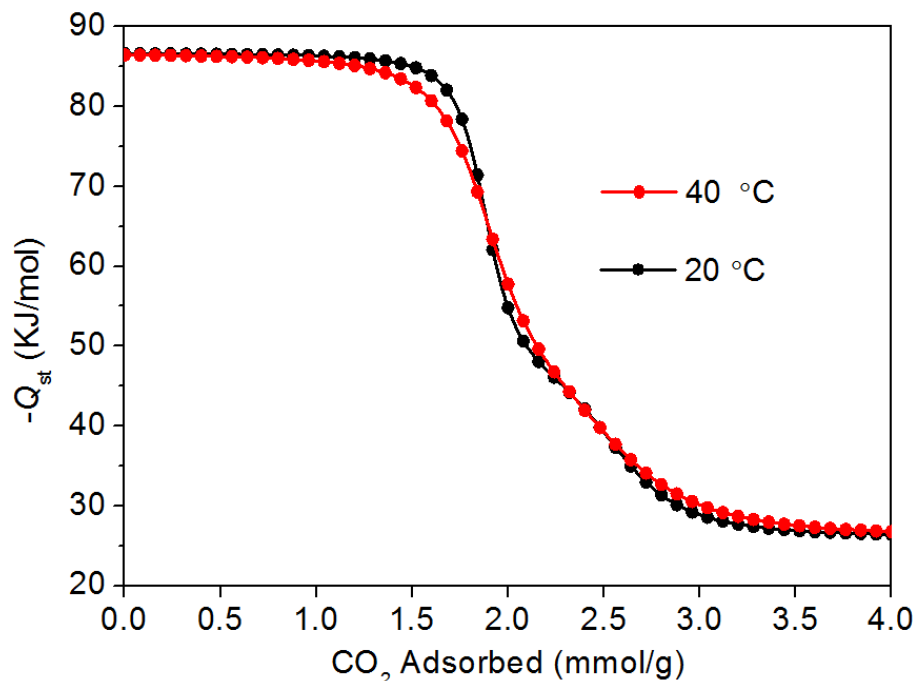


Figure 43. Isosteric heat of CO₂ adsorption at 20 °C and 40 °C of Cr-MIL-101-SO₃H-TAEA prepared at the optimal experimental conditions.

To assess Cr-MIL-101-SO₃H-TAEA as a regenerable adsorbent, we simulated temperature and vacuum swings by saturating the sample with CO₂ gas up to 150 mbar at 40 °C or 0.4 mbar at 20 °C, followed by a high vacuum for 60 min at 80 °C. After 15 cycles, there was no apparent loss in CO₂ uptake, with 2.28 mmol/g at 150 mbar and 40 °C (**Figure 44**, the condition related to CO₂ capture from flue gas), and 1.12 mmol/g at 0.4 mbar and 20 °C (**Figure 45**, the condition relevant to CO₂ capture directly from air). Therefore, it is evident that Cr-MIL-101-SO₃H-TAEA is an efficient and regenerable adsorbent with very strong affinity towards CO₂, suggesting the good potential of their applications in CO₂ capture.

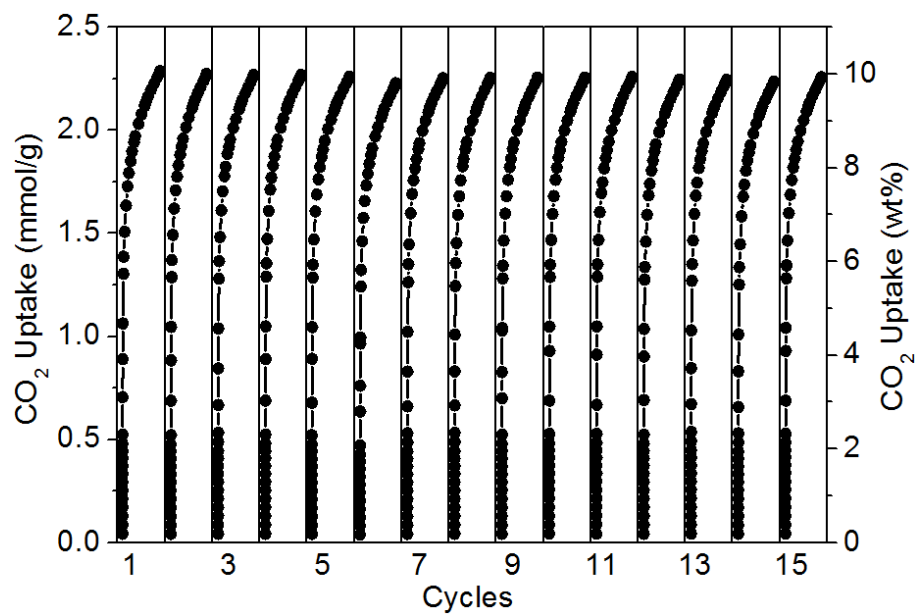


Figure 44. Fifteen cycles of CO₂ adsorption at 150 mbar and 40 °C of Cr-MIL-101-SO₃H-TAEA prepared at the optimal experimental conditions.

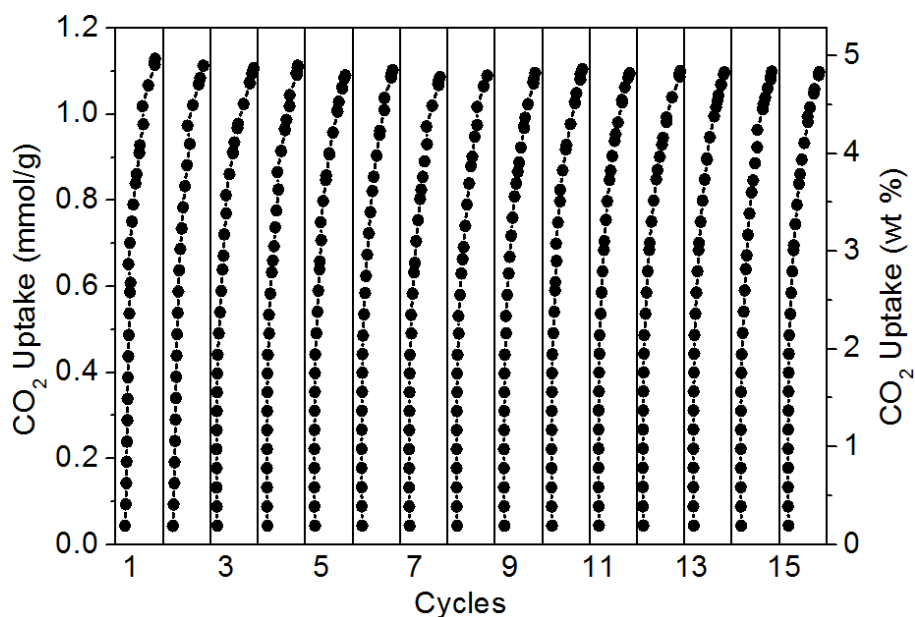


Figure 45. Fifteen cycles of CO₂ uptake at 0.4 mbar and 20 °C of Cr-MIL-101-SO₃H-TAEA prepared at the optimal experimental conditions.

2.4 Conclusion

Based on the acid-base reaction between sulfonic acid groups and amine groups, we successfully incorporated alkylamines into Cr-MIL-101-SO₃H through a fast and mild approach. We also systematically optimized the experimental factors in the amine tethering process to maximize the CO₂ uptake of the resulting adsorbent. The experimental outcomes in optimization were rationalized from a thermodynamic perspective. At the optimal amine tethering condition, the obtained Cr-MIL-101-SO₃H-TAEA has high cyclic CO₂ uptakes of 2.28 mmol/g at 150 mbar and 40 °C, and 1.12 mmol/g at 0.4 mbar and 20 °C. Beyond that, the inexpensive starting materials and simple synthetic procedure for Cr-MIL-101-SO₃H-TAEA all promise its superiority in large-scale production in real application.

CHAPTER III

**THERMODYNAMICALLY DIRECTED METHOD TO TETHER
ALKYLAMINE INTO METAL-ORGANIC FRAMEWORK FOR
EXCEPTIONALLY HIGH CO₂ UPTAKE**

3.1 Introduction

Global warming continues to be a severe environmental problem due to the increasing atmospheric CO₂ concentration,^{4, 80, 81} which is mainly attributed to the combustion of fossil fuels.¹ To contain the CO₂ concentration in the atmosphere, researchers have been striving hard to develop solid adsorbents for CO₂ capture,⁸²⁻⁸⁴ especially amine-functionalized solid materials.^{85, 86} These materials have multiple advantages, such as strong and specific affinity towards CO₂ over other gas species, low regeneration energy for their solid nature, less corrosive effect on infrastructure, etc.^{85, 86} Among them, metal-organic frameworks (MOFs), with high porosity, high surface area and tunable structure,^{87, 88} stand out as a promising candidates for amine tethering. A lot of attempts have been made to explore their CO₂ capture capability.^{5, 10-13, 15-20, 65, 89}

Cr-MIL-101 is a highly porous framework with two types of mesopores and large cage windows to allow the incorporation of functional species.⁷⁶ Furthermore, it can survive in many harsh chemical conditions,^{72, 90-92} including highly basic alkylamine solution.⁷³ Therefore, Cr-MIL-101 is regarded as a good platform to conduct alkylamine tethering for CO₂ capture. In addition, the cheap starting reagents and easy synthetic procedure of Cr-MIL-101 make it facile to prepare the material in large quantities for industrial applications.⁹³⁻⁹⁵ Preliminary attempts were made to tether alkylamine into Cr-

MIL-101, however, the CO₂ capacity of the obtained materials were not very impressive.^{73, 96,97} As we known, the experimental condition applied in amine-tethering procedure can significantly affect the performance of the resulting materials.¹⁵ It encouraged us to develop a more efficient method to incorporate alkylamine into Cr-MIL-101 and improve its carbon capture capacity.

Herein, we report a thermodynamically directed method to tether alkylamine into a highly stable mesoporous MOF, Cr-MIL-101, aiming for an enhanced alkylamine loading amount and a significantly improved CO₂ uptake.

3.2 Experimental Section

3.2.1 Chemicals

Chromium (III) nitrate nonahydrate (Cr(NO₃)₃·9H₂O), terephthalic acid (H₂BDC), hydrofluoric acid 48-51 wt% (HF), ethanol (EtOH), ammonium fluoride (NH₄F), acetone, cyclohexane (CH), dichloromethane (DCM), diethylenetriamine (DETA), tris(2-aminoethyl)amine (TAEA), 40% w/w sodium deuterioxide (NaOD) solution in deuterium oxide (D₂O) solution(99.5%) were purchased from VWR. All commercial chemicals were used without further purification unless otherwise mentioned.

3.2.2 Instruments

Powder X-ray diffraction (PXRD) was carried out with a BRUKER D8-Focus Bragg-Brentano X-ray Powder Diffractometer equipped with a Cu sealed tube ($\lambda = 1.54178 \text{ \AA}$) at 40 kV and 40 mA. Images and analyses of SEM/EDX were taken by FEI Quanta 600 FE-SEM. The Quanta 600 FEG is a field emission scanning electron microscope capable of generating and collecting high-resolution and low-vacuum images.

It is equipped with a motorized x-y-z-tilt-rotate stage, providing the following movements: x = y = 150 mm (motorized); z = 65 mm (motorized); Tilt +70 degrees to -5 degrees (motorized); Source: Field emission gun assembly with Schottky emitter source. Voltage: 200 V to 30 kV. Beam Current: >100 nA. Equipment associated with the Quanta 600 includes: conventional Everhart-Thornley detector, back-scattered electron detector, IR-CCD chamber camera, Oxford EDX system equipped with X-ray mapping and digital imaging, HKL/Oxford EBSD system incl. geological phase database for phase ID, Gatan panchromatic cathodoluminescence detector with RGB filters and a Zyvex S100 nanomanipulator. N₂ and CO₂ adsorption-desorption isotherms were measured using a Micromeritics ASAP 2020 system at different temperatures. Nuclear magnetic resonance (NMR) analysis was taken by Mercury 300 (300 MHz routine walkup H/C system). Thermogravimetric analysis (TGA) were performed on Mettler Toledo TGA/DSC (DSC = Differential Scanning Calorimetry) 1 Star System. The TGA instrument is coupled with a mass spectrometer (OmniStar ThermoStar GSD320 Gas Analysis System) for the analysis of gas species.

3.2.3 Preparation of Cr-MIL-101

A mixture of Cr(NO₃)₃·9H₂O (400 mg, 1.0 mmol), terephthalic acid (H₂BDC, 166 mg, 1.0 mmol), deionized water (4.75 mL), and HF (48-51 wt%, 20 μL) was added into a 5 mL autoclave and heated at 200 °C for 8 h.⁷² After the mixture was cooled to room temperature (RT), the as-synthesized solid was washed successively with 100 mL 95:5 EtOH:H₂O (v/v) solution at 80 °C for 24 h, 90 mL 30 mmol/L NH₄F solution at 70 °C for 24 h, and 67 mL deionized water at 90 °C for 3h.. The resulting solid was further washed

with acetone for three times and desiccated in air, before being activated at 160 °C for 12h under vacuum prior to further use.

3.2.4 Preparation of Alkylamine Modified Cr-MIL-101

Two different methods were employed to incorporate alkylamine into Cr-MIL-101 with solvent as the main discrepancy, detailed below.

Method A: Activated Cr-MIL-101 (100 mg, 0.146 mmol) was added to an alkylamine (1.11 mmol) solution in anhydrous dichloromethane (DCM, 8 mL). The suspension was stirred at RT for 24 h. The resulting solid was washed with anhydrous DCM for three times and desiccated under a Schlenk line.

Method B: Activated Cr-MIL-101 (100 mg, 0.146 mmol) was added to an alkylamine (1.11 mmol) solution in anhydrous cyclohexane (CH, 8 mL). The suspension was stirred at RT for 5 min. The resulting solid was washed with anhydrous CH for three times, and desiccated under a Schlenk line.

Sample 1 and **2** were prepared according to **Method A** using diethylenetriamine (DETA) and tris(2-aminoethyl)amine (TAEA) as alkylamine, respectively, and were activated at 60 °C. **Sample 3** and **4** were prepared according to **Method B** using DETA and TAEA as the alkylamine, respectively, and were activated at 55 °C.

3.2.5 Gas Sorption Measurement of Alkylamine-Modified Cr-MIL-101

The gas sorption isotherms were collected using Micromeritics ASAP 2020 system. N₂ uptakes were measured at 77 K, while CO₂ uptakes were measured at 0 °C, 25 °C, and 40 °C.

3.2.6 PXRD Measurement of Alkylamine-Modified Cr-MIL-101

The PXRD patterns of the samples were collected using a BRUKER D8-Focus Bragg-Brentano X-ray Powder Diffractometer equipped with a Cu sealed tube ($\lambda = 1.54178 \text{ \AA}$) at 40 kV and 40 mA. The ranges of 2θ are all set to be 2-35 degree.

3.2.7 EDX Measurement of Alkylamine-Modified Cr-MIL-101

The sample was first desiccated to remove moisture or solvent residues. After that, it is finely ground before being applied on a conductive tape which is adhered to the EDX sample holder. After the sample was placed into the SEM/EDX instrument, the inner pressure was decreased below 10^{-6} torr before measurements were conducted. The scanning areas were all set to be approximately $40 \times 40 \mu\text{m}^2$ to detect the element species.

3.2.8 Quantitative Analysis of the Organic Components in Alkylamine-Modified Cr-MIL-101 Using NMR

10 mg activated alkylamine-modified Cr-MIL-101 sample was added to a solution prepared from 200 μL 40% w/w NaOD solution in D_2O (99.5%) and 1200 μL D_2O for decomposition. After 24 h, the suspension became clear pale green solution, which was analyzed using NMR spectroscopy for the molar ratio between BDC and TAEA in the sample.

3.2.9 TGA Measurement of Alkylamine-Modified Cr-MIL-101

The sample (around 10 mg) was placed in an aluminum pan, and kept under helium at 30 $^\circ\text{C}$ for 15 min to remove solvent residues. After that, the TGA measurement was initiated with the increasing rate of temperature set as 5 $^\circ\text{C}/\text{min}$. The emitted gas species were detected using an OmniStar ThermoStar GSD320 Gas Analysis System

The recyclability of the sample was also performed with TGA. About 13 mg sample was loaded into the aluminum pan. In the preparation step, the sample was maintained in a He flow at 40 °C for 10 min before being heated to 70 °C to remove the remaining solvent or moisture. After 10 min, the temperature was lowered back to 40 °C and also held for 10 min before the recyclability test. In each cycle, the adsorption condition was 40 °C with a gas flow of 15% CO₂ and 85% He, while the desorption condition was 90 °C with a pure He flow. The adsorption and desorption time were set to be 30 min and 10 min, respectively. After each cycle, the sample was maintained in a He flow at 40 °C for 10 min before the next cycle.

3.2.10 Dynamic Column Breakthrough Experiments*

Dynamic column breakthrough experiments (**Figure 46**) were carried out using a home-built setup coupled with a mass spectrometer (Hiden QGA).^{98,99} Dry MOF powders (0.2-0.4 g for 7 cm column) were packed into a stainless-steel column (diameter: 0.46 cm). The packed column was purged under a constant He flow (8 ± 0.5 mL min⁻¹) before experiments. CO₂/N₂ [(15±1)/(85±1)] binary mixture gas with a flow rate of 9.5 ± 0.5 cc min⁻¹ was introduced through the by-pass line with a resistance (controlled by a needle valve) comparable to the pressure drop (0-0.2 bar) to calibrate the system dead volume. After that, a helium flow (8 ± 0.5 mL min⁻¹) was introduced to clean the whole system. Breakthrough experiments were initiated by switching the purge gas (He) to CO₂/N₂ binary mixture gas [(15±1)/(85±1) composition, 9.5 ± 0.5 cc min⁻¹, stabilized for 30 min

* The breakthrough experiment was performed by Dr. Zhigang Hu and Prof. Dr. Dan Zhao of the Department of Chemical and Biomolecular Engineering in National University of Singapore.

in advance]. An argon flow was kept at 5 cc min^{-1} during the experiments as an internal reference gas to calibrate the flow rate of exit gas. The upstream and downstream pressures were recorded by pressure gauges. The original breakthrough curves were obtained as a relative $\text{CO}_2/\text{N}_2/\text{Ar}$ composition versus elapse time and then converted to molar fraction ratio versus elapse time.^{10, 16, 100}

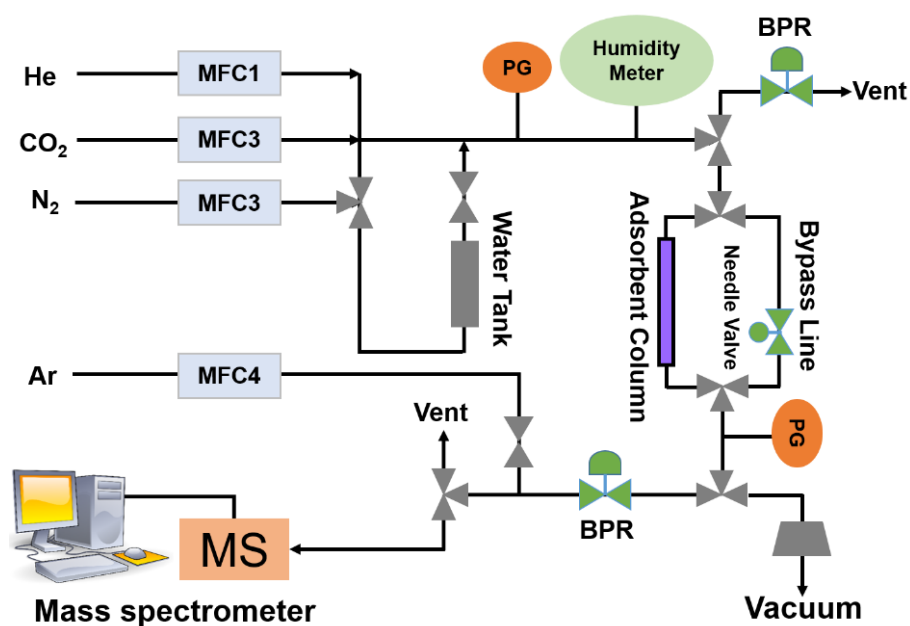


Figure 46. The layout of the home-built breakthrough equipment.

3.2.11 Selectivity Calculation Based on Breakthrough Experiment Data*

The simplified model for breakthrough related calculations is presented in **Figure 47**.

* The selectivity calculation based on breakthrough experiment data was performed by Dr. Zhigang Hu and Prof. Dr. Dan Zhao of the Department of Chemical and Biomolecular Engineering in National University of Singapore.

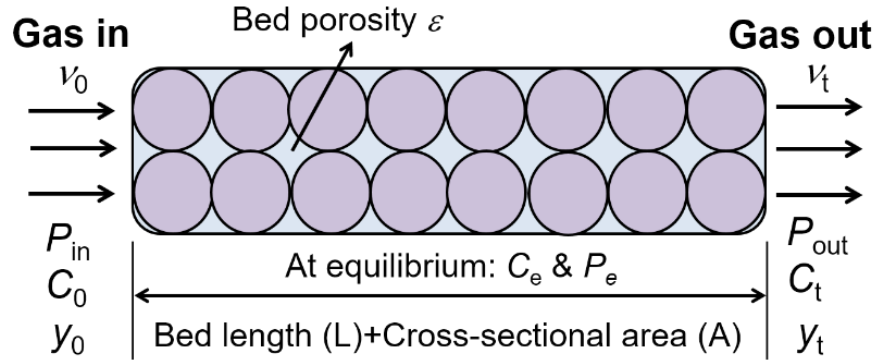


Figure 47. Model and parameters for adsorptive separation calculations.

From the mass balance across the whole packed column (*In-Out = Accumulation*), the following equations can be obtained:

$$\varepsilon v_0 C_0 A t_c - \int_0^{t_c} \varepsilon v_t C_t A dt = LA \varepsilon C_e + LA(1 - \varepsilon) q_e \quad \text{(Equation 4)}$$

$$C_0 = \frac{P_{in}}{R \times T}; C_t = \frac{P_{out}}{R \times T} \quad \text{(Equation 5)}$$

Where ε is the bed porosity, v_o is the interstitial feed gas velocity (cm s^{-1}), C_o is the inlet gas concentration (mmol cm^{-3}), A is the cross sectional area of column (cm^2), t_c is the elution time (s), v_t is the interstitial outlet gas velocity (cm s^{-1}), C_t is the outlet gas concentration (mmol cm^{-3}), C_e is the average gas concentration at adsorption equilibrium (mmol cm^{-3}), P_{in} and P_{out} , and P_e are the inlet (upstream), outlet (downstream), and equilibrium pressure (assume an average value of upstream and downstream pressure), respectively, L is the column length (cm), r_p is the radius of crystal size; q_e is the equilibrium concentration of adsorbate in the adsorbent (mmol cm^{-3}).^{98,101,102}

According to Darcy's Law, the cross-column pressure drop is related to the viscosity of gas phase (μ), interstitial gas velocity and column length:

$$\frac{\Delta P}{L} = \frac{150}{4r_p^2} \left(\frac{1-\varepsilon}{\varepsilon} \right)^2 \mu v_t \quad \text{(Equation 6)}$$

Since the product of viscosity of gas phase (μ), interstitial gas velocity (v_t) can be deemed as a constant, it is reasonable to assume the equilibrium pressure (P_e) equals to:

$$P_e \approx \frac{P_{in} + P_{out}}{2}; C_e \approx \frac{P_{in} + P_{out}}{2R \times T} \quad \text{(Equation 7)}$$

After rearrangement, the **Equation 4** for mean residence time \bar{t} is obtained:

$$\int_0^{t_f} \left(1 - \frac{v_t C_t}{v_0 C_0} \right) dt = \bar{t} = \frac{L}{v_0} \left[\frac{P_{in} + P_{out}}{2P_{in}} + \frac{1-\varepsilon}{\varepsilon} \left(\frac{q_e}{C_0} \right) \right] \quad \text{(Equation 8)}$$

For each component, y_0 and y_t represent the inlet and outlet molar fraction:

$$\begin{aligned} y_0 &= y_t^\infty; v_0 = v_t^\infty \times \frac{P_{out}}{P_{in}}; \frac{v_t}{v_0} = \frac{v_t}{v_t^\infty} \times \frac{P_{in}}{P_{out}} \\ \frac{C_t}{C_0} &= \left(\frac{y_t P_{out}}{R \times T} \right) / \left(\frac{y_0 P_{in}}{R \times T} \right) = \frac{y_t}{y_0} \times \frac{P_{out}}{P_{in}}; \\ \frac{C_t}{C_0} \times \frac{v_t}{v_0} &= \frac{v_t y_t}{v_t^\infty y_t^\infty} \end{aligned} \quad \text{(Equation 9)}$$

The following equation can be obtained:

$$\int_0^{t_f} \left(1 - \frac{v_t y_t}{v_t^\infty y_t^\infty} \right) dt = \bar{t} = \frac{L}{v_0} \left[\frac{P_{in} + P_{out}}{2P_{in}} + \frac{1-\varepsilon}{\varepsilon} \left(\frac{q_e}{C_0} \right) \right] \quad \text{(Equation 10)}$$

The mean residence time obtained from the experimentally measured breakthrough responses after subtracting blank residence time (dead volume calibration) can be used to calculate q_e/C_0 . The adsorption selectivity (S_{ads}) at equilibrium pressure can be calculated:

$$S_{\text{ads}} = \frac{(q_e / C_0)_{\text{CO}_2}}{(q_e / C_0)_{\text{N}_2}} \quad \text{(Equation 11)}$$

3.3 Results and Discussion

3.3.1 CO₂ Uptakes and Pore Volumes of the Alkylamine-Modified MOFs

The CO₂ uptakes of **Sample 1, 2, 3** and **4** were measured at 25 °C to evaluate their CO₂ capture abilities. (**Figure 48**) Their significantly different performance reveals the chemistry involved in the amine tethering process. Compare the samples prepared with the same method (**Sample 1, 2** and **Sample 3, 4**), the CO₂ uptake of TAEA-modified Cr-MIL-101 at 150 mbar is approximately 10% higher than that of DETA-tethered one in both sets of the data. (**Table 29**) This suggests that TAEA is able to endow Cr-MIL-101 with a stronger affinity towards CO₂ than DETA, which might be ascribed to the higher number of amine groups in a TAEA molecule. By contrast, the samples prepared with the same alkylamine but different methods (**Sample 1, 3** and **Sample 2, 4**) display remarkably distinct CO₂ adsorption behaviors. The CO₂ uptakes of **Sample 3** and **4** at 150 mbar are more than twice as much as those of **Sample 1** and **2**, respectively. This strongly indicates that solvent is a more crucial experimental factor in determining the CO₂ capture performance of the alkylamine-modified MOFs.

Judging from the pore size distribution diagrams of the four samples derived from their N₂ adsorption isotherms at 77 K (**Figure 49-53**), the size and population of mesopores in the frameworks considerably decrease after alkylamine incorporation. Since their PXRD patterns remain unchanged compared with that of the pristine Cr-MIL-101 (**Figure 54**), the possibility of phase change or structural collapse of the MOFs is

eliminated. Therefore, the reduction in mesopores is mainly attributed to the incorporation of alkylamine into the frameworks.

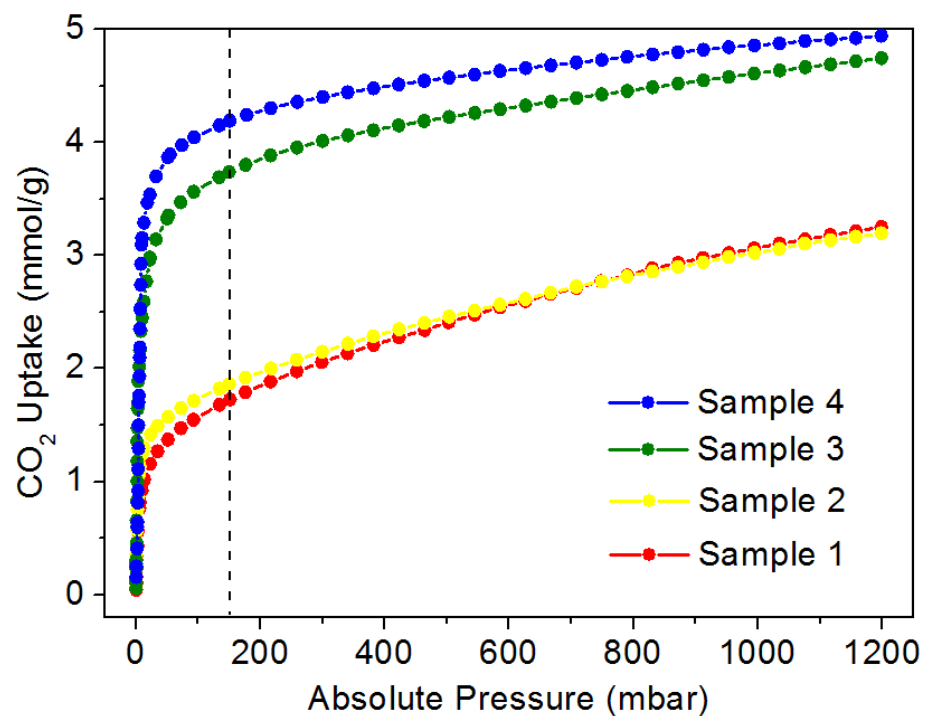


Figure 48. CO₂ uptakes at 25 °C of **Sample 1-4**.

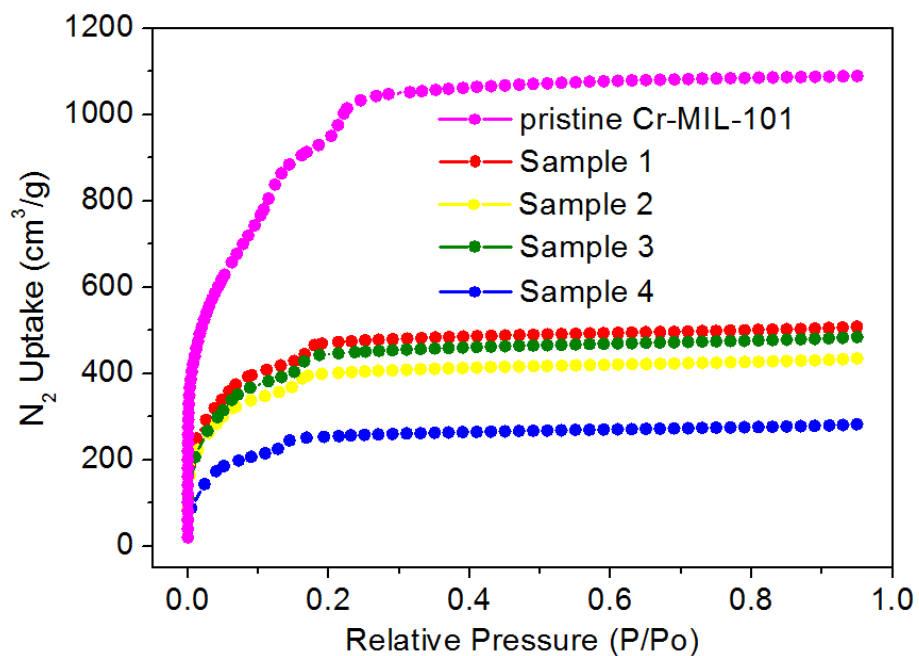


Figure 49. N₂ uptakes at 77 K of **Sample 1-4** in comparison with that of the pristine Cr-MIL-101.

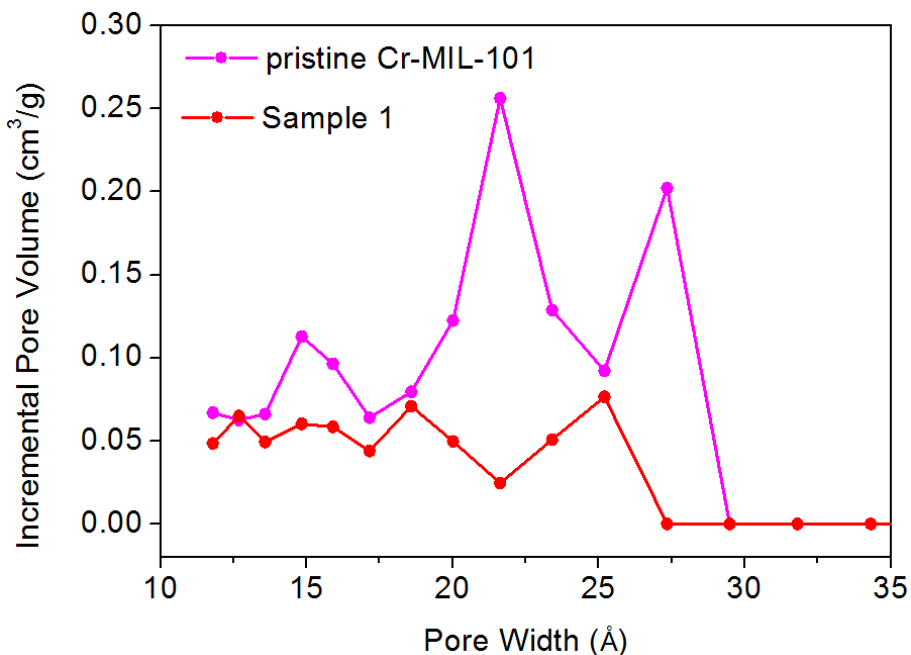


Figure 50. Pore size distribution of **Sample 1** in comparison with that of the pristine Cr-MIL-101. The total pore volume in **Sample 1** ($\leq 272.71 \text{ \AA}$) is determined to be $0.67651 \text{ cm}^3/\text{g}$.

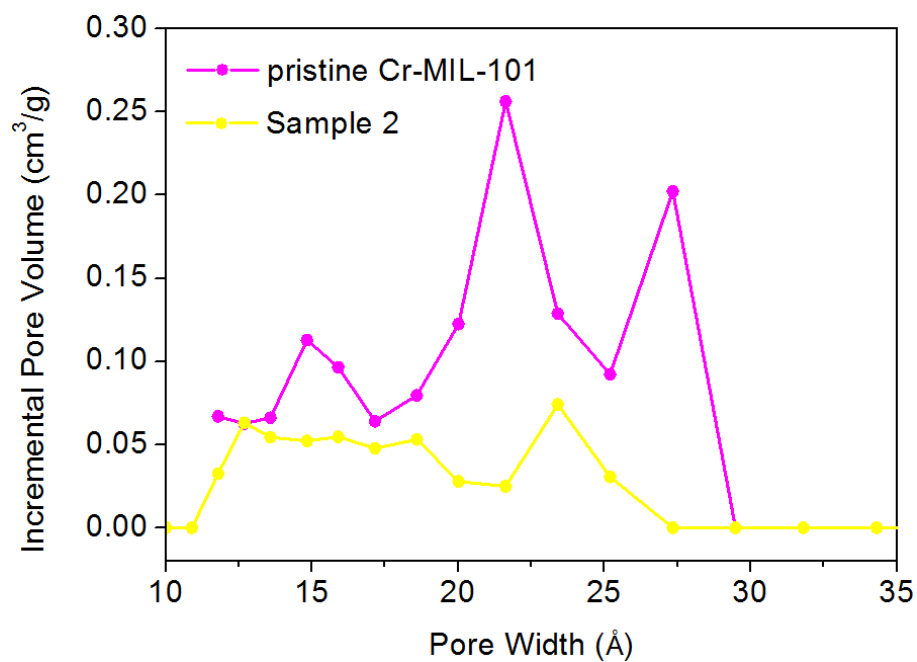


Figure 51. Pore size distribution of **Sample 2** in comparison with that of the pristine Cr-MIL-101. The total pore volume in **Sample 2** ($\leq 272.71 \text{ \AA}$) is determined to be $0.57573 \text{ cm}^3/\text{g}$.

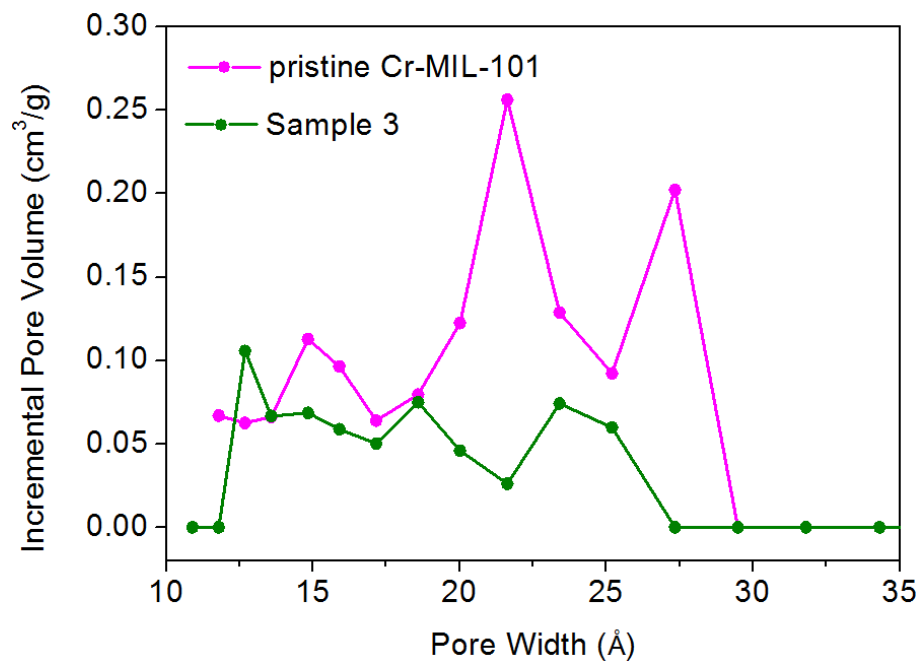


Figure 52. Pore size distribution of **Sample 3** in comparison with that of the pristine Cr-MIL-101. The total pore volume in **Sample 3** ($\leq 272.71 \text{ \AA}$) is determined to be $0.64595 \text{ cm}^3/\text{g}$.

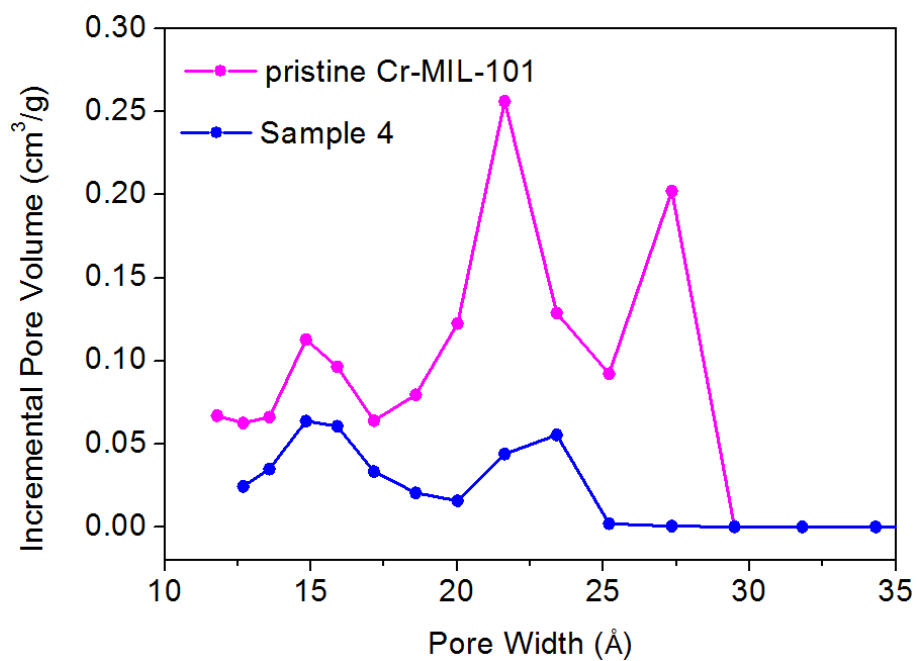


Figure 53. Pore size distribution of **Sample 4** in comparison with that of the pristine Cr-MIL-101. The total pore volume in **Sample 4** ($\leq 272.71 \text{ \AA}$) is determined to be $0.35535 \text{ cm}^3/\text{g}$.

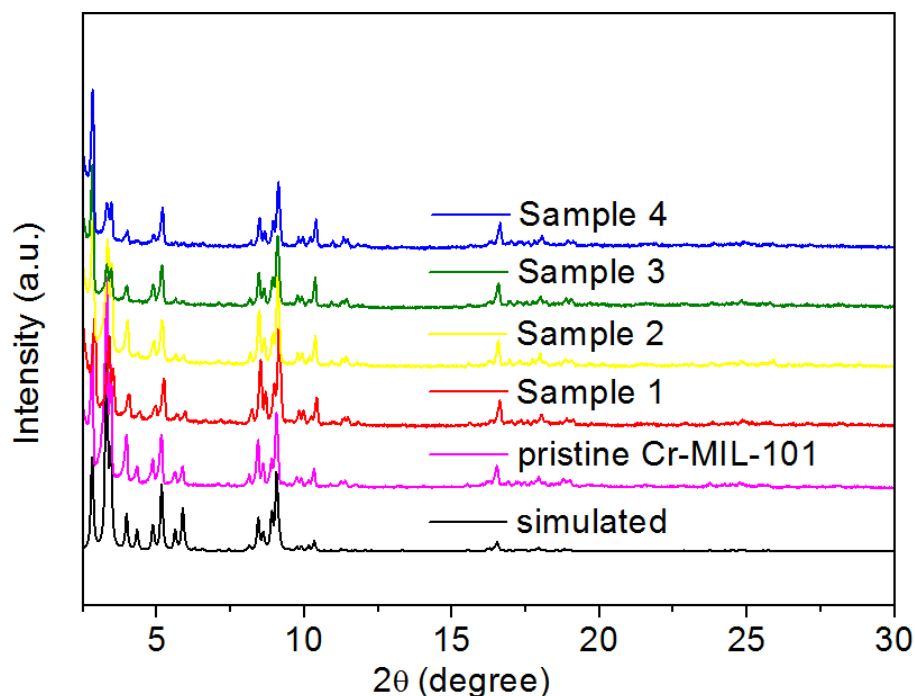


Figure 54. Comparison of the PXRD patterns of **Sample 1-4** with that of pristine Cr-MIL-101 and the simulated pattern.

3.3.2 Thermodynamic Equilibrium in Alkylamine Incorporation

To investigate why solvent is remarkably influential in the CO₂ capture behaviors of the generated samples, it is essential to discern whether the question is thermodynamic- or kinetic-related.

In **Method A**, the reaction time is 24 h, which is sufficiently long for the system to reach its equilibrium. By contrast, the reaction time in **Method B** is 5 min, which is comparatively inadequate to arrive at a similar solid conclusion. To clarify this point, control experiments were conducted by extending the reaction time in **Method B** to 30 min while still utilizing DETA (**Sample 5**) and TAEA (**Sample 6**) as the integrating alkylamines. It turns out that both the CO₂ and N₂ adsorption isotherms of **Sample 5** nearly

overlap those of **Sample 3**. The same result also occurs in the comparison of **Sample 6** and **Sample 4** (Figure 55-59). The above outcomes strongly demonstrate that the equilibria of the corresponding systems were also attained when the samples were produced with **Method B**. As a result, the effect of solvent should be rationalized from a thermodynamic prospective.

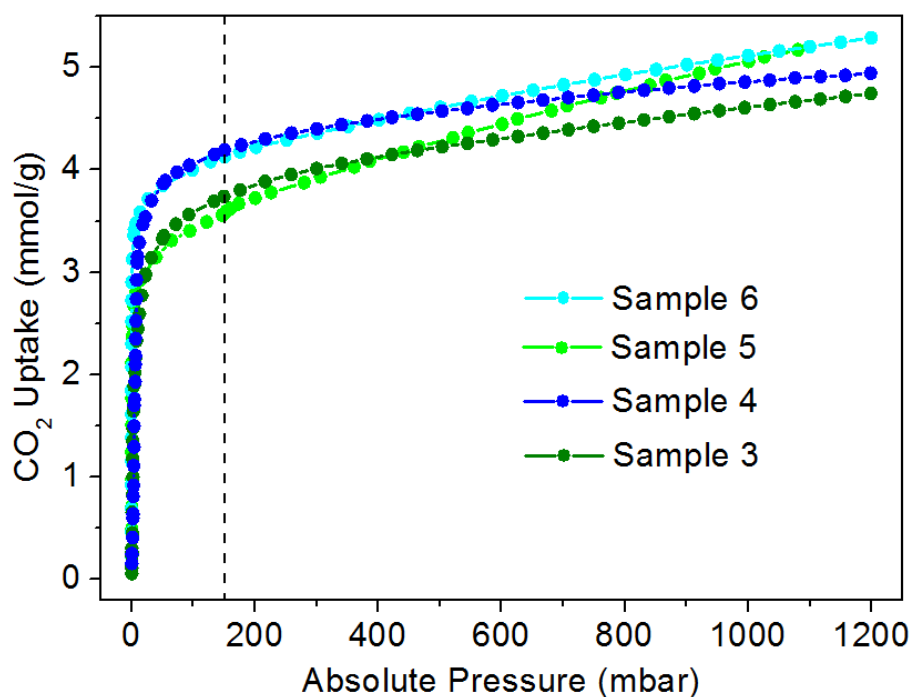


Figure 55. CO₂ uptakes at 25 °C of **Sample 3-6**.

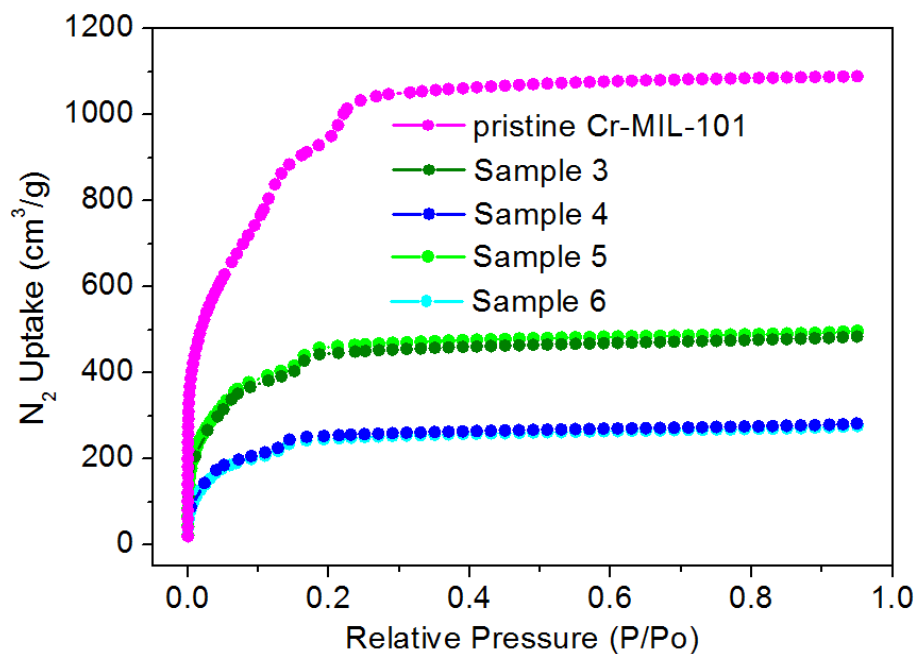


Figure 56. N₂ uptakes at 77 K of **Sample 3-6** in comparison with that of the pristine Cr-MIL-101.

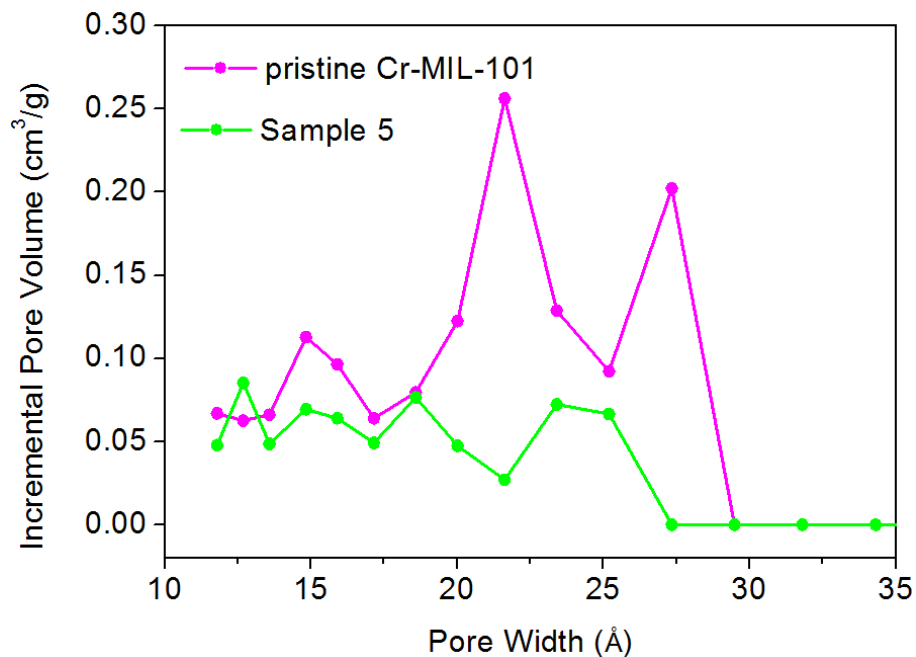


Figure 57. Pore size distribution of **Sample 5** in comparison with that of the pristine Cr-MIL-101. The total pore volume in **Sample 5** ($\leq 272.71 \text{ \AA}$) is determined to be $0.66322 \text{ cm}^3/\text{g}$.

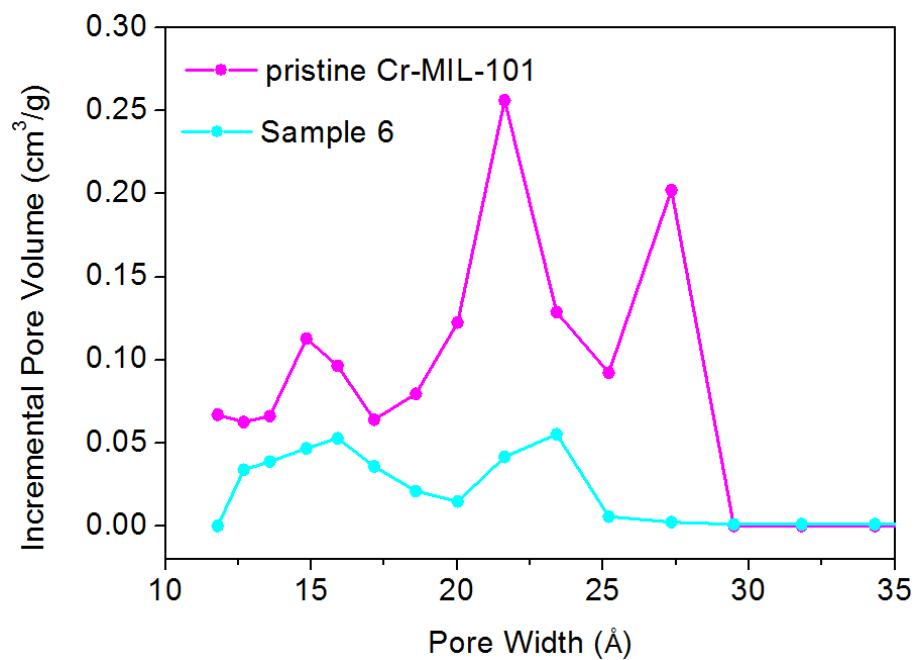


Figure 58. Pore size distribution of **Sample 6** in comparison with that of the pristine Cr-MIL-101. The total pore volume in **Sample 6** ($\leq 216.60 \text{ \AA}$) is determined to be $0.36741 \text{ cm}^3/\text{g}$.

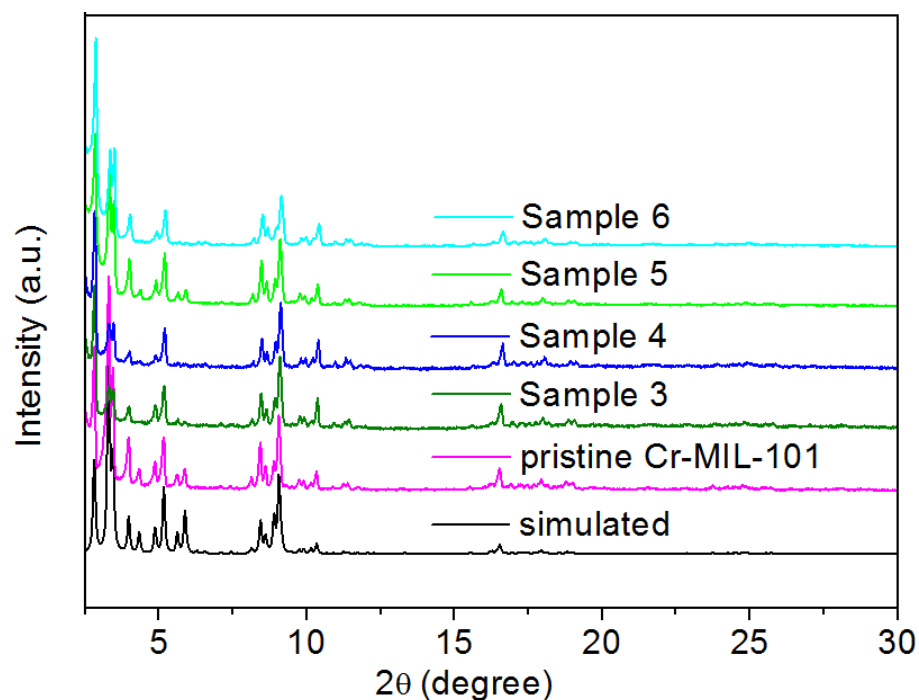


Figure 59. Comparison of the PXRD patterns of **Sample 3-6** with that of pristine Cr-MIL-101 and the simulated pattern.

We propose the polarity of solvent is the essential property that affects the loading amount of alkylamine and consequently the CO₂ uptake of the modified MOF. Compared with DCM, CH is far less polar. Accordingly it has relatively weaker interaction with the alkylamine molecules dispersed in it. This gives rise to a higher chemical potential of the alkylamine in CH than that in DCM. Thermodynamically speaking, the higher chemical potential acts as a stronger driving force for more alkylamine molecules to migrate into the framework from the solution.

The point stated above can also be reasoned in the following way. Assume there are two Cr-MIL-101 samples with equal mass. Alkylamine incorporation are conducted with these two samples using CH and DCM as the solvent, respectively. For easy

comparison, we suppose there are equal amounts of a certain alkylamine molecule integrated into these two sample. Thus, the chemical potentials of alkylamine in the both MOFs are the same, denoted as μ_{MOF} . When a thermodynamic equilibrium is reached between the solution phase and the MOF phase, the chemical potential of alkylamine in the solution $\mu_{\text{c, solvent}}$ should be equal to that in the MOF μ_{MOF} .

$$\mu_{\text{c, CH}} = \mu_{\text{c, CH}}^{\circ} + RT\ln(c_{\text{CH}}/c^{\circ}) = \mu_{\text{MOF}} \quad \text{(Equation 12)}$$

$$\mu_{\text{c, DCM}} = \mu_{\text{c, DCM}}^{\circ} + RT\ln(c_{\text{DCM}}/c^{\circ}) = \mu_{\text{MOF}} \quad \text{(Equation 13)}$$

$\mu_{\text{c, solvent}}^{\circ}$ is the standard chemical potential of alkylamine in the solvent. c_{solvent} is the molar concentration of alkylamine in the solvent. c° is the standard molar concentration.

Therefore we have,

$$\mu_{\text{c, CH}}^{\circ} + RT^{\circ}\ln(c_{\text{CH}}/c^{\circ}) = \mu_{\text{c, DCM}}^{\circ} + RT^{\circ}\ln(c_{\text{DCM}}/c^{\circ})$$

So we have,

$$c_{\text{CH}}/c_{\text{DCM}} = \exp[(\mu_{\text{c, DCM}}^{\circ} - \mu_{\text{c, CH}}^{\circ})/RT] \quad \text{(Equation 14)}$$

Since $\mu_{\text{c, DCM}}^{\circ} - \mu_{\text{c, CH}}^{\circ} < 0$,

Thus, $c_{\text{CH}}/c_{\text{DCM}} < 1$.

This indicates that the c_{CH} required to incorporate a certain amount of alkylamine into Cr-MIL-101 is less than c_{DCM} . Considering the premise that equal quantity of alkylamine is put into the two MOFs, it is obvious that a higher percentage of alkylamine migrate into the framework, suggesting the beneficial role of the less polar CH in driving more alkylamine molecules into the MOF.

Due to the fact that mesopores still exist in **Sample 1-4** after alkylamine incorporation in Cr-MIL-101, internal space is not a limiting factor in their CO_2 adsorption

amounts. Instead, the quantity of the integrated alkylamine determines their CO₂ affinities and thus CO₂ uptakes. Therefore, solvents with smaller polarities are more favorable for the inclusion of alkylamine into MOFs and contribute to MOFs with higher CO₂ capacities (**Figure 60**).

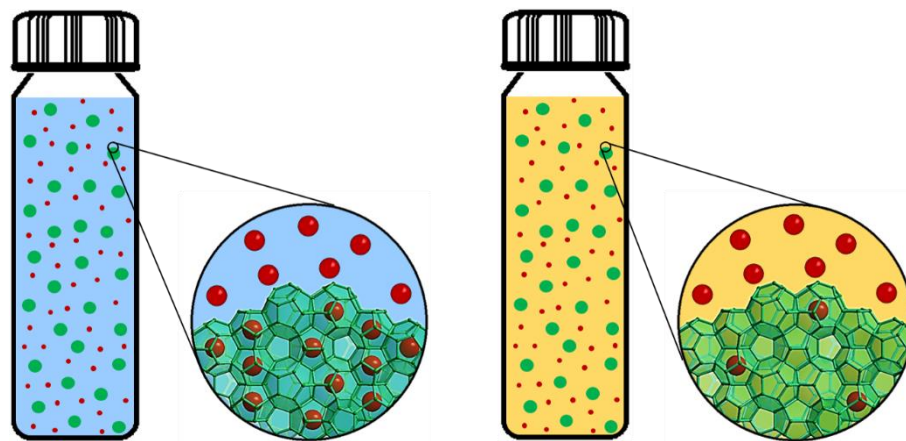


Figure 60. Illustration of the different alkylamine loading amounts in the MOF samples prepared in CH (left) and DCM (right). Cr-MIL-101 is represented with green balls and green networks. Red balls represent alkylamine molecules. Blue and yellow liquids stand for CH and DCM, respectively.

Our explanation is well supported by two classes of experimental evidences. First, the amount of integrated alkylamines is quantified by combining the results from elemental analysis (N/F ratio, **Table 10-11**) and energy dispersive X-ray spectroscopy (F/Cr ratio, **Figure 61-84** and **Table 12-27**), as indicated by N/Cr ratio (**Table 28**). It turns out that MOFs prepared in CH have higher N/Cr values than that in DCM. Moreover, the CO₂ uptakes of these materials are positively correlated to their N/Cr ratios (**Table 29**), well substantiating our rationalization. Second, the quantity of incorporated alkylamine

can also be represented the pore volume of the modified MOFs. Because the framework remain robust during the modification, a low value in pore volume suggests a high loading amount. As shown in **Table 29**, the CO₂ adsorption amounts of these MOFs are negatively correlated to their pore volumes, which is harmoniously consistent with our reasoning as well.

Table 10. EA data of **Sample 1-4**. Each sample was measured twice.

Sample	C weight%	H weight%	N weight%	F weight%
Sample 1	37.35	5.61	8.93	1.68
	37.54	5.61	9.01	1.59
Sample 2	38.27	6.29	10.84	1.30
	38.15	6.18	10.75	1.21
Sample 3	40.22	5.77	11.78	1.28
	40.03	5.77	11.72	1.22
Sample 4	40.64	6.22	14.56	1.10
	40.46	6.16	14.49	1.07

Table 11. Average N weight%, average F weight% and the calculated N/F atomic ratios of **Sample 1-4**.

Sample	Average N weight%	Average F weight%	N/F atomic ratio
Sample 1	8.97	1.64	7.42
Sample 2	10.80	1.26	11.63
Sample 3	11.75	1.25	12.76
Sample 4	14.52	1.08	18.25

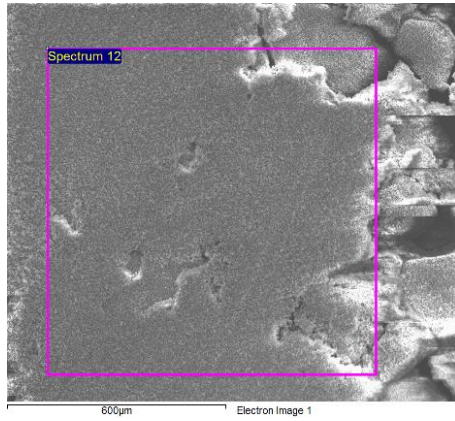


Figure 61. SEM image of **Sample 1** (1st measurement).

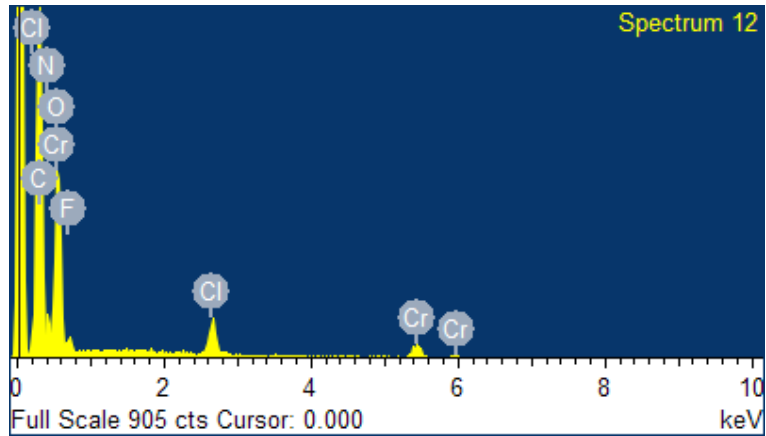


Figure 62. EDX spectrum of **Sample 1** (1st measurement).

Table 12. EDX data for **Sample 1** (1st measurement).

Element	Weight%	Atomic%
C K	40.05	56.55
N K	8.15	9.87
O K	19.55	20.72
F K	3.12	2.78
Cl K	3.83	1.83
Cr L	25.31	8.26
Totals	100.00	

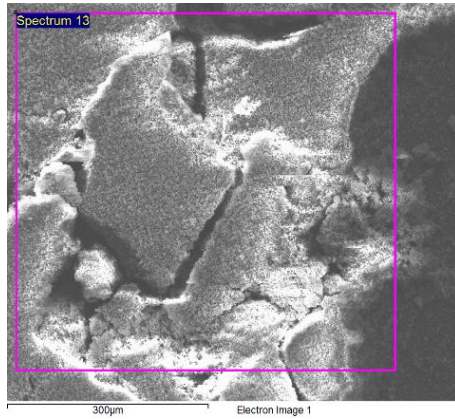


Figure 63. SEM image of **Sample 1** (2nd measurement).

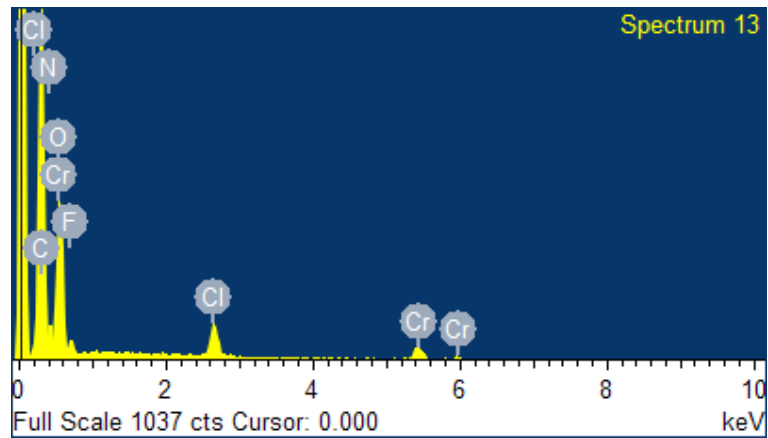


Figure 64. EDX spectrum of **Sample 1** (2nd measurement).

Table 13. EDX data for **Sample 1** (2nd measurement).

Element	Weight%	Atomic%
C K	41.97	57.99
N K	8.10	9.60
O K	20.05	20.80
F K	2.69	2.35
Cl K	3.97	1.86
Cr L	23.22	7.41
Totals	100.00	

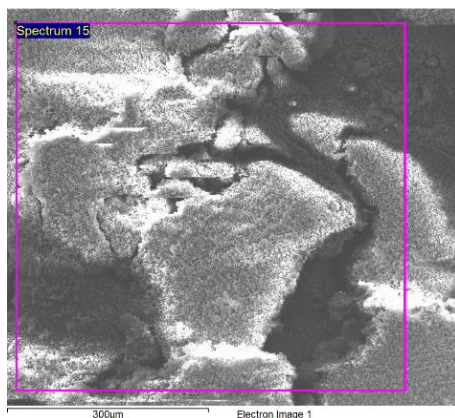


Figure 65. SEM image of **Sample 1** (3rd measurement).

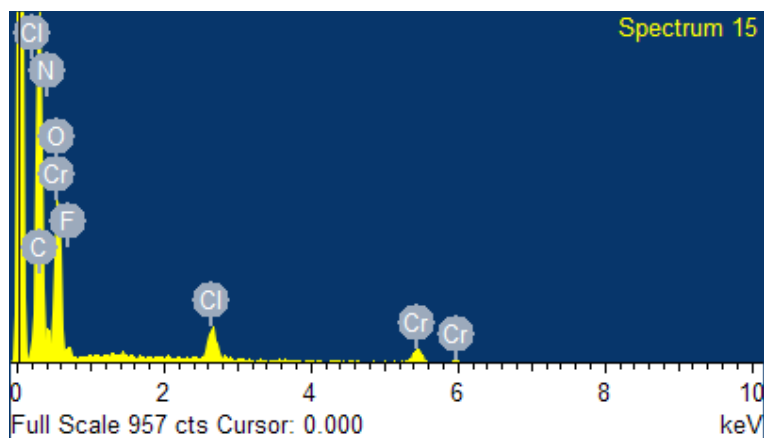


Figure 66. EDX spectrum of **Sample 1** (3rd measurement).

Table 14. EDX data for **Sample 1** (3rd measurement).

Element	Weight%	Atomic%
C K	43.34	58.61
N K	7.73	8.96
O K	21.39	21.71
F K	2.80	2.39
Cl K	4.09	1.87
Cr L	20.65	6.45
Totals	100.00	

Table 15. Summary of the EDX data of **Sample 1**.

	F atomic%	Cr atomic%
1st measurement	2.78	8.26
2nd measurement	2.35	7.41
3rd measurement	2.39	6.45
Average	2.51	7.37

F/Cr molar ratio = F Atomic% / Cr Atomic% = 2.51% / 7.37% = 0.340

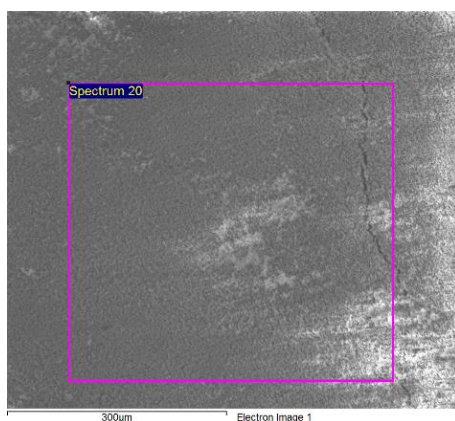


Figure 67. SEM image of **Sample 2** (1st measurement).

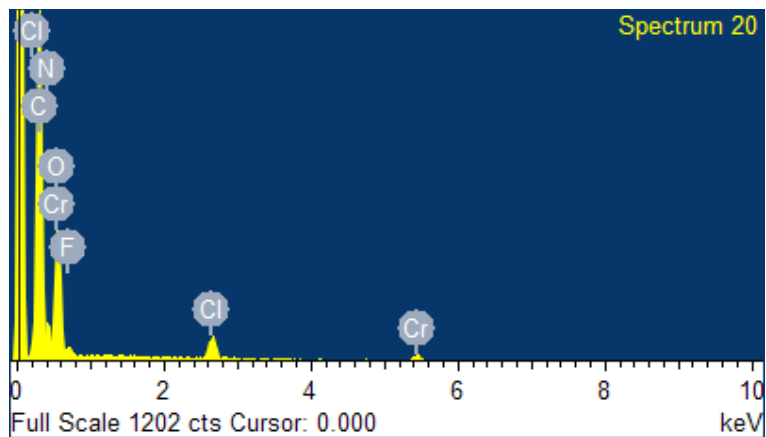


Figure 68. EDX spectrum of **Sample 2** (1st measurement).

Table 16. EDX data for **Sample 2** (1st measurement).

Element	Weight%	Atomic%
C K	43.37	58.72
N K	10.50	12.19
O K	18.30	18.60
F K	2.47	2.11
Cl K	3.03	1.39
Cr L	22.33	6.98
Totals	100.00	

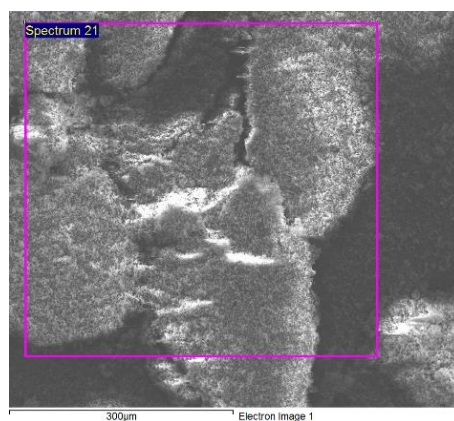


Figure 69. SEM image of **Sample 2** (2nd measurement).

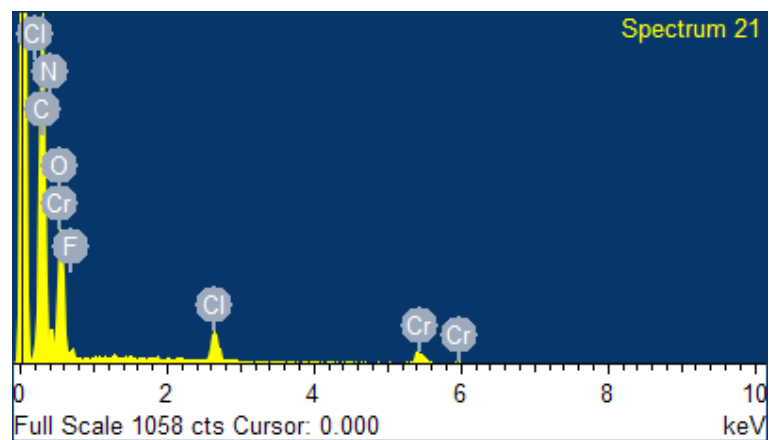


Figure 70. EDX spectrum of **Sample 2** (2nd measurement).

Table 17. EDX data for **Sample 2** (2nd measurement).

Element	Weight%	Atomic%
C K	46.61	60.60
N K	9.71	10.83
O K	20.56	20.07
F K	1.84	1.52
Cl K	4.23	1.86
Cr L	17.04	5.12
Totals	100.00	

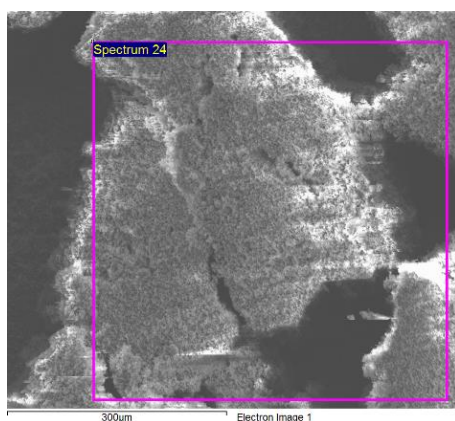


Figure 71. SEM image of **Sample 2** (3rd measurement).

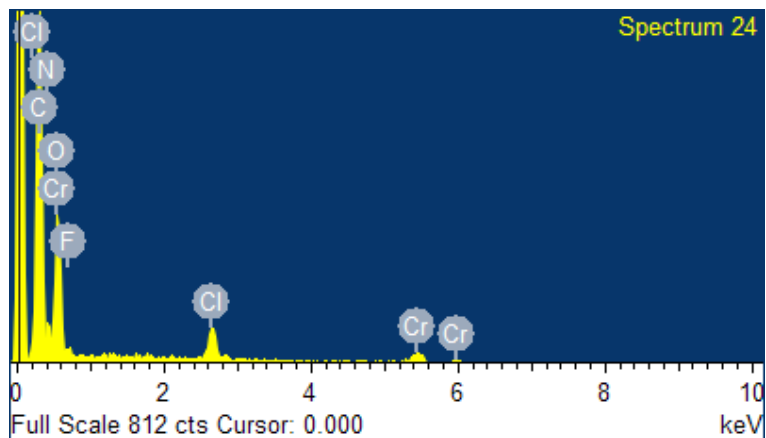


Figure 72. EDX spectrum of **Sample 2** (3rd measurement).

Table 18. EDX data for **Sample 2** (3rd measurement).

Element	Weight%	Atomic%
C K	46.11	60.02
N K	10.05	11.22
O K	20.59	20.12
F K	1.89	1.55
Cl K	4.69	2.07
Cr L	16.66	5.01
Totals	100.00	

Table 19. Summary of the EDX data of **Sample 2**.

	F atomic%	Cr atomic%
1st measurement	2.11	6.98
2nd measurement	1.52	5.12
3rd measurement	1.55	5.01
Average	1.73	5.70

F/Cr molar ratio = F Atomic% / Cr Atomic% = 1.73% / 5.70% = 0.304

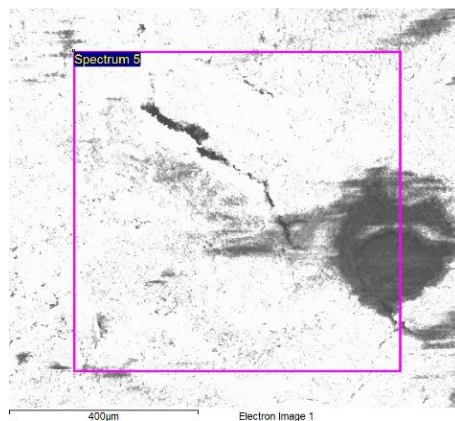


Figure 73. SEM image of **Sample 3** (1st measurement).

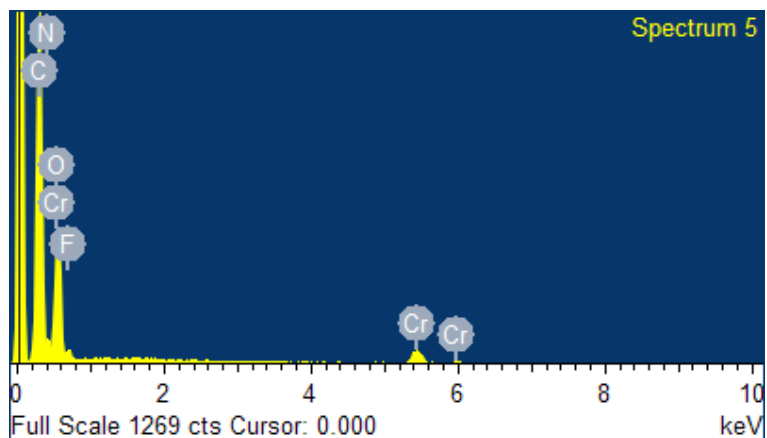


Figure 74. EDX spectrum of **Sample 3** (1st measurement).

Table 20. EDX data for **Sample 3** (1st measurement).

Element	Weight%	Atomic%
C K	44.70	59.80
N K	5.60	6.43
O K	23.89	23.99
F K	3.36	2.84
Cr L	22.46	6.94
Totals	100.00	

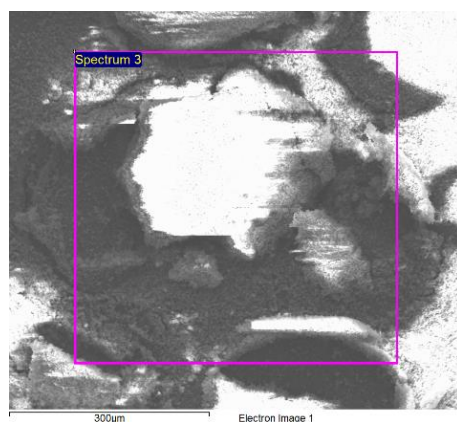


Figure 75. SEM image of **Sample 3** (2nd measurement).

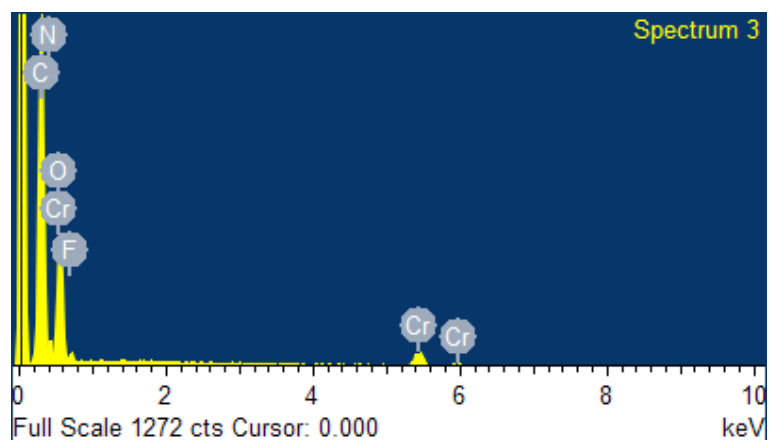


Figure 76. EDX spectrum of **Sample 3** (2nd measurement).

Table 21. EDX data for **Sample 3** (2nd measurement).

Element	Weight%	Atomic%
C K	42.88	59.28
N K	6.38	7.56
O K	21.04	21.84
F K	3.29	2.88
Cr L	26.42	8.44
Totals	100.00	

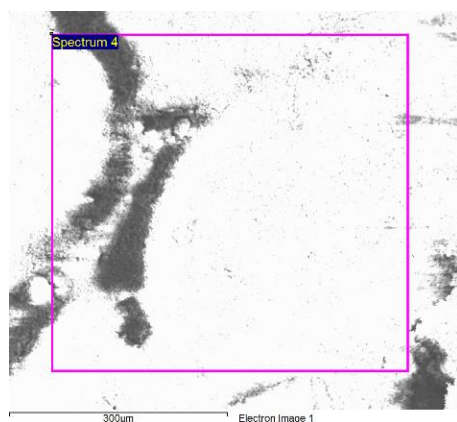


Figure 77. SEM image of **Sample 3** (3rd measurement).

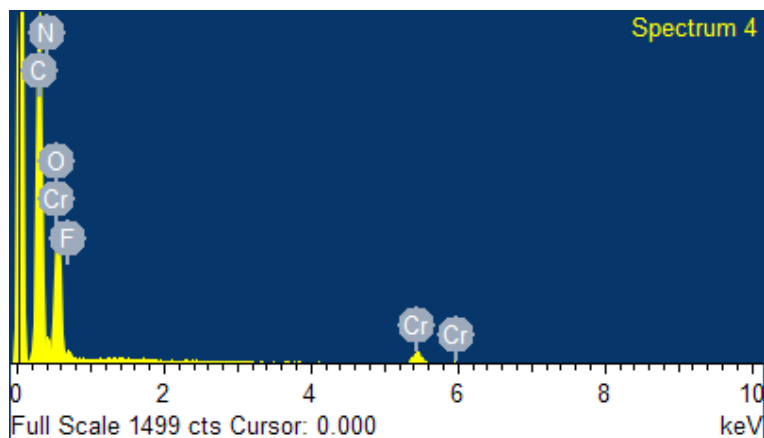


Figure 78. EDX spectrum of **Sample 3** (3rd measurement).

Table 22. EDX data for **Sample 3** (3rd measurement).

Element	Weight%	Atomic%
C K	43.43	59.86
N K	5.20	6.15
O K	22.47	23.25
F K	2.78	2.42
Cr L	26.13	8.32
Totals	100.00	

Table 23. Summary of the EDX data of **Sample 3**.

	F atomic%	Cr atomic%
1st measurement	2.06	6.92
2nd measurement	2.88	8.44
3rd measurement	2.42	8.32
Average	2.45	7.89

$$\text{F/Cr molar ratio} = \text{F Atomic\%} / \text{Cr Atomic\%} = 2.45\% / 7.89\% = 0.310$$

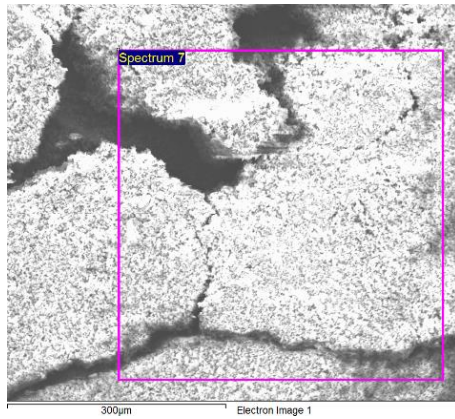


Figure 79. SEM image of **Sample 4** (1st measurement).

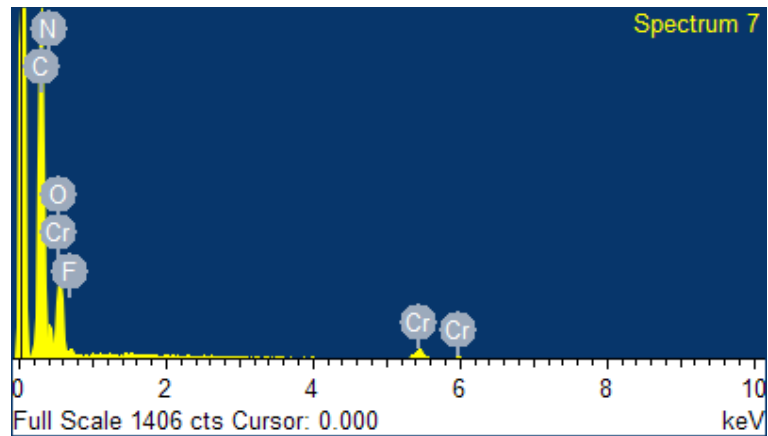


Figure 80. EDX spectrum of **Sample 4** (1st measurement).

Table 24. EDX data for **Sample 4** (1st measurement).

Element	Weight%	Atomic%
C K	46.01	60.10
N K	13.38	14.99
O K	16.87	16.54
F K	2.29	1.89
Cr L	21.46	6.47
Totals	100.00	

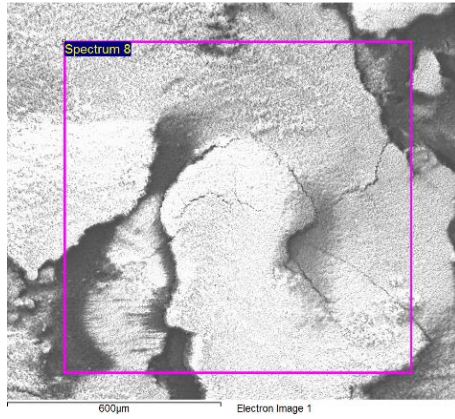


Figure 81. SEM image of **Sample 4** (2nd measurement).

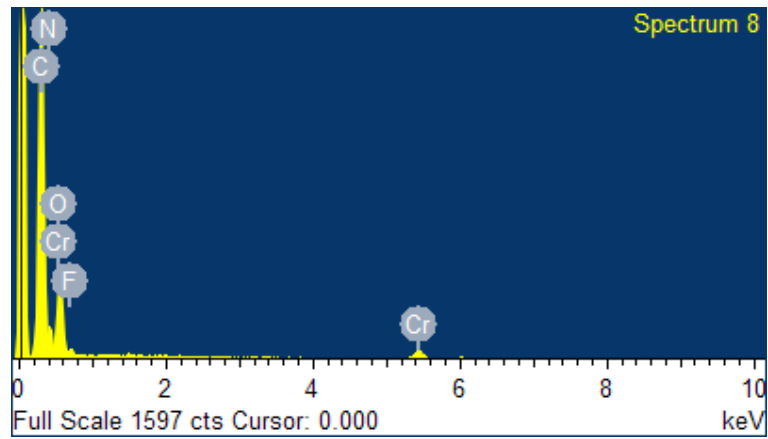


Figure 82. EDX spectrum of **Sample 4** (2nd measurement).

Table 25. EDX data for **Sample 4** (2nd measurement).

Element	Weight%	Atomic%
C K	46.83	60.93
N K	12.84	14.33
O K	17.05	16.65
F K	2.11	1.73
Cr L	21.18	6.36
Totals	100.00	

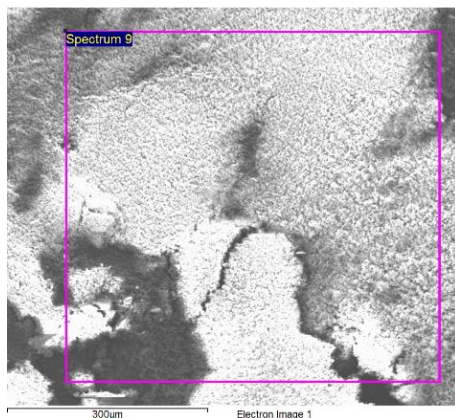


Figure 83. SEM image of **Sample 4** (3rd measurement).

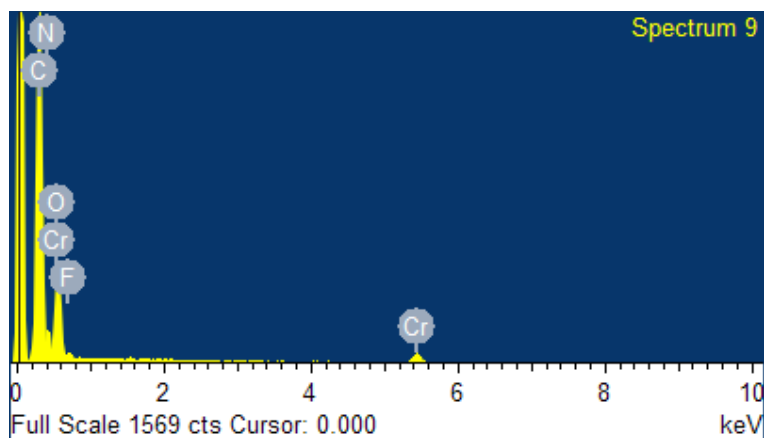


Figure 84. EDX spectrum of **Sample 4** (3rd measurement).

Table 26. EDX data for **Sample 4** (3rd measurement).

Element	Weight%	Atomic%
C K	45.04	59.73
N K	12.22	13.90
O K	17.33	17.25
F K	2.51	2.11
Cr L	22.90	7.02
Totals	100.00	

Table 27. Summary of the EDX data of **Sample 4**.

	F Atomic%	Cr Atomic%
1st measurement	1.89	6.47
2nd measurement	1.73	6.36
3rd measurement	2.11	7.02
Average	1.91	6.62

$$\text{F/Cr molar ratio} = \text{F Atomic\%} / \text{Cr Atomic\%} = 1.91\% / 6.62\% = 0.288$$

Table 28. Summary of the N/F and F/Cr molar ratios of **Sample 1-4**, and their consequent N/Cr molar ratios.

Sample	n(N)/n(F)	n(F)/n(Cr)	n(N)/n(Cr)
Sample 1	7.42	0.34	2.52
Sample 2	11.63	0.30	3.49
Sample 3	12.76	0.31	3.96
Sample 4	18.25	0.29	5.29

Table 29. Summary of the CO₂ uptakes, the total pore volumes, and the molar ratios between alkylamine and Cr in the amine-modified samples.

Sample	CO ₂ uptake (mmol/g) ^a	Total pore volume (cm ³ /g)	n(N)/n(Cr)	n(alkylamine)/n(Cr)
Sample 1	1.70	0.68	2.52	0.84
Sample 2	1.88	0.58	3.49	0.87
Sample 3	3.75	0.64	3.96	1.32
Sample 4	4.20	0.36	5.29	1.32
Sample 5	3.58	0.66	N/A	N/A
Sample 6	4.13	0.37	N/A	N/A

^a. The CO₂ uptakes are the values at 150 mbar on the adsorption isotherms at 25 °C.

The molar ratio between alkylamine and Cr in **Sample 4** was further confirmed by the NMR spectrum of the degraded **Sample 4** in NaOD/D₂O solution (**Figure 85**). The protons on BDC and TAEA were integrated to be 4 and 17.11, respectively. Therefore, $n(\text{TAEA})/n(\text{BDC}) = (17.11/12)/(4/4) = 1.43/1$. Since in Cr-MIL-101, $n(\text{Cr})/n(\text{BDC}) = 1/1$,

it is reasonable to arrive at the conclusion that in **Sample 4**, $n(\text{TAEA})/n(\text{Cr}) = 1.43/1$. This result is comparable to the value obtained by EDX and EA.

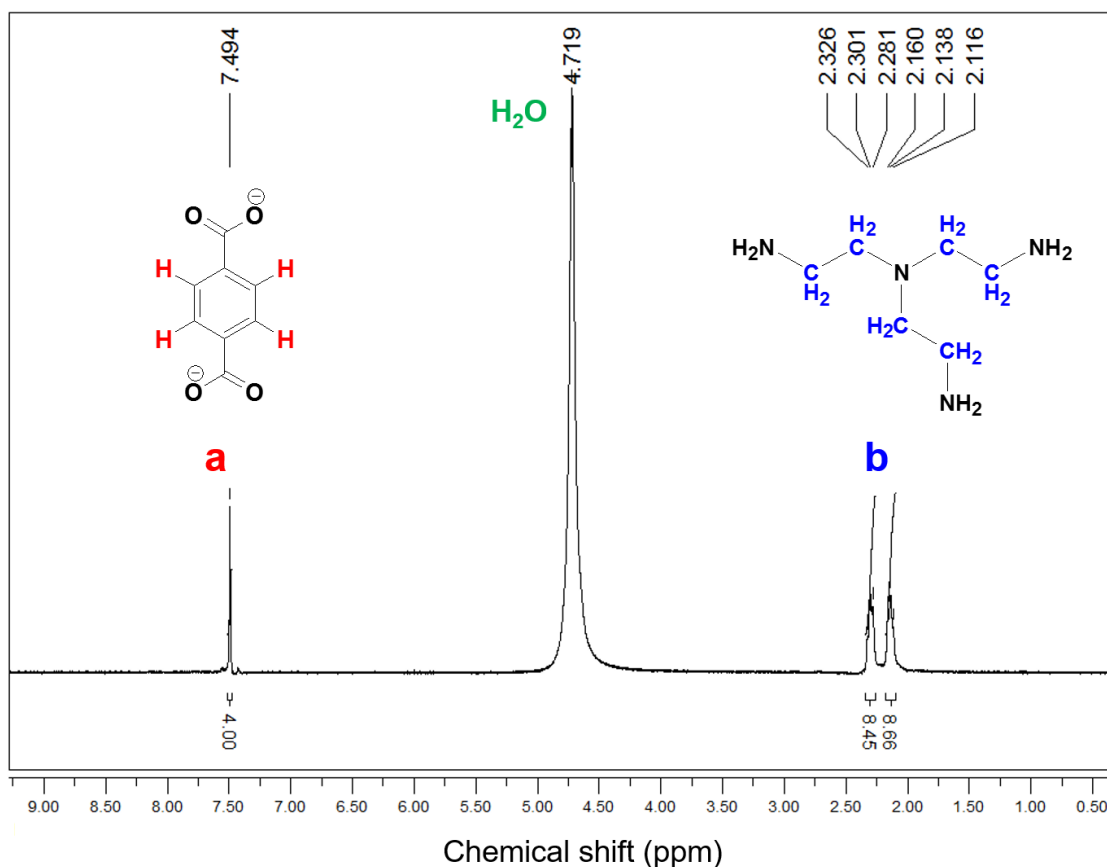


Figure 85. NMR spectrum of the decomposed **Sample 4** in a NaOD solution in D₂O.

3.3.3 Alkylamine Binding Interaction

It is worth noting that for samples prepared in CH, the ratio between alkylamine and Cr is higher than 1 (**Table 29** and **Figure 85**). This implies not all alkylamine molecules are bound onto Cr-MIL-101 through the coordination bonds formed between Cr open metal sites (OMSs) and amine groups. Considering the chemistry environment

around incorporated alkylamines, there are two types of alkylamine binding interaction in these MOFs. One type is the aforementioned coordination bonds generated between Cr and alkylamine (**Type I**), while the other is the hydrogen bonds between the amine groups from two different alkylamine molecules. (**Type II**). As the temperature increases, the amount of the CO₂ adsorbed through the chemical bonds with amine groups will decrease as normal. On the other side, a higher temperature can also break the hydrogen bonds in **Type II** interactions and generate some free amine groups that are ready to capture CO₂. It can lead to an increment in the amount of CO₂ adsorbed at high temperatures.¹⁰³ The net result from these two aspects is that CO₂ uptake has no apparent decrease as the temperature increases, which is likely to result in an overestimated isosteric heat of adsorption $-Q_{st}$. These inferred features in CO₂ adsorption isotherms truly occur in the case of **Sample 4 (Figure 86)**, with CO₂ uptakes as high as 4.21 mmol/g at 25 °C and 150 mbar, and 4.05 mmol/g at 40 °C and 150 mbar. The data provide additional experimental evidence to corroborate the existence of the two types of alkylamine binding interaction in **Sample 4**.

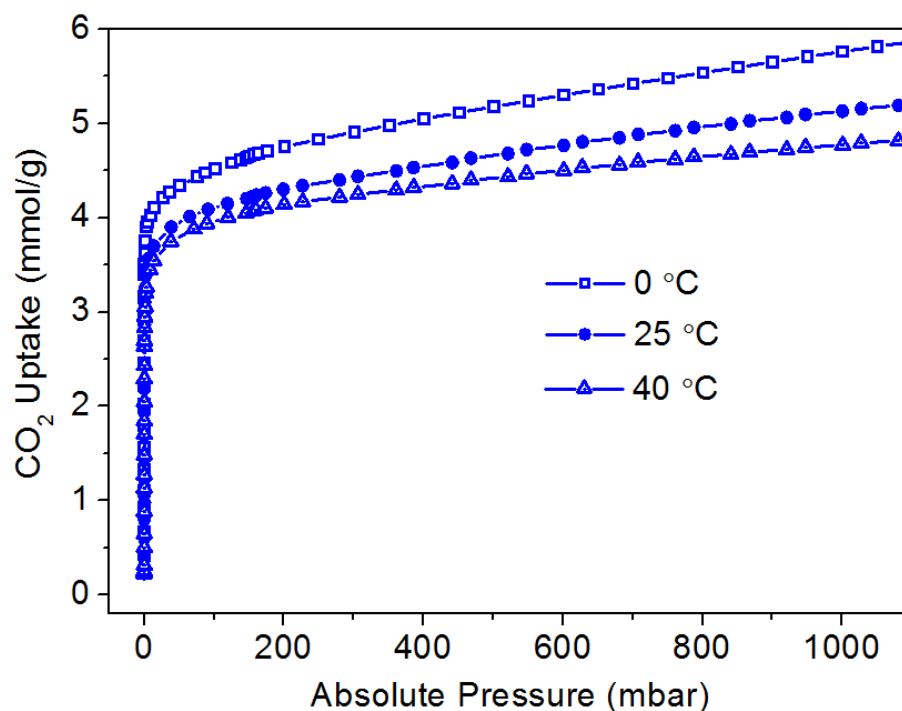


Figure 86. CO₂ adsorption isotherms of **Sample 4** at 0°C, 25 °C and 40 °C.

3.3.4 Thermal Stability of Sample 4 Assessed by TGA-MS

As stated previously, the hydrogen bonds between alkylamines can be relieved as these materials usually suffer from alkylamine leaching problem during desorption procedure. To clarify whether there is alkylamine leaching from **Sample 4** under the CO₂ desorption condition, its thermal stability was evaluated by TGA-MS. The results show that there are no apparent mass peaks of alkylamine before the temperature reaches 150 °C (**Figure 87**). This temperature is higher than the usual CO₂ desorption temperature. It demonstrates even **Type II** binding interaction is firm enough to fixate the alkylamines anchored in Cr-MIL-101 and there will be no leaching of alkylamine during recovery of the material. Moreover, because of the structural confinement in the framework, it is very likely that the alkylamine molecules have multiple hydrogen bonds with their surrounding

molecules. This would endow the amine-modified MOFs with higher thermal stability as well.

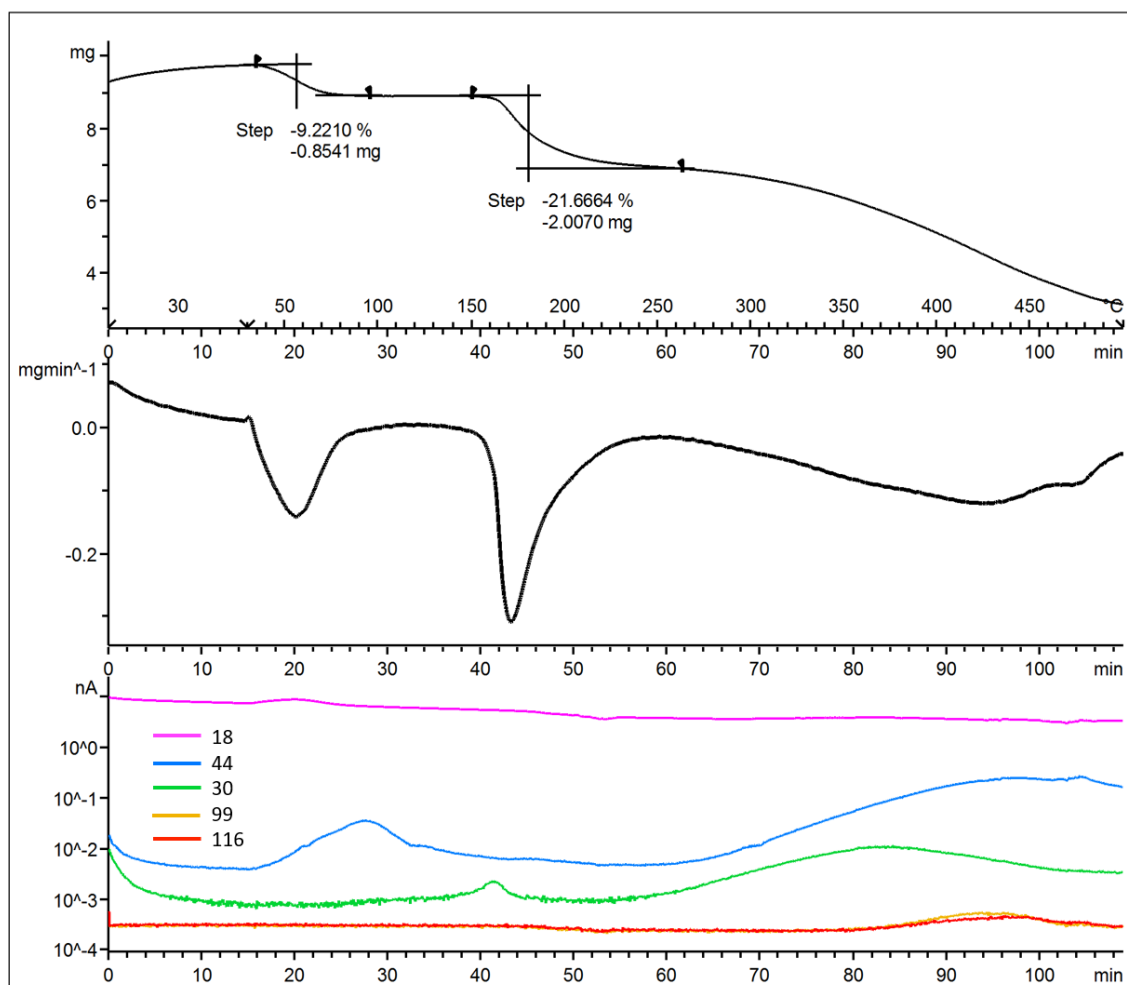


Figure 87. The TGA curve (upper), the first derivative of the TGA curve (middle) and the mass spectrum of the released species (lower) of **Sample 4** as the temperature rises from 30 °C to 500 °C.

3.3.5 Recyclability Test and Breakthrough Measurements

To evaluate the CO₂ capture performance of **Sample 4**, the recyclability of **Sample 4** was tested by exposing the material under a mixed gas flow (15% CO₂ and 85% He) at 1 bar and 40 °C for 30 min and activated at 90 °C under a flow of pure He for 10 min. There are no apparent changes of CO₂ uptake during the recyclability test (**Figure 88**), suggesting that the material is a regenerable adsorbent.

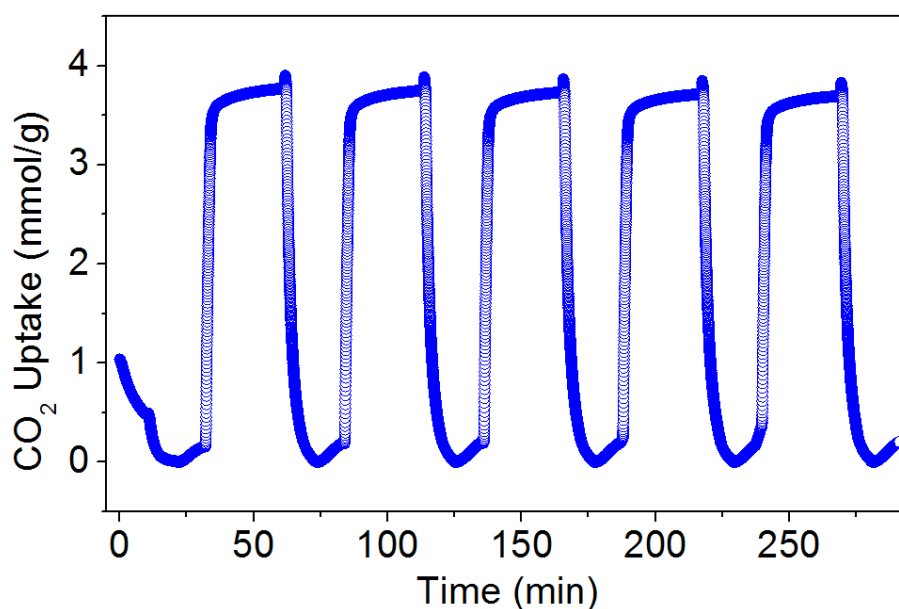


Figure 88. Recycling test of **Sample 4** on TGA equipment with the adsorption and desorption conditions set as a 15:85 CO₂/He (v/v) flow at 40 °C and 1 bar, and a pure He flow at 90 °C and 1 bar, respectively.

To further assess the CO₂ capture capability of the material in mixed gas scenarios, we conducted dynamic column breakthrough experiments on **Sample 4** with a binary gas mixture composed of 15% CO₂ and 85% N₂. The breakthrough curves indicate that CO₂ would not be detected as the end of the column until the CO₂ injection amounts were 5

mmol/g at 25 °C and 4.3 mmol/g at 40 °C. By contrast, N₂ would break through at much earlier stages at both temperatures (**Figure 89-92**). This suggests the highly selective adsorption of CO₂ over N₂ in **Sample 4**. Based on the data from breakthrough experiments, the CO₂ capacities of **Sample 4** were calculated to be 4.42 mmol/g at 25 °C and 4.28 mmol/g at 40 °C, which are comparable to the results attained in its CO₂ isotherms.

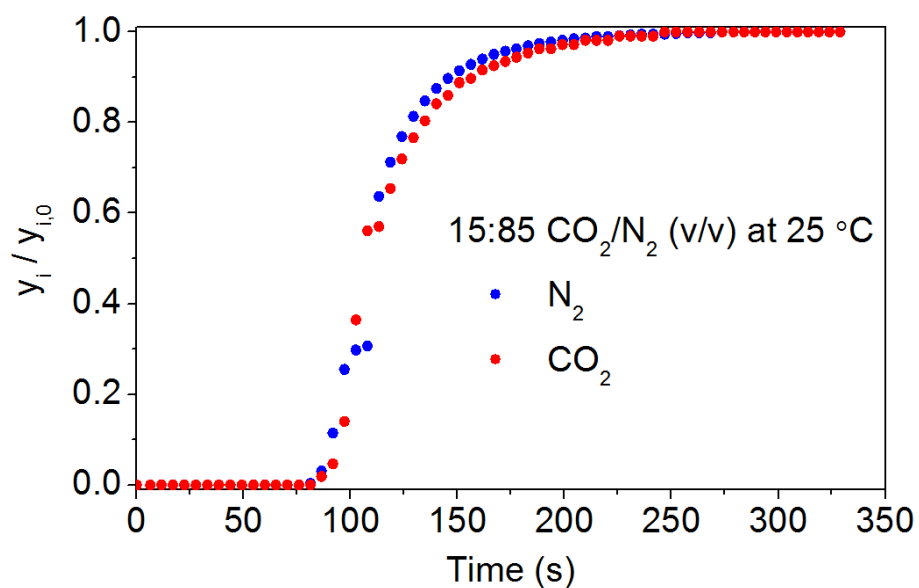


Figure 89. Breakthrough curves of an empty column for dead volume calibration at 25 °C. The gas flow is a 15:85 CO₂/N₂ (v/v) flow at 25 °C with an inlet pressure of 1.2 bar and an outlet pressure of 1 bar. y_i and $y_{i,0}$ are the molar fractions of a gas species at the inlet and outlet, respectively.

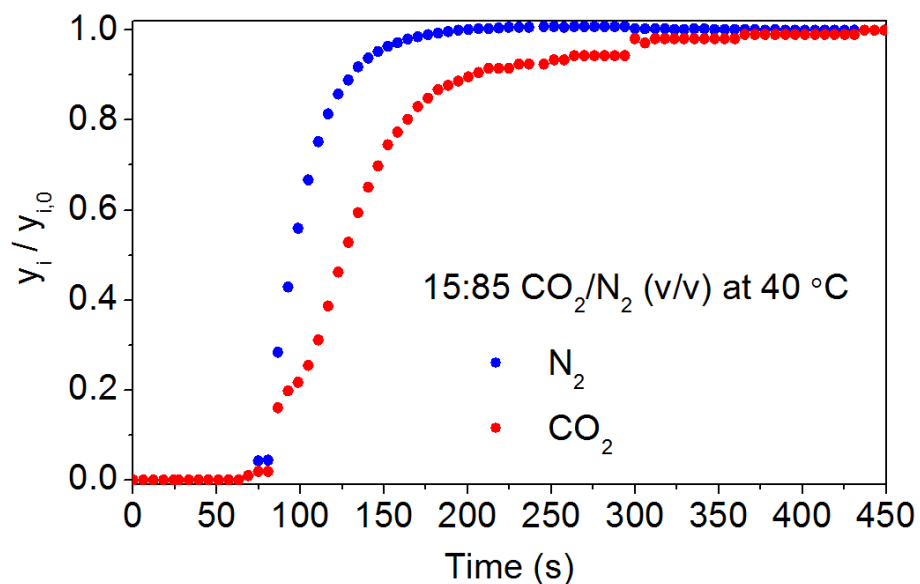


Figure 90. Breakthrough curves of an empty column for dead volume calibration at 40 °C. The gas flow is a 15:85 CO_2/N_2 (v/v) flow at 40 °C with an inlet pressure of 1.2 bar and an outlet pressure of 1 bar. y_i and $y_{i,0}$ are the molar fractions of a gas species at the inlet and outlet, respectively.

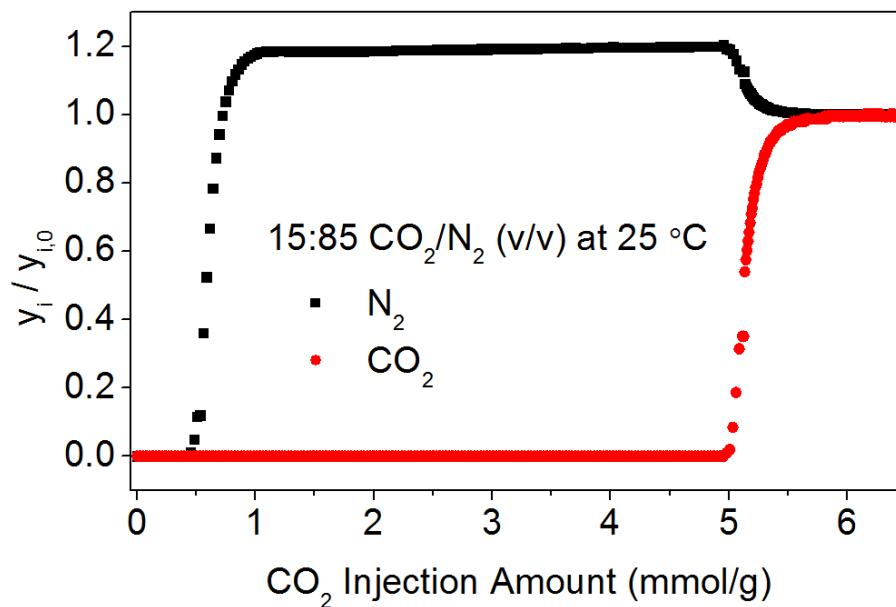


Figure 91. Breakthrough curves of a column packed with **Sample 4**. The gas flow is a 15:85 CO_2/N_2 (v/v) flow at 25 °C with an inlet pressure of 1.2 bar and an outlet pressure of 1 bar. y_i and $y_{i,0}$ are the molar fractions of a gas species at the inlet and outlet, respectively.

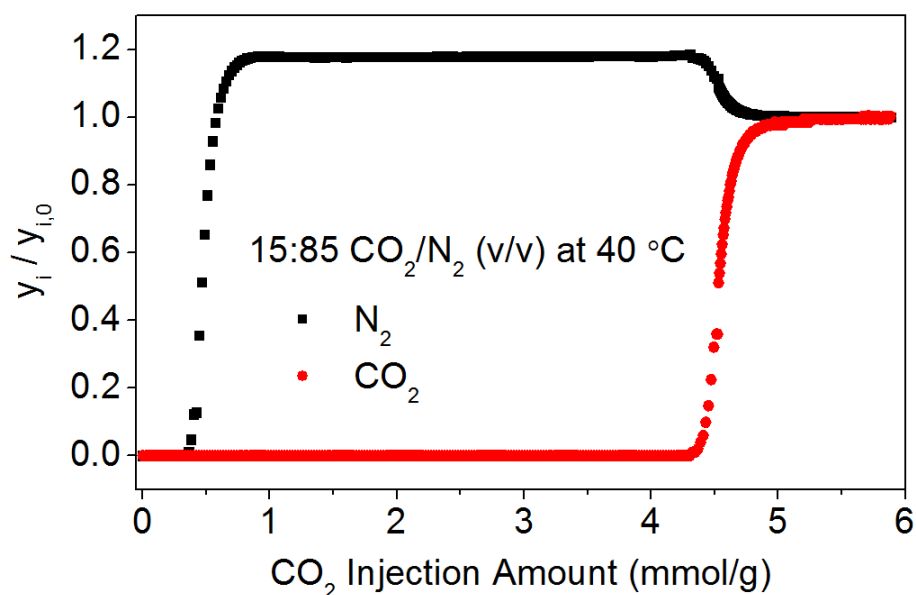


Figure 92. Breakthrough curves of a column packed with **Sample 4**. The gas flow is a 15:85 CO_2/N_2 (v/v) flow at 40 °C with an inlet pressure of 1.2 bar and an outlet pressure of 1 bar. y_i and $y_{i,0}$ are the molar fractions of a gas species at the inlet and outlet, respectively.

3.4 Conclusion

With rationally selected alkylamine and solvent species, a high loading amount of alkylamine was successfully achieved in the mesoporous and chemically stable Cr-MIL-101. This result indicates that nonpolar or less polar solvents are more favorable for the migration of alkylamine from alkylamine solutions to MOFs, because the alkylamine dispersed in less polar solvents are endowed with a higher chemical potential that drives more alkylamine molecules to be incorporated into the frameworks. The obtained material displays pure-component CO_2 uptakes as high as 4.21 mmol/g at 25 °C and 150 mbar, and 4.05 mmol/g at 40 °C and 150 mbar. No apparent decline can be discerned in its CO_2

capacity after multiple adsorption-desorption cycles. Its performance was also characterized by dynamic column breakthrough experiments, which demonstrate that the material exhibits good CO₂/N₂ selectivity in the case of binary gas mixtures.

CHAPTER IV

SUMMARY

Alkylamine-functionalized metal-organic frameworks (MOFs) are regarded as promising candidates to selectively capture CO₂ from flue gas, air, or other forms of gas mixtures, because the incorporated alkylamine molecules endow MOFs with strong CO₂ affinity, which is essential for selective CO₂ adsorption at low pressure. To date most alkylamine grafting experiments in MOFs are conducted using the coordination bonds between open metal sites (OMSs) and amine groups. This inevitably would set an upper limit on alkylamine density in MOFs and the corresponding CO₂ uptakes that can be achieved.

To overcome this obstacle, a new approach to integrating alkylamine into MOFs is explored, which exploits the Brønsted acid-base reaction between MOFs and alkylamine. In this work, the pristine MOF utilized is Cr-MIL-101-SO₃H, which has a high density of inherent sulfonic acid groups. Cr-MIL-101-SO₃H was modified with a series of alkylamines with different structures. The FTIR spectrum of the amine-modified Cr-MIL-101 with the highest CO₂ uptake was collected. The peaks at 3234 cm⁻¹ and 3147 cm⁻¹ are characteristic of the stretching vibrations of -NH₃⁺, which serve as a good evidence for the proton transfer and subsequent ionic interaction between -SO₃H and -NH₂. Moreover, experimental factors, such as alkylamine structure, alkylamine quantity, reaction time, reaction temperature, and solvent, are systematically optimized by control experiments to maximize the CO₂ uptake of the obtained material. The results reveal that apart from the alkylamine structure, the solvent species used to conduct alkylamine

incorporation is also crucial in determining the CO₂ capture performance of the generated adsorbent. This point is rationalized thoroughly from the perspective of chemical potential. Specifically, alkylamine has weaker interaction with less polar solvents than with more polar ones. This makes the alkylamine dispersed in less polar solvents comparatively unstable and have a higher chemical potential than in more polar ones. The higher chemical potential serves as a stronger driving force to push more alkylamine molecules into the frameworks and result in a higher CO₂ capture amount of the material.

Under the optimal alkylamine tethering condition, the obtained TAEA-functionalized Cr-MIL-101-SO₃H (Cr-MIL-101-SO₃H-TAEA, TAEA = Tris(2-aminoethyl)amine) has cyclic CO₂ uptakes of 2.28 mmol/g at 150 mbar and 40 °C, and 1.12 mmol/g at 0.4 mbar and 20 °C. Its isosteric heat of CO₂ adsorption calculated from the CO₂ uptake data at 0 °C, 20 °C and 40 °C is 87 kJ/mol at zero coverage. This indicates a strong interaction between CO₂ and Cr-MIL-101-SO₃H-TAEA. The low-cost starting materials and simple synthetic procedure of Cr-MIL-101-SO₃H-TAEA all suggest its advantages in large-scale production for practical applications.

The aforementioned conclusion that alkylamine structure and solvent species are very influential in the alkylamine loading amount in MOFs and their CO₂ uptakes was further applied in another mesoporous and highly stable MOF, Cr-MIL-101. With cyclohexane (CH) and TAEA as the judiciously selected solvent and alkylamine, respectively, the obtained material displays pure-component CO₂ uptakes as high as 4.21 mmol/g at 25 °C and 150 mbar, and 4.05 mmol/g at 40 °C and 150 mbar. No apparent decline can be observed in its CO₂ capacity after multiple adsorption-desorption cycles. It

also exhibits a comparable CO₂ adsorption amount in the dynamic column breakthrough experiment, in which the adsorbent was exposed to a binary gas mixture of 15% CO₂ and 85% N₂. Thus, a good CO₂/N₂ selectivity of the material was also demonstrated with this breakthrough experiment. As indicated by the atomic ratio between N and Cr of the resulting adsorbent, the integrated alkylamine molecules outnumber the OMSs in Cr-MIL-101. This implies that a portion of alkylamine is tethered through the hydrogen bonds between alkylamine molecules. Thermal stability of the adsorbent is assessed with thermogravimetric analysis (TGA), suggesting that there is no alkylamine leaching before the temperature is higher than 150 °C. Therefore, the material is confirmed to be sufficiently robust at adsorption and desorption temperatures, which indicates its good eligibility as adsorbents in real CO₂ capture scenarios.

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